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PHOTODYNAMIC THERAPY WITH PHTHALOCYANINES: A FIGHT AGAINST CANCER

(VIEWS FROM THE RECENT LITERATURE)



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PART A: Comprehensive photodynamic therapy studies

These studies contain sophisticated antitumoral experiments in which phthalocyanines are actually used as a photosensitizer. They represent the superior part of the science and they should be contacted by the authors of other, non-comprehensive studies to see if the synthesized compounds in the latter show surprising behavior.

Authors chemically coupled heptamethylphthalocyanine (a near-infrared fluorescent dye) referred to as IR-780 and cabazitaxel, a chemotherapeutic drug based on paclitaxel. Then, the authors evaluated the potential value in castration-resistant prostate cancer. Tumor-bearing mouse xenograft models were used in the distribution, antitumoral effect, and safety of the drug. The authors state that the drug was efficiently absorbed by DU145 and PC-3 cells and the coupled molecule has a significantly stronger cytotoxicity than the counterparts. Therefore, the new synthetic drug obtained by chemical coupling of two entities has potential applications in the diagnosis and treatment (Zheng et al., 2020).

As a new class of cancer therapy, near-infrared photoimmunotherapy is based on the coupling of mAb to a phthalocyanine, the product therefore coined as IR700, and then NIR light irradiation is applied along with molecular-targeted fluorescence imaging. NIR light is applied upon the tumoral cell, and the obtained conjugate causes a target-selective and rapid death of the cell and necrosis is what happens at the end. HER2-positive cancer cell lines were tested for the antitumoral effect of near-infrared photoimmunotherapy with trastuzumab conjugated to IR700. It is important to state here that some cancer cells do not die and undergo regrowth after near-infrared photoimmunotherapy in an in vivo model; this is partly because mAb-IR700 conjugate are distributed heterogeneously in the tumor (Nishimura et al., 2020).

Treatment options for malignant pleural mesothelioma (MPM) are very limited. Molecular targeted therapy for malignant pleural mesothelioma is not available for now; new targeted therapeutic options are welcome. Near-infrared photoimmunotherapy is a recent application of antibodies' specific properties directed to the tumoral cell and the toxicity of the photoabsorber is combined with NIR-light exposure. Nishinaga et al. developed an antibody—photosensitizer conjugate having a phthalocyanine dye. In a flank model, in vivo near-infrared photoimmunotherapy caused a significant reduction in both tumoral volume and the activity of luciferase. In an MPM orthotopic mouse model, phototherapy targeting podoplanin (PDPN)-targeted NIR-PIT gave way to a significant antitumoral effect. Authors suggest that phototherapy targeting podoplanin-targeted near-infrared-photoimmunotherapy is a recent treatment for malignant pleural mesothelioma, showing great promise. (Nishinaga et al., 2020).

Ahmadi and co-workers, in their review article, reported the effect of the composition of liposomes on the release of encapsulation under ultrasound (US). The synthesis of liposomes based on dioleovl phosphatidyl ethanolamine and hydrogenated soy phosphatidylcholine was reported. They also reported chloroaluminum(III) phthalocyanine tetrasulfonic acid as the near-infrared fluorochrome drug model. AlPcS4 was found to be included, in an efficient manner, in both liposomes, and this was confirmed with concentration-dependent fluorescent quenching. The liposomes were stable in serum and they did not give out any fluorescent signal. When 1.1 MHz ultrasound was applied, dioleolyl phosphatidyl ethanolaminebased liposomes led to ca. 40% drug release after 1 min and hydrogenated soy phosphatidylcholine was less active, only leading to about 20% drug release. The in vivo studies led to the conclusion that ultrasound application caused the release of the fluorescent phthalocyanine from the first type of liposome and a fluorescent signal intensity (P < 0.05) increased for 100%, where for the latter type, no signal enhancement was obtained. The study therefore describes the ultrasound-triggered release of the fluorescent phthalocyanine (Ahmadi et al., 2020).

Figure 1. The structure of chloroaluminum(III) phthalocyanine tetrasulfonic acid.

Photodynamic therapy (PDT), is an innovative and non-invasive form of the therapy of cancer, based on the photodynamic effect. The authors determined intracellular uptake, singlet oxygen production, and in vitro photodynamic potential of the nanosystem against pancreatic cancer cells, in which the nanosystem included cetixumab, peripherally octakis (tert-butylphenoxy) substituted zinc(II) phthalocyanine, and mesoporous silica nanoparticles. The quantum yield of the complex was found as 0.60 in toluene solution. The authors tested the dark phototoxicity and the photoxicity at different light fluences of several concentrations of the zinc complex by using pancreatic cells. No dark toxicity was present and the zinc complex alone showed a mild photodynamic activity, yet when included in silica nanoparticles, the phototoxic effect in the presence of

Cetuximab was much greater. EGFR receptor-expressing pancreatic cells are excellent targets for this nanomolecular vehicles. (Er et al., 2018).

Photochemical reactions usually have dramatic effects on the physical properties of molecules in a reaction. In the study by Sato et al., the authors demonstrated that NIR light induces a release from axial position, and this alters dramatically the hydrophilicity of a SiPc (IR700) dye, ultimately leading to conjugate's shape change hence its tendency for aggregation in aqueous solution. The authors proposed that this photochemistry causes a major mechanism of cellular death along with NIR-PIT, which is considered as an efficient cancer therapy that is targeted in a molecular manner.

When the binding of antibody-IR700 conjugate has effected, NIR light activation causes physical changes in the shape of antibody-antigen complexes, and this in turn provides physical stress in the cellular membrane; the transmembrane water flow increases, leading to a burst of the cell and necrosis (Sato et al., 2018).

Figure 2. The structure of silicon phthalocyanine.

Considerable attention has been gained from multifunctional phototheranostic nanoparticles, which are prepared from photosentisizing molecules and they are successfully used in tumoral phototherapies. Photothermal activity (PTT) is more encountered than photodynamic therapy (PDT) in self-assembled photosensitizing nanoparticles, in which the latter is greatly restricted. Wang and their colleagues prepared substituted zinc phthalocyanine molecules which are capable of performing PTT and PDT at the same time with 650 nm irradiation. They obtained an outstanding therapeutic effect due to the synergistic action of PTT and PDT, in which substituted ZnPc particles behave as effective and safe cancer treatment agent in clinical research (Wang et al., 2019).

Figure 3. *The structure of non-peripherally substituted zinc phthalocyanine.*

In a review by Roguin et al., the researchers mention of photodynamic therapy applications, exhaustively using zinc phthalocyanine derivatives and they have not yet been approved as photosensitizers for clinical therapy. The authors gave physicochemical properties of ZnPc derivatives, biological results of them in the in vitro and 3D cultures, chicken chorioallantoic membranes, and tumor-bearing mice. All these media constitute of more complex models. The authors also mention of combined therapy cases including zinc phthalocyanines and their clinical trials that are currently available (Roguin et al., 2019).

Nanodrugs which are capable of performing more than one task are highly desired in the treatment of cancer. There are few examples of two-dimensional nanomaterials in the form of coordination nanosheets. Authors propose phthalocyanine-based coordination nanosheets in which the photoactive entity is manganese(II) phthalocyanine tetracarboxylic acid and hyaluronic acid is coated on their surface, forming a nanosheet structure abbreviated as MnPc@HA. The near-infrared photothermal effect, along with photothermal conversion, had an efficiency of about 73% and this value is superior to those reported in the literature. MnPc@HA nanosheets are capable of enduring high drug-loading capacity for curcumin, a chemotherapeutic agent (Zeng et al., 2019).

Elevated glutathione (GSH) levels are responsible for hindering targeted therapeutic efficiency. Wei and their co-workers studied on a non-symmetrical zinc phthalocyanine, whose brief name was ZnPc-C8-

Len, and they coupled lenvatinib (a VEGFR inhibitor) to it through an alkyl chain in a hope that multidrug resistance was reversed and enhanced antitumoral therapy was achieved. With irradiation, ZnPC-C8-Len produces reactive oxygen species that depleted intracellular glutathione and causing apoptotic cellular death. ZnPc-C8-Len had fluorescent features and the changes could be monitored in vivo, making it possible to decide when to apply the therapy (Wei et al., 2019). From a chemist's viewpoint, the molecule ZnPc-C₀-Len is not possible to occur as the amide counterpart of the molecule is not capable of acting as an amine, whereas the other molecules' formations are feasible.

Figure 4. The structure of ZnPc-C8-Len.

Because telomerase plays a crucial role in the unlimited proliferation of cancer cells, inhibiting this enzyme is a promising method, by stabilizing G-Quadruplex DNAs by guest molecules in a host-guest fashion. To see if a compound is eligible as a telomerase inhibitor, it is necessary to assess whether this compound is selective for quadruplex DNA over the duplex one. In the study by Ramos et al., tetrapyrrolic compounds having four or eight quaternized groups were reviewed, because they are considered as G-Quadruplex stabilizing ligands. It was concluded that a balance

exists between the numbers and positions of the quaternized groups in the phthalocyanine; it is an important asset for selecting G-Quadruplex structures over duplex DNA. A phthalocyanine structure reported by Ramos et al., having four or eight peripheral positive charges, led to a high selectivity and affinity for G-Quadruplex over duplex DNA and aggregated in the nuclei of UM-UC-3 bladder cancer cells (Ramos et al., 2019).

Figure 5. Methoxypyridinium phthalocyanine.

Non-symmetrical cobalt phthalocyanines with two sulfonic acid sodium salt entities were interacted with calf thymus DNA and there were two types of DNA binding. CoPcS₂ was found to bind to the phosphate end of DNA and formed aggregates on the periphery of the DNA helix. In this interaction, DNA played the role of a matrix and the number of stacking interactions increased. The second type of cobalt-DNA interaction takes some time to manifest itself. Cobalt's coordination sphere allows some atoms from DNA to coordinate to the metal ion and the whole structure is reorganized. The minor groove of DNA is not affected from the interaction. The N7 guanine, the most important part of the DNA, is unaffected, therefore the intercalation model is completely excluded. The phthalocyanines' planes showed an orientation that was normal to the DNA's axis. DNA's rigidity did not change after the interaction as suggested by the optical anisotropy and intrinsic viscosity of DNA (Kasyanenko et al., 2019).

Figure 6. Cobalt phthalocyanine disulfonic acid disodium salt.

For photodynamic therapy (PDT), self assembly of phototheranostic nanobodies are of great importance along with many advantages. A novel method was designed for size-tunable self-assembly of nano-sized photosensitizers for PDT. A silicon(IV) phthalocyanine and biotin formed a conjugate in which the size was governed by Cremophor EL surfactant. The particle size of the prepared nanostructures also dominates the fluorescence and generation of reactive oxygen species (ROS). Tumoral over-expressing biotin receptors cause disassembly of the nanostructures and the quenching be recovered. Self-assembly structures were capable of invading HepG2 cells in a biotin-receptor pathway and biotin-receptor-triggering fluorescent recovery was obtained in a scope of cells. In the nanochemical series, the smallest member managed to provide the highest photocytotoxicity, improved intracellular fluorescence, highest ratio for reactive oxygen species, the best localization in the subcellular scale, tumor targeting in vivo, and the best photodynamic efficiency. (D. Li et al., 2019).

For photodynamic therapy, phthalocyanines are considered to have good potential, however their poor solubility, high aggregation tendency, and low specificity to tumors are key problems that must be solved for successful applications. The authors have synthesized a biotin-containing silicon(IV) phthalocyanine molecule to be a water-soluble photosensitizer capable of tumor-targeting. Axial ligation decreased the aggregative behavior of Pcs. In vitro experiments provided the information that the compound accumulates successfully in biotin receptor-positive HeLa cells. Since it is water-soluble and does not aggregate much, the compound was recommended for a xenograft tumor model. In vivo imaging and tissue distribution studies revealed that the compound selects, predominantly, the tumor cells in a tumor-bearing mice model (tolerable signals coming from the vicinity of the cell exist). In vivo, under irradiation, the compound was successful in depressing the progression of the tumor. A 14-day treatment showed that the volumes of tumor were smaller than the beginning size (K. Li et al., 2019).

Figure 7. *Biotin-containing silicon(IV) phthalocyanine.*

Matlou and colleagues investigated the photodynamic therapy activity of cinnamic acid-containing zinc and indium phthalocyanines and they coupled the carboxylic acid bond to amino-functionalized magnetic nanoparticles. In terms of triplet and singlet oxygen quantum yields, non-symmetrical ZnPc displayed a better performance than the symmetrical analogues. When the phthalocyanine contained amino-functionalized magnetic nanoparticles, the triplet quantum yield was seen to increase. The complexes displayed an increased in-vitro phototoxicity against MCF-7 cells when the authors increased the drug concentration. At a concentration of $80~\mu g/mL$, the conjugates gave a higher singlet oxygen quantum yield and their cytotoxic effects on cancer cells were higher in the presence of light (Matlou et al., 2019).

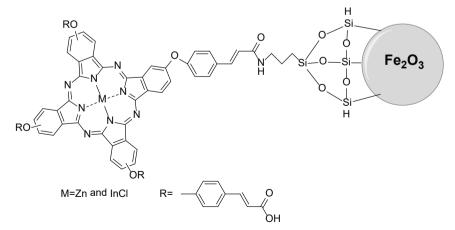


Figure 8. Cinnamic acid-containing zinc and chloroindium phthalocyanines and their chemical linkage to amino-functionalized magnetic nanoparticles.

Authors describe the last five-year history of phthalocyanines and discuss about designing these macrocycles for medicinal applications. This literature is a good starting point for those who want to learn more about the latest discussions on Pcs in PDT (X. Li et al., 2019).

The problem of delivering poorly soluble drugs in the diagnosis and treatment of cancer cells could be solved by the application of multifunctional nanodots. The authors applied a cryodesiccation-driven crystallization approach for preparing the nanocrystallites of water-insoluble photosensitizer (ZnPc). When the surface was modified with nanodots along with Pluronic F127 and folic acid, the resulting complex gains excellent water-solubility. Under near-infrared photoexcitation at 808 nm, the nanodots showed the production of singlet oxygen (Ding et al., 2019).

Yu and colleagues (2019) used zinc phthalocyanine for targeted photodynamic therapy and they used a tumor-directing entity and epidermal growth factor receptor (EGFR) binding peptide for the coupling required. As a result, the biocompability and tumoral selectivity increased. This phthalocyanine-peptide conjugate was investigated in terms of its biotoxicity, photophysical properties, cellular uptake, and in vivo biodistribution. Cells like A431 which over-express EGFR were more vulnerable to the Pc's effects, but low-EGFR-expressing tumoral cells like human breast adenocarcinoma MCF7 have little cellular uptake. When the authors administered the compound to the A431-tumor-bearing mice, the compound, namely Pc-GE11 conjugate, led to a much higher tumoral accumulation than the non-targeted control. According to the results, Pc-GE11 conjugate is a good candidate for targeted photodynamic therapy and bioimaging (Yu et al., 2019).

Chloroaluminum phthalocyanine tetrasulfonic acid (AlPcS₄) is a synthetic aromatic macrocycle. Since it could be activated with a far red light, it can penetrate deeply into the tissue, along with its low photobleaching and high quantum yields and it is also stable. Pulmonary cancer has a ratio of 1 to 5 in all cancer-related deaths worldwide. Crous et al. explored the photochemical properties of chloroaluminum phthalocyanine tetrasulfonic acid, its uptake into pulmonary cancer, intracellular localization of it and the photodynamic application on the cancer (A549 cells were employed). ClAlPc tetrasulfonic acid was reported to be a robust molecule, localizing in the intracellular organelles, including mitochondrion and lysosomes. PDT application revealed that cellular death increased and cellular proliferation and viability decreased. Under in vitro conditions, the photosensitizer proved to be effective, but photodynamic efficacy also relies upon tumor vasculature and tumor-specific accumulation also must be considered when a clinical application is being planned. Photodynamic

therapy could be taken into account as an adjunct therapy, until standard protocols for a number of tumoral types with a relevant photosensitizer is validated (Crous et al., 2019).

Nitrogen monoxide (NO) generation is important, especially when it is combined to photobiomodulation and photodynamic therapy. Negri et al. stated that photobiomodulation, as combined with photodynamic therapy, could be a beneficial option of cancer treatment. The authors used two ruthenium compounds, namely ruthenium(II) phthalocyanine and trans(nitrosyl)(nitro)(phthalocyaninato) ruthenium, and around 660 nm in the UV-Vis spectrum there was a band. 0.5 µM ruthenium (nitrosyl)(nitro) (phthalocyanine) compound caused the light irradiation at the O band reduce viable A375 (human melanoma) cells to ca. 50% against the control being ruthenium phthalocyanine. Between singlet oxygen and nitric oxide, there possibly might a synergistic effect. In a combination of these compounds, PDT at 660 nm and PBM at 850 nm caused more photocytotoxicity. A possible explanation was that PBM increased the cellular metabolism (that is, production of ATP), and consequently, RuPc compounds are being taken inside the cell and apoptosis was more efficient. In the field of photodynamic therapy, metal-based photosensitizes, as combined with light therapy, seems to introduce an advanced field. (Negri et al., 2019).

The general insolubilities of phthalocyanines poses a complexity in the applications of them. Lu et al. provided a way for solubilizing the ZnPc and a mixed micellar structure was introduced. They employed sodium deoxycholate as the first and D-alpha tocopheryl polyethylene glycol succinate as the second micellar structures. The phthalocyanine, which is highly hydrophobic, reside in the hydrophobic part of the micelles, through a supramolecular self-assembly. A series of mixed micellar architectures containing ZnPc at the core were prepared and their physicochemical properties, drug release characters, photostabilities, photodynamic activities, and cellular viability. Irradiated by a 650-nm red laser, the ZnPc-loaded micellar structure produced singlet oxygen with 1,3-diphenylisobenzofuran as the probe.

The mixed micellar organizations had higher singlet oxygen quantum yields than the control compound (ZnPc), and three of them in the series had very promising singlet oxygen generation ability. The micellar structure was stable in terms of photodynamic therapeutic timescale, along with a small amount of photobleaching effect was observed after 20 minutes of irradiation. A sustained release profile was suggested for the mixed micellar structure's in vitro drug release behavior. At the same concentration, the mixed micellar structure was found to be more toxic than unbonded zinc phthalocyanine against A549 cells (Lu et al., 2019).

Figure 9. Structures of SDC, TPGS, and ZnPc.

PART B: STUDIES, WHICH MAY PROCEED TO COMPREHENSIVE PHOTODYNAMIC THERAPY AFTER SOME EFFORT

These reports contain some basic measurements of synthesized compounds which will guide to sophisticated experiments of cancer therapy, given the opportunity. There are many studies like those cited below, and they, we believe, deserve further study after a good collaboration.

At the peripheral and non-peripheral positions, the researchers prepared piperidin-containing phthalocyanines and they then quaternized the products, leading to water-soluble molecules. When they irradiated the compounds, they led to significant cleavage effects against plasmid DNA. The phthalocyanine derivatives with non-peripheral substitution were more successful in terms of cleaving the DNA, in the presence of a series of oxidizing agents. The mentioned compound was suggested as a potent photosensitizer for photodynamic therapy due to its potential to damage DNA with irradiation applied (Özel et al., 2019).

Figure 10. *Piperidin-containing phthalocyanines and their quaternized forms.*

Yalazan and coworkers examined tosylated 4-morpholinoaniline-bearing tetrasubstituted phthalocyanines at peripheral positions and their behaviors as anticancer agents. They prepared metal-free, copper, zinc, cobalt(II), and magnesium phthalocyanines. ZnPc, CoPc, and MgPc were accepted into more advanced PDT applications, say oncologic and molecular biologic studies like pBR322 plasmid DNA cleavage on agarose gel electrophoresis. ZnPc was reported to cleave pBR322 plasmid DNA with irradiation. An MTT assay towards human colorectal (HCT-116) and cervical (HeLa) cells was conducted with ZnPc, allowing to investigate its cytotoxic/phototoxic properties. Cellular viabilities for ZnPc against HCT-116 was $73 \pm 1.6\%$ and against HeLa was $65 \pm 5.5\%$ at 100 mM concentration, with applied irradiation (Yalazan et al., 2020).

Figure 11. Structures of tosylated 4-morpholinoaniline-bearing tetrasubstituted phthalocyanines.

M= 2H, Zn, Cu, Mg, Co

A study by Şen et al. provides a report about phthalocyanines having benzimidazole units and their unmetallated, zinc, chloroindium(III), and chlorogallium(III) phthalocyanines were prepared. Their photodynamic efficacies and photophysicochemical properties were studied in dimethyl sulfoxide. The metal-free phthalocyanine presented a higher fluorescent quantum yield and its fluorescent lifetime was also higher than that of the metallophthalocyanines investigated. Chloroindium(III) phthalocyanine, in the phthalocyanine series studied, yielded the highest values of singlet oxygen ($\Phi\Delta$) and triplet state quantum yield (Φ T) (Sen et al., 2019).

Figure 12. Phthalocyanines having benzimidazole units.

Şen and Nyokong synthesized a novel silicon(IV) phthalocyanine bearing, at axial positions, a palladium(II)-Schiff base complex. This silicon(IV) phthalocyanine was assayed for its photophysical and photochemical properties and its dimethylsulfoxide solution was subjected to UV-Vis absorption, fluorescence spectrophotometry, quantum yield of singlet oxygen, quantum yield at triplet state, and excited state lifetimes. The complex's very low fluorescent behavior led to an efficient intersystem crossing, high triplet lifetimes, and high singlet oxygen quantum yields. The compound displayed excellent photodynamic activity, along with a 99.94% reduction percentage and 3.26 as a log red value (Sen & Nyokong, 2019).

Figure 13. A silicon(IV) phthalocyanine bearing two palladium(II)-Schiff base complexes at axial positions.

A study by Demirbaş et al reported the preparation of a novel peripherally triazole-substituted ZnPc phthalocyanines synthesized from four different precursors. Using comparative method in calculating the fluorescence quantum yields is common in this type of studies and fluorescence quenching properties of the studied phthalocyanines was reported. In addition, photodegradation quantum yields of the complex was reported. Singlet oxygen quantum yields of the compounds were calculated via the comparative method in the presence of 1,3-diphenylisobenzofuran as the quenching entity. The quantum yield values of the complex was in the range of 0.8 – 0.85 and the compared unsubstituted ZnPc's quantum yield was 0.67. The compound with a tolyl group as the substituent proved the highest singlet oxygen quantum yield and this high value is very rare (Demirbaş et al., 2019).

A Bulgarian-Turkish collaboration yielded a paper in which Aliosman et al. prepared an octasubstituted zinc phthalocyanine containing 4-tyrosylamidophenyl substituents at the periphery and introductory photodynamic activity tests were conducted. The team also prepared the tetrasubstituted phthalocyanine and unsubstituted ZnPc for comparison. Absorption, fluorescence, singlet oxygen photochemistry, and photostability, and in vitro phototoxicity were investigated. The team obtained an improved photophysicochemical result for the substituted ZnPc complexes. The octasubstituted ZnPc molecule is more advantageous since its water solubility is greater, possibly leading its way to more sophisticated PDT tests (Aliosman et al., 2019).

Figure 14. Octasubstituted zinc phthalocyanine containing 4-tyrosylamidophenyl substituents.

Aluminum octacarboxyphthalocyanine and nanoparticles were studied together to see the interaction between them. When phthalocyanine molecules are adsorbed on the nanoparticle, yet only to a certain limit, the non-radiative energy transfer efficiency was seen to be increased. When the number of phthalocyanine molecules are increased further, their concentration resulted in a decrease of its sensitized fluorescence due to the dimerization of dye molecules on the nanoparticle surface. When subjected to infrared irradiation, phthalocyanine molecules in the hybrid complex generated singlet oxygen. The observed effects are of interest in regard to the targeted search for new components of efficient third generation hybrid photosensitizers (Gvozdev et al., 2019).

In the work by Köksoy et al., the novel phthalonitrile precursor compounds having 7-oxy-3-(p-thiomethylphenyl) coumarin substituents and their corresponding peripheral/nonperipheral phthalocyanines with zinc and chloroindium metals at the core were synthesized and purified. The photochemical and photophysical properties of the complexes were

examined in dimethylformamide solutions (Köksoy et al., 2019).

Figure 15. Zinc and acetatoindium phthalocyanines containing (methylsulfanyl) coumarin moieties at peripheral positions.

Majeed et al. have prepared four carbazole-containing phthalocyanines and performed their essential spectral characterizations. Owing to the charge transfer effect and prolonged excited state lifetime of 5 ns and high singlet quantum yield of 0.825 were obtained (Majeed et al., 2019).

M= 2H, Mg, GaCl, InCl

Figure 16. *Carbazole-containing phthalocyanines.*

Keleş et al. worked about pyridine-containing silicon(IV) phthalocyanine, and its naphthalocyanine counterpart, along with their quaternized, water-soluble derivatives. The cytotoxicity, DNA interaction, topoisomerase inhibitory property, and antitumoral effect against five carcinoma cell lines were examined. Among the five carcinoma cells, the CC50 values of the water-soluble compounds were determined as 2.15-52.97 μM_{\odot}

Against the carcinoma cell lines, the water-soluble silicon phthalocyanine compound displayed the SI values of 4.51, 1.43, and 3.66, respectively. With flow cytometry, A549 cells suffered cellular cycle arrest and water-soluble silicon phthalocyanine and naphthalocyanine compounds led to cellular cycle arrest at G0/G1 phase. The water-soluble silicon phthalocyanine is active against pulmonary, breast, and melanoma cancer lines with high selectivity (Keleş et al., 2019).

Figure 17. *Pyridine-containing silicon(IV) phthalocyanine.*

CONCLUSION

There are two main types of studies in the literature. One of them uses phthalocyanines as active agents for singlet oxygen generation and diminishing tumor volumes, while the other type of studies investigates photophysical and photochemical properties of phthalocyanine derivatives as candidates for real antitumor studies. We hope that the endeavors and molecules synthesized in the latter reach the former, therefore the library of molecules increase in the fight against cancer.

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INTRODUCTION

Zinc oxide (ZnO), an n-type semiconductor with a wide optical band gap of about 3.3 eV at room temperature, has high optical transmittance and low electrical resistivity [1, 2]. It is also an environmentally friendly material. Thanks to these superior features, it is a very popular material with a wide range of applications, especially in varistor, solar cell, sensor, photodiode and transistor applications [3-10]. It is possible to enhance the physical properties of ZnO, such as electrical, optical, morphological, magnetic or mechanical properties, by using different growth parameters or by doping a suitable element. ZnO films can be produced by using various methods, which are requiring vacuum or not, in accordance with the place and purpose of use. These methods include both expensive systems [11-13] and low-cost solution methods like electrodeposition [14], spraying [15], chemical bath deposition [16] and sol-gel [17, 18]. However, the quality of the films produced using solution methods is generally poor, and consequently films with the desired physical properties are not always obtainable. In film production stage, as well as the method-dependent film deposition parameters, the solution parameters play also an important role in film quality. Therefore, the properties of solution play very important role in production of high-quality ZnO film.

Ultrasounds are sound waves that have frequencies outside the human hearing threshold limits (between 20 kHz and 1 GHz). Since it does not harm the living tissue and creates heat and pressure in the applied area, it is widely used in both medical and diagnostic fields. In the industrial field, it is used in powdering metals, removing rodents from the environment, preparing very fine particle solutions, drying foods, sterilizing milk and much more [19-25]. If high-intensity ultrasound waves are applied to a liquid, they create physical, chemical and biochemical effects in the environment in which it advances depending on the amplitude and frequency of the ultrasound waves applied. [26-28]. Low frequency (20 kHz-100 kHz) and high intensity power ultrasound waves create cavitation as it progressives in a medium. Cavitation, as soon as ultrasound waves cause dilution of the particles in the medium, a void-shaped interrupt occurs. This void is filled with saturated vapor of the medium. During compression, the vapor condenses, and the cavity is disrupted by the effect of increased pressure. When the compression ceases, shock waves occur in the surrounding liquid, which evaporates rapidly in the medium. As a result of this phenomenon called cavitation, over 1000 atm pressure and a large energy are released. This energy heats the area of the bubbles and causes chemical reactions. In cavitation bubbles, a very high temperature (about 5000 K) and pressure (estimated 500 MPa) occurs in their structures at the time of explosion. Temperature, frequency and amplitude are the most important parameters affecting cavitation. While ultrasound waves are progressing in a medium, it progresses by performing compression and relaxation. In a few acoustic cycles, bubbles form, develop and rise. This bubble rise provides energy to the nucleation process as early as possible. Nucleation occurs at the lowest super-saturation, where crystallization occurs, where redissolution occurs in the suspension.

In the available literature, there are many studies where solutions are prepared by sol-gel method to produce ZnO films [9, 17, 29-32]. In these studies, silicone or glass substrates were used to produce films, and dip coating or spin-coating techniques were preferred. Recent years, there have been also a few studies in which the prepared solutions were sonicated with ultrasound waves before they were coated onto substrate [33-36]. In these studies, the sonication time applied to the solutions prepared using ultrasonic water bath was kept constant and preferred as 30 minutes or 1 hour. Sarip et al. [37] reported that the solutions were obtained by sol-gel method using different precursors (zinc nitrate hexahydrate and zinc acetate dehydrate), and then sonication process was used different durations (0, 30s, 90s and 120s) to the prepared solutions. They produced ZnO films using spin-coating. As substrate, they used glass substrates seeded ZnO by chemical bath deposition method. They suggested that the ZnO films produced by their process have good c-axis orientation and optical properties.

In the present study, ultrasound waves were applied to the solutions at different durations, prepared via sol-gel method. Then, ZnO films were produced using these solutions by spin-coating technique. It was investigated how sonication duration affects the structural, morphological and optical properties of the produced films. In particular, microstructural, optical and dispersion parameters were studied in detail.

EXPERIMENTAL

All the used chemicals were commercially purchased and directly used without any further purification. The solutions were prepared by the sol-gel method. Zinc (II) acetate dihydrate $(Zn(CH_3COO)_2.2H_2O; ZnAc; \%99.999$ purity; Sigma-Aldrich), 2-Methoxyethanol $(C_3H_8O_2; MEt; \ge 99.9\%$ purity; Sigma-Aldrich) and monoethanolamine $(NH_2(CH_2)_2OH; MEA; \ge 99\%$ purity; Merck) were used as starting salt, solvent and stabilizer, respectively. Solution (0.5M), used for producing the ZnO film, was prepared by using ZnAc salt, MEt solvent and MEA stabilizer.

To obtain a clear, transparent and free of precipitate solution, the prepared solutions were stirred separately in a magnetic stirrer at 60 °C during 2 hours, then heater was turned off and stirring continued about 1 hour until it dropped to room temperature. Lastly, ultrasound waves were applied without-waiting to the prepared solution. Sonication was applied to 10 ml of mixture in a glass vessel using a 20 kHz 700 W ultrasonic horn (QSONICA Q700 Sonicators) at an amplitude of 50% (700 W) for 1, 5 and 15 minutes. The during of sonication process, the temperature was maintained constant at 296.15 K, using a cooling jacket. The schematic sonication set-up is given in Fig. 1.

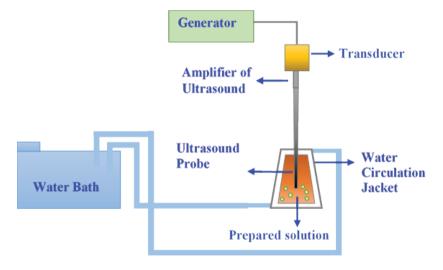


Figure 1. The schematic ultra-sonication set-up.

After the solutions (with and without sonication) were prepared, film production stage was immediately began. The chemically cleaned borofloat substrate (about 1x1 cm) was placed on the sample holder of the spin-coater (LAURELL WS-400B-6NPP/LITE), dripped from the prepared solution to a sufficient amount of substrate, and then it was spun at spin speed of 3000 rpm during 30 s. The coated wet substrate was dried at 300 °C during 10 min in a tube furnace after each coating. The spin coating—drying process was repeated ten times to obtain a 10-layer film. For better crystallization, ZnO film was annealed at a temperature greater than the preheating temperature. In order to prevent possible cracks on the film surface, the film was kept in the furnace for 20 minutes, until the temperature increased from 300 °C to 600 °C without removing the film in the furnace after preheating. Finally, the film was annealed by keeping

the temperature of 600 °C during 60 min, and the film was removed from this furnace when the temperature dropped to room temperature.

In order to check the crystalline phase and crystallite orientation of the produced ZnO films were carried out using BRUKER D8 Advance Xray powder diffractometer equipped with $Cu_{K\alpha}$ radiation (λ =1.54059 Å). Measuring interval of X-ray diffraction (XRD) measurement was 30° < $2\theta \le 60^{\circ}$, scan speed was 0.02° s⁻¹ at a divergent slit of 0.5 mm width. The XRD sample-holder was rotated 15 rpm during measurements. The surface uniformity was investigated by the roughness analysis of the atomic force microscopy (AFM) images of the produced ZnO films performed with Park Systems XE-100 model AFM. The surface roughness parameters were determined by the XEI version 1.7.1 software. The transmittance and diffuse reflectance spectra of the performed by SHIMADZU produced ZnO films were spectrophotometer in the wavelength range of 300-800 nm. Before taking the transmittance measurements, the background was taken using a precleaned blank borofloat glass substrate. A barium sulfate (BaSO₄) powder was used as a reference while performing diffuse reflectance measurements.

RESULTS AND DISCUSSION

Structural Parameters

Crystal structure analysis was determined by the XRD data of the produced ZnO films. XRD patterns of these films are displayed in Fig. 2.

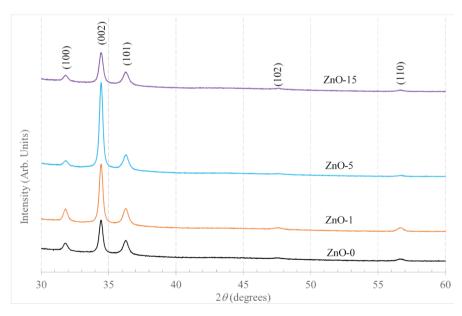


Figure 2. X-ray diffraction patterns of the ZnO films.

The XRD peaks belong to ZnO hexagonal wurtzite within Zincite-type structure (JCPDS-ICDD card Pdf number: 036-1451). Depending on the sonication duration, peak intensity and widths have changed a little. The observed 2θ values match to (100), (002), (101), (102) and (110) diffraction planes for ZnO. Obviously, increasing the duration of sonication is useful to significantly enhancement the crystalline quality of ZnO film. However, high sonication duration (15 min) deteriorates the crystalline quality again.

For all ZnO films, the angle of diffraction (2θ) and lattice spacing (d) of the phases specified along with the related (hkl) planes are given in Table 1. Structural parameters such as texture coefficient (TC_{hkl}) , lattice constants (a and c), Zn-O bond length (L), O-Zn-O dihedral angles (a and $\beta)$, volume of the unit cell (V), strain (ϵ_z) , stress (σ_c) , average crystallite size (D_{ave}) , and dislocation density (δ) values can be calculated using XRD spectrum data. The XRD data of ZnO films were analyzed using the Diffrac.EVA software program.

Table 1. 2θ , d ,	γ , D_{ave} and α	δ values of the	produced ZnO	films.
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Film Code	(hkl)	2θ (degree)	d (Å)	γx 10 ⁻³ (radian)	D _{ave} (nm)	δx10 ⁻⁴ (nm ⁻¹)
	(100)	31.762	2.8164			
ZnO-0	(002)	34.411	2.6054			
	(101)	36.247	2.4775	5.059	30	11.14
	(102)	47.610	1.9094			
	(110)	56.599	1.6256			
	(100)	31.772	2.8155			
ZnO-1	(002)	34.412	2.6053			
	(101)	36.242	2.4778	4.902	31	10.46
	(102)	47.621	1.9089			
	(110)	56.611	1.6253			
	(100)	31.786	2.8143			
ZnO-5	(002)	34.414	2.6052			
	(101)	36.245	2.4777	4.797	32	10.01
	(102)	47.634	1.9084			
	(110)	56.656	1.6241			
	(100)	31.792	2.8138			
ZnO-15	(002)	34.416	2.6050			
	(101)	36.251	2.4773	5.478	28	13.06
	(102)	47.649	1.9079			
	(110)	56.638	1.6245			

The preferential growth orientation belonging to the crystal structure is determined by following expression [38]:

$$TC_{hkl} = \frac{I_{hkl} / I_{hkl}^s}{n^{-1} \sum_{n} I_{hkl} / I_{hkl}^s}$$

where I_{hkl} , I_{hkl}^s and n denote peak intensity corresponding to (hkl) planes, the reference peak intensity of the randomly oriented crystallites and the number of the diffraction peaks taking into account, respectively. TC_{hkl} values for five peaks were calculated using I_{hkl}^s , of 57, 44, 100, 23 and 32 (according to the 36–1451 JCPDS-ICDD card) for the (100), (002), (101), (102) and (110) peaks, respectively. Fig. 3 displays the calculated TC_{hkl} values of the produced ZnO films. As can be seen from this figure, TC_{002} is greater than 2 for all the films. As the sonication time increases, the TC_{002} value also increases. However, a long sonication time (15 min), causes the lowering of TC_{002} values. The best crystallizing film is ZnO-5.

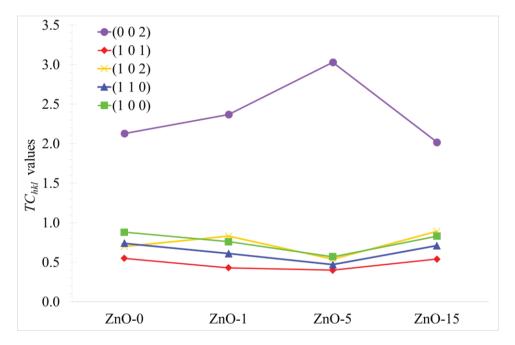


Figure 3. The calculated TC_{hkl} values of the produced ZnO films.

To calculate the lattice constants of the films, formed in hexagonal structure, the following equation [39] was used

$$\frac{1}{d_{hkl}^2} = \frac{4}{3} \left(\frac{h^2 + hk + k^2}{a^2} \right) + \frac{l^2}{c^2}$$

where d_{hkl} is distance between planes corresponding to (hkl). The calculated lattice constants are presented in Table 2. It can be seen from the table that the increase in sonication duration causes the lattice shrinkage.

Parameter	ZnO-0	ZnO-1	ZnO-5	ZnO-15
a (Å)	3.2521	3.2511	3.2497	3.2491
c (Å)	5.2108	5.2106	5.2103	5.2100
$V(\mathring{\mathbf{A}}^3)$	47.73	47.70	47.65	47.63
L(nm)	1.9792	1.9788	1.9782	1.9779
α (degrees)	108.45	108.47	108.49	108.49
β (degrees)	110.54	110.52	110.50	110.50
$\epsilon_z (x10^{-4})$	7.95	7.67	7.11	6.54
σ_c (x10 ⁸ Nm ⁻	-1.80	-1.74	-1.61	-1.48
2)				

Table 2. The calculated lattice constants, volume, Zn-O bond length and angles, strain and stress values for the produced ZnO films.

Wurtzite structure has basic hexagonal symmetry, and a schematic illustration of this structure for ZnO is displayed in Fig. 4. Each unit cell in the structure have four atoms. The positions of these atoms in Cartesian coordinates are (0,0,0), (0,0,uc), $a\left(\frac{1}{2},\frac{\sqrt{3}}{6},\frac{c}{2a}\right)$, and $a\left(\frac{1}{2},\frac{\sqrt{3}}{6},\frac{(u+0.5)c}{a}\right)$. The parameter of u is related to crystal structure, and for wurtzite structure, it is expressed as [40]

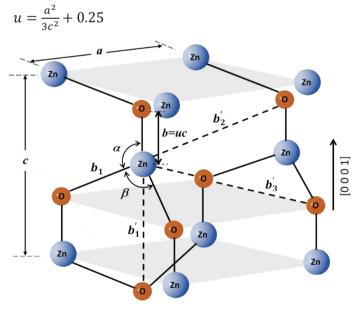


Figure 4. Schematic illustration of a wurtzitic ZnO structure with lattice constants, the nearest-neighbor distance, O-Zn-O bond dihedral angles.

The nearest-neighbor bond length (L) between Zn and O is calculated using the below equation [41, 42]:

$$L = \left(\frac{a^2}{3} + (0.5 - u)^2 c^2\right)^{1/2}$$

the L values decreased with increasing sonication duration.

The bond angles, which are α and β , are given by [2]

$$\alpha = 0.5 \pi + \arccos\left(\left[\sqrt{1 + 3(c/a)^2(0.5 - u)^2}\right]^{-1}\right)$$

and

$$\beta = 2\arcsin\left(\left[\sqrt{4/3 + 4(c/a)^2(0.5 - u)^2}\right]^{-1}\right)$$

These angles are 109.47° in ideal crystal for a wurtzitic ZnO structure. The calculated values of the produced ZnO films are tabulated in Table 2. These values are varied from 108.45° to 110.54°.

The volume of the unit cell for the hexagonal structure can be calculated using the below equation [2]

$$V = \frac{\sqrt{3}}{2} a^2 c$$

The calculated *V* values for the produced ZnO films (seen in Table 2) indicates that the volume of the unit cell reduces as the sonication duration increases.

Strain, which is deformation in crystalline structure, is given by the following expression [43].

$$\epsilon_Z = \frac{d - d_0}{d_0}$$

where d_0 is the distance between the standard planes of the bulk material $(d_{002}=2.60332 \text{ Å for ZnO})$. To calculate ϵ_z values for (002) plane of the ZnO films, Eq. (7) was used, and the calculated values are given in Table 2. It is clearly seen from this table that the ϵ_z values increases with increasing the sonication duration.

For hexagonal structure, stress parallel to the crystal surface is expressed as follows [44]:

$$\sigma_{c} = \left(C_{13} - \frac{C_{33}(C_{11} + C_{12})}{2C_{13}}\right)\epsilon_{z}$$

where C_{II} , C_{I2} , C_{I3} and C_{I4} are elasticity constants, which are $2.097 \times 10^{11} \text{ Nm}^{-2}$, $1.211 \times 10^{11} \text{ Nm}^{-2}$, $1.051 \times 10^{11} \text{ Nm}^{-2}$ and $2.109 \times 10^{11} \text{ Nm}^{-2}$, respectively [45]. If these elasticity constants are replaced in Eq. (9), the stress can be written as follows:

$$\sigma_c = -2.268 \times 10^{11} \epsilon_z$$

 σ_c values of the produced ZnO films were found using the calculated the ϵ_z values. The calculated σ_c values are given in Table 2. As seen from this table, the stress on the all films has the negative sign. This implies that c lattice constant of the produced ZnO films elongated along the growth direction [46].

The peak broadening in XRD patterns was used to evaluate average crystallite size (D_{ave}) by the use of the Scherrer formula [47],

$$D_{ave} = \frac{0.94 \,\lambda}{\gamma \cos \theta}$$

where λ , γ and θ are the X-ray wavelength (CuK_{α} =1.54059 nm), the full width half maximum (FWHM) peak breadth and the Bragg angle, respectively. Crystalline defects (dislocation) depend on the morphological properties of the material surface and the formation and size of the particles; directly affect the strength, hardness and ductility of the material [48]. To determine the amount of these defects, dislocation density given below is used the expression [49]:

$$\delta = \frac{1}{D_{ave}}$$

The calculated D_{ave} and δ values of the produced ZnO films are tabulated in Table 1.

Morphological Parameters

Surface morphology and degree of roughness is an important factor affecting film properties and their use in technological applications. The surface topography of the produced ZnO films were determined by AFM. Fig. 5 shows the 3x3 µm² three-dimensional AFM micrographs of the produced ZnO films. The surfaces of the produced ZnO films are almost homogeneous, continues and dense. There are not the defect or void formation on the surface of these films. In some regions on the surfaces, it can be seen that there are clusters with having different heights and sizes. The light colours on the surface indicate the places where atoms prefer to grow, whereas the dark ones indicate where there are fewer film formations. The clusters formed on the surface of the film during te layer-

by-layer film formation may be due to the surface properties of the used substrates.

The calculation of surface roughness is one of the most used parameter for investigation of changes in surface topography. In order to characterize the surface roughness, peak-valley (R_{pv}), root-mean-square (R_q) and average surface roughness (R_a) values were calculated for each ZnO film using the XEI version 1.7.1 software. The roughness values of the produced ZnO films are given in Table 3. It is clearly seen that change of sonication duration affected the roughness values. These values have the same trend with each other. When the sonication duration increased, the surface roughness of the produced ZnO films first increased. As the sonication duration increased further, the surface roughness of these films decreased.

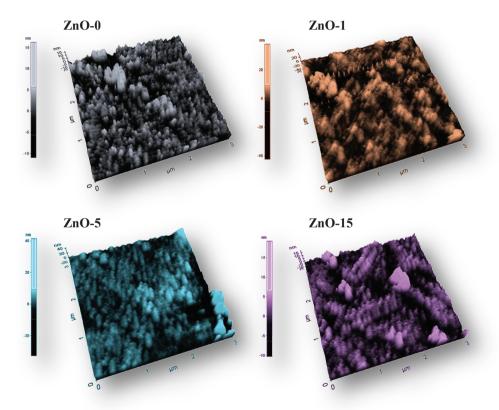


Figure 5. Three dimensional AFM micrographs and histograms of the produced ZnO films.

Table 3. Roughness parameters, Skewness and Kurtosis values of the produced ZnO films.

Parameter	ZnO-0	ZnO-1	ZnO-5	ZnO-15
R_{pv} (nm)	27.27	80.31	74.11	29.52
R_q (nm)	2.89	4.72	4.89	3.31
R_a (nm)	2.19	3.59	3.20	2.46
R_{sk}	-0.451	-0.312	-0.907	-0.888
R_{ku}	4.448	5.233	14.066	5.880

Fig. 6 presents height distribution histograms of the produced ZnO films. The statistical analysis of AFM data for the produced ZnO films was performed using the height distributions. The height asymmetry can be defined by statistical parameters of surface Skewness (R_{sk}) and Kurtosis (R_{ku}) [50]. The R_{sk} parameter represents a measure of the asymmetry of the distribution of heights [51]. The R_{sk} values of all the films are the negative, and given in Table 3. Negative R_{sk} means that the surface is planar and the surface has more valleys than peaks [52]. So, the negative sign of R_{sk} for all the produced films indicates that the surface morphology is a rough surface with planar regions and a high density of valleys. The R_{ku} parameter is useful in characterization of surfaces because of being a measure of the sharpness of the height distribution. In case of high R_{ku} value (>3, leptokurtic), the surface has with a spiky surface. That is, relatively the surface has a lot numbers both many high peaks and low valleys [53]. The R_{ku} values of all the produced ZnO films have the higher than 3 (Table 3). The fact that the height distributions have stronger and sharper peaks means that the surface of the film consists of several very deep valleys or very high summits. It was reported that metal surfaces with negative Skewness and higher Kurtosis values trend to decrease friction [54, 55].

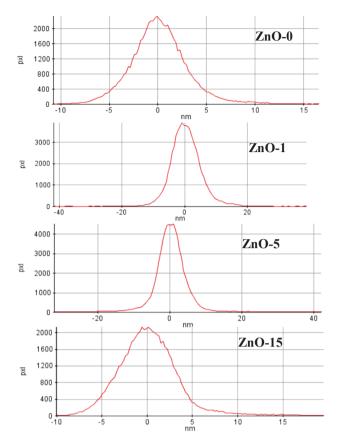


Figure 6. Height distribution histograms for the produced ZnO films.

Optical Parameters

Fig. 7 show the transmittance (T) and diffuse reflectance (R) spectra of the produced ZnO films. The T_{ave} values in the visible region of these films are higher than 85%, the R_{ave} values of them are lower than 14%. The fundamental absorption edge of the films is related to electron transitions between valence band and conduction band. To calculate the optical band gap of the films, this edge can be used. Tauc plot for the direct transition is expressed by [56]

$$(\alpha h v) = A' \big(h v - E_g \big)^{1/2}$$

where α , hv, A' and E_g denote absorption coefficient, photon energy, a constant and direct band gap, respectively. The E_g values of the produced ZnO films were calculated from Tauc plots of $(\alpha hv)^2$ versus hv (Fig. 8) and are given in Table 4. As seen, the values of E_g first decreased with sonication duration and then increased.

It can be an exponential part called the Urbach tail that extends inward from the edge of the optical band. This exponential tail is associate with the disordered materials because such materials have localized states, which elongated in the optical band gap. Urbach empirical relation with $h\nu$ photon energy and α absorption coefficient is given the following [57].

$$\alpha = \alpha_0 \exp\left(\frac{hv}{E_U}\right)$$

where α_0 is a constant. E_U is called Urbach energy and corresponds the energy of the band tail. If the logarithm of both sides of the Eq. (14) is taken, a straight line equation can be obtained as follows:

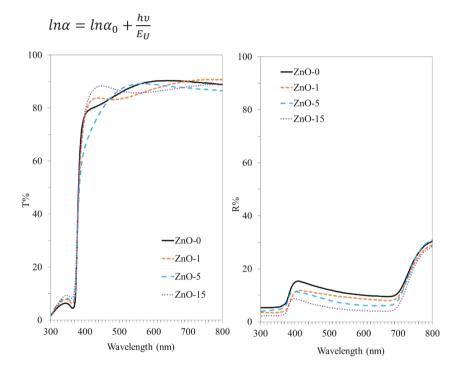


Figure 7. The transmittance and diffuse reflectance spectra of the produced ZnO films.

Table 4. Optical and dispersion parameters of the produced ZnO films.

Parameter	ZnO-0	ZnO-1	ZnO-5	ZnO-15
E_g (eV)	3.282	3.275	3.273	3.275
E_U (meV)	91	104	149	114
σ	0.27	0.24	0.17	0.22
E_d (eV)	14.531	11.148	10.244	_
E_o (eV)	5.214	5.185	5.165	_
M_{-1}	2.787	2.150	1.983	_
$M_{-3} ({\rm eV})^{-2}$	0.103	0.080	0.074	_
E_0/E_g	1.59	1.58	1.58	_

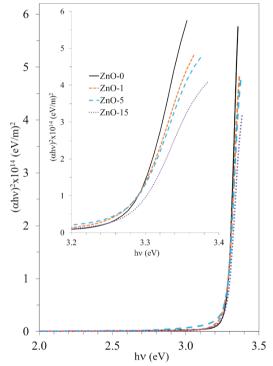


Figure 8. Tauc plots of the produced ZnO films.

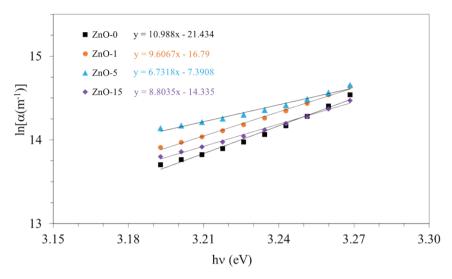


Figure 9. $\ln(\alpha)$ vs. $h\nu$ plots of the produced ZnO films.

The plots of $\ln \alpha$ versus $\hbar \nu$ for the films are given in Fig. 9. The E_U values were determined from the inverse of the slope of these plots, and tabulated in Table 4. The increase of E_U values for the produced ZnO films suggests that the atomic structural disorder increases with sonication duration. The E_U and E_g values change inversely. The absorption edge broaden resulting of the electron-phonon or excitonphonon interactions. This broadening is characterized by the steepness parameter (σ), and calculated by following relation [58]

$$\sigma = \frac{kT}{E_U}$$

The σ values of the produced ZnO films were calculated using Eq. (16) (for T=300 K). The calculated values are given in Table 4. The values of σ indicate that the absorption edge varies with sonication duration, and they are in same trend with E_g values.

The refractive index (n) is a considerable parameter for optical materials and their in potential applications, and can be determined using following equation [59].

$$n = \left[\frac{1+R}{1-R}\right] + \sqrt{\frac{4R}{(1-R)^2} - k^2}$$

where k is extinction coefficient, and expressed by

$$k = \frac{\alpha\lambda}{4\pi}$$

Fig. 10 shows a plot of the calculated n and k values as a function of wavelength. As seen, the values of n decrease with increasing sonication duration, and is almost constant in the range of 400-700 nm.

The real (ε_l) and imaginary (ε_2) parts of the dielectric constant depend on the values of n and k. The values of ε_l and ε_2 are calculated using the following equations [60]:

$$\varepsilon_1 = n^2 - k^2$$

$$\varepsilon_2 = 2nk$$

For the produced ZnO films, variation of the dielectric constants as a function of wavelength are given in Fig. 11. The values of ε_l are higher than the values of ε_2 .

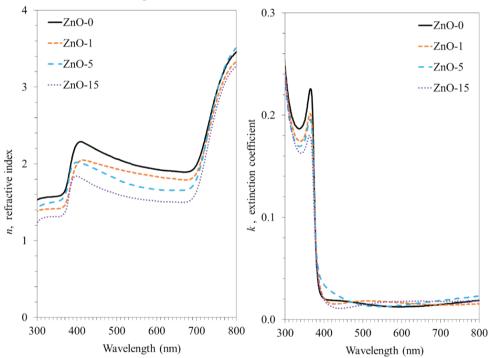


Figure 10. The variations of *n* and *k* values of the produced ZnO films as a function of wavelength.

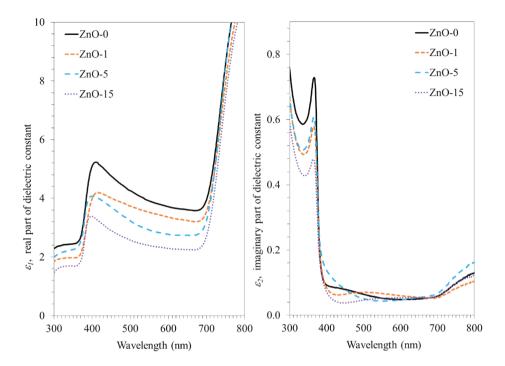


Figure 11. The variations of ε_l and ε_2 values of the produced ZnO films as a function of wavelength.

The real (σ_l) and imaginary (σ_2) parts of the optical conductivity related to the dielectric constants. The σ_l and σ_2 are calculated using the following equations [60]:

$$\sigma_1 = 2\pi c \varepsilon_0 \frac{\varepsilon_2}{\lambda}$$

$$\sigma_2 = 2\pi c \varepsilon_0 \frac{\varepsilon_1}{\lambda}$$

where c and ε_0 are the speed of light and the free space dielectric constant, respectively. For the produced ZnO films, optical conductivity spectra are given in Fig. 12. The values of σ_2 are higher than the values of σ_1 .

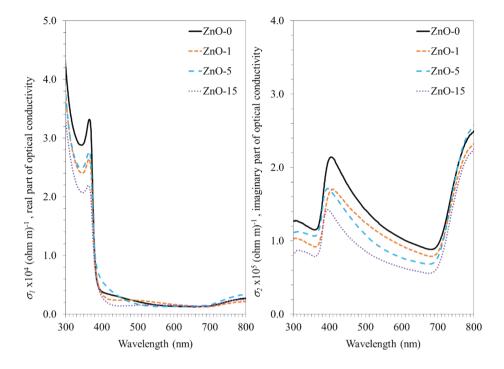


Figure 12. The variations of σ_1 and σ_2 values of the produced ZnO films as a function of wavelength.

Dispersion Parameters

The dispersion acts a significant role in the investigation of optical materials due to its importance in designing optical devices for spectral dispersion. The dispersion parameters (E_0 and E_d) are determined in view of Wemple–DiDomenico model according to the equation [61]

$$n^2 = 1 + \frac{E_0 E_d}{E_0^2 - (h\nu)^2}$$

where E_0 is oscillation energy and E_d is dispersion energy. The graphs $(n^2-1)^{-1}$ against the $(h\nu)^2$ of the produced ZnO films were drawn, and dispersion parameters were extracted from slopes and intercepts of these plots using Eq. (23). The calculated dispersion parameters are given in Table 4. The produced ZnO films except for the ZnO-15 film obeyed the single-oscillator model. The calculated E_o and E_g values are agreement with the reported values for ZnO [62, 63].

CONCLUSIONS

In present study, ZnO films were successfully produced by lowcost spin-coating technique using un-sonicated/sonicated solutions. XRD patterns, AFM images, transmittance and diffuse reflectance spectra of these films were analyzed, and determined some physical parameters. The XRD results showed that the produced ZnO films are of polycrystalline nature with hexagonal wurtzite structure. They exhibit a Zincite-type structure with the c-axis preferred orientation. The structural parameters were calculated using XRD data of the produced ZnO films. To determine of the surface morphology and degree of roughness of the films, AFM images were taken. The roughness parameters, Skewness and Kurtosis were determined. AFM measurement results showed increased roughness of the produced ZnO film because of the increased grain size. The optical band gap, Urbach energies and optical constants of the ZnO films were calculated using the transmittance and diffuse reflectance spectra. The values of optical band gap decreased and then increased with increasing sonication duration. The ZnO-5 film has a minimum optical band gap value. The dispersion parameters were calculated by Wemple-DiDomenico model. It was observed that the produced ZnO films (except ZnO-15) obeyed this model.

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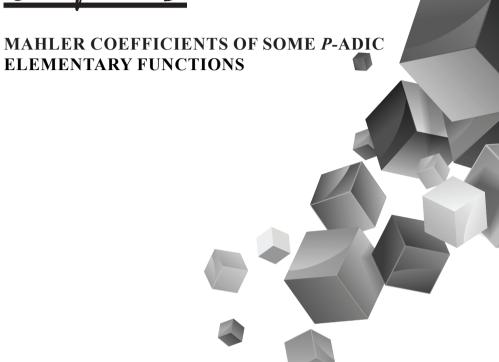
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1. Introduction

In scientific studies and in daily life we generally use the rational numbers field \mathbb{Q} . The rational numbers field \mathbb{Q} is complete field only according to the trivial norm, and by completing the rational numbers field \mathbb{Q} according to the usual absolute value, we obtain the real numbers field \mathbb{R} . According the Ostrowski Theorem, each norm on \mathbb{Q} is either equivalent to the usual absolute value norm or to the p-adic norm, where p is a prime number [1]. Now, let us briefly introduce the p-adic norm. Let p be an arbitrary prime number. For any rational number different from zero $x \in \mathbb{Q}$, $x \neq 0$, there exist $a, b, \alpha \in \square$

$$x = p^{\alpha} \cdot \frac{a}{b}$$
, $p \nmid a.b$

Here, $\alpha \in \mathbb{Z}$ is uniquely determined by prime p. Hence, the function $|\cdot|_p:\Box \to [0,\infty)$ is defined by formula

$$|x|_p = p^{-\alpha}$$

If x = 0, then we take $|0|_p = 0$. In fact, the function $|\cdot|_p$ is a norm and it is called *p-adic norm*. It holds the *ultra-metric triangle inequality* which is more strongly than triangle inequality

$$|x + y|_p \le \max\{|x|_p, |y|_p\}$$

A norm that satisfies the ultrametric triangle inequality is called a *non Archimedean norm*, and the analysis performed with such this norm is called *non Archimedean Analysis*.

For a given prime p, the p-adic numbers field \mathbb{Q}_p is the completion of the rational numbers with_respect to p-adic norm . The p-adic numbers field was first described by Kurt Hensel in 1897, in a certain period of time; these numbers were used only in one area of number theory [2]. It

began to question why only real numbers were used in physics, and in 1968, two theoretical mathematicians had used *p*-adic numbers in physics. In 1984 two theoretical physicists had applied the *p*-adic numbers successfully to string theory, then it has begun to be used *p*-adic numbers in applied fields [3]. Moreover, fields such as *p*-adic quantum theory, the *p*-adic universe model, *p*-adic string theory have emerged in physics.

Studies made using p-adic numbers in mathematics were collected in a field called p-adic analysis. Concepts made with real numbers have begun to be redefined and interpreted by using p-adic numbers [3-4].

By \mathbb{C}_p we denote the completion of the algebraic closure of padic numbers field \mathbb{Q}_p . Let $\mathbb{Z}_p = \left\{ x \in \mathbb{C}_p \, \big| \, x \, \big|_p \leq 1 \right\}$. It is well known
from the Kaplansky's theorem that each function f from $C(\mathbb{Z}_p \to \mathbb{C}_p)$ can
be uniformly approximated by polynomials [5]. For any $x \in \mathbb{Z}_p$ the
binomial coefficient polynomial $\binom{x}{n}$ is defined by $\binom{x}{0} = 1$ and

$$\binom{x}{k} = \frac{x.(x-1)...(x-k+1)}{k!} (k > 0).$$

Hence, $\binom{x}{k}$ is a polynomial of degree k.

The set of polynomial functions $\{x \to \binom{x}{k} : x \in \mathbb{Z}_p, k \in \mathbb{N}\}$ constitutes an orthonormal basis for the space $C(\mathbb{Z}_p \to \mathbb{C}_p)$ according to the supremum norm. In the other words, for each function $f \in C(\mathbb{Z}_p \to \mathbb{C}_p)$, there exist unique numbers $a_0, a_1, a_2, \ldots \in \mathbb{C}_p$ \ni

$$f(y) = \sum_{k=0}^{\infty} a_n {y \choose k}$$

The set $\{\binom{*}{n}: k \in \mathbb{N}\}$ is called the *Mahler* basis of the space $C(\mathbb{Z}_p \to \mathbb{C}_p)$, and the element of $\{a_k: k \in \mathbb{N}\}$ in

$$f(y) = \sum a_k \binom{y}{k}$$

are called the *Mahler coefficients* of $f \in C(\mathbb{Z}_p \to \mathbb{C}_p)$ (see [6]).

In the present study, we investigate the Mahler expansions and Mahler coefficients of some p-adic trigonometric functions. Also, we give their Volkenborn integral related some special numbers.

The Volkenborn integral of any function f from $C^1(\mathbb{Z}_p \to \mathbb{C}_p)$ is defined by the formula

$$\int_{\mathbb{Z}_p} f(x)dx = \lim_{k \to \infty} \frac{1}{p^k} \sum_{j=0}^{p^k - 1} f(j)$$

If f(x) = c, then

$$\int_{\mathbb{Z}_n} f(x) dx = c$$

Also, the property holds

$$\int_{\mathbb{Z}_p} f(x+1)dx - \int_{\mathbb{Z}_p} f(x)dx = f'(0)$$

In addition, if the function f in the form of

$$f(x) = \sum_{n=0}^{\infty} a_n {x \choose n} \in C^1(\mathbb{Z}_p \to \mathbb{C}_p)$$

then, its Volkenborn integral is

$$\int_{\mathbb{Z}_p} f(x)dx = \sum_{n=0}^{\infty} a_n \frac{(-1)^n}{n+1}$$

(For details see [5-7]). We note that the Volkenborn integral was first defined by A. Volkenborn in his twin papers [8], [9]. By the agency of *p*-adic Volkenborn integrals and their extensions, very important studies have done on some special numbers, polynomials and functions such as Bernoulli, Euler, Changhee, Daehee, Bernstein and their extensions. ([10]-[24]). In fact, almost all special numbers and polynomial mentioned above are described by *p*-adic Volkenborn integrals and their extensions.

We note that an infinite series $\sum_{n=0}^{\infty} a_n$ with $a_n \in \mathbb{C}_p$ convergences if and only if $|a_n|_p \to 0$. In p-adic content the elementary functions as logarithmic, exponential and trigonometric functions are described by their usual power series but, their regions of convergence are absolutely different from the complex cases. Here, we recall only few ones of them.

The power series

$$\sum_{n=1}^{\infty} \frac{(-1)^{n+1} \cdot x^n}{n}$$

converges for $|x|_p < 1$, and it is denoted by $\log_p(1+x)$. So, the *p*-adic logarithmic function is defined by

$$\log_p x = \sum_{n=1}^{\infty} \frac{(-1)^{n+1} \cdot (x-1)^n}{n},$$

and it converges for $|x-1|_p < 1$.

The exponential function in *p*-adic content is defined by formula

$$\exp_p x = \sum_{n=0}^{\infty} \frac{x^n}{n!}$$

and its convergence domain is the set $E = \{x \in \mathbb{C}_p : |x|_p < p^{\frac{1}{1-p}}\}.$

In *p*-adic content all trigonometric functions are defined by their power series in real case. Here, we give few examples. The *p*-adic sine and cosine functions are defined by the usual power series, respectively,

$$\sin_p x = \sum_{n=0}^{\infty} \frac{(-1)^n x^{2n+1}}{(2n+1)!},$$

$$\cos_p x = \sum_{n=0}^{\infty} \frac{(-1)^n x^{2n}}{(2n)!},$$

and they converge for all $x \in E$. The *p*-adic sine and cosine hyperbolic functions are defined by the power series, respectively,

$$\sinh_p x = \sum_{n=0}^{\infty} \frac{(-1)^n x^{2n+1}}{(2n+1)!},$$

$$\cosh_p x = \sum_{n=0}^{\infty} \frac{(-1)^n x^{2n}}{(2n)!},$$

and they converge for all $x \in E$. The *p*-adic tangent and cotangent are described similarly and their convergence domain is \mathbb{Z}_p .

For any $x \in \mathbb{Z}_p$ ve $m \in \mathbb{N} \cup \{0\}$, let us consider the equality

$$x^m = \sum_{n=0}^{\infty} a_{n,m} \binom{x}{n}$$
.

The numbers $a_{n,m}$ which are determined by the equality above are called the *Stirling numbers of second type*.

Lemma 1.1([5]) The Stirling numbers of second type $a_{n,m}$ hold the relations for $m \in \{0,1,2,...\}$:

i)
$$a_{0,0} = 1$$
; $a_{0,m} = a_{0,n} = 0$ $(m,n) \neq 0$; $a_{n,m} = 0$ $(n > m)$,

ii)
$$a_{n,m} = \sum_{j=0}^{n} (-1)^{n-j} \binom{n}{j} j^m$$
,

iii) $a_{n,m}$ is an integer and divided by n!

iv) Let
$$f_{n,m}(x) = (x \frac{d}{dx})^m (x-1)^n$$
 $(x \in \mathbb{Z}_p)$. Then,

$$a_{n,m} = f_{n,m}(1)$$

$$v$$
) $a_{n,m+1} = n(a_{n,m} + a_{n-1,m}) \quad (n \ge 1)$

$$vi)$$
 $a_{n+1,m} = \sum_{j=0}^{m-1} {m \choose j} a_{n,j} \quad (m \ge 1)$

Lemma 1.2 [[5]) Let p is a prime, p > 2, $a \in p\mathbb{Z}_p$ and

$$\sin_p(ax) = \sum_{n=0}^{\infty} a_n \binom{x}{n},$$

$$\cos_p(ax) = \sum_{n=0}^{\infty} b_n \binom{x}{n} ,$$

Then, for all *n* the relations hold:

$$a_{2n} = (-1)^n 2^{2n} (\sin_p \frac{1}{2} a)^{2n} \sin_p (na),$$

$$a_{2n+1} = (-1)^n 2^{2n+1} (\sin_p \frac{1}{2} a)^{2n+1} \cos_p (n + \frac{1}{2}) a,$$

$$b_{2n} = (-1)^n 2^{2n} (\sin_p \frac{1}{2} a)^{2n} \cos_p (na),$$

$$b_{2n+1} = (-1)^n 2^{2n+1} (\sin_p \frac{1}{2} a)^{2n+1} \sin_p (n + \frac{1}{2}) a.$$

2. Main Results

In the present study we study some *p*-adic elementary functions. We investigate the Mahler expansions and Mahler coefficients of some p-adic elementary functions. Also, we compute their Volkenborn integrals related the Stirling numbers of second type.

For the Mahler expansion of p –adic hyperbolic sine function, the next theorem holds:

Theorem 2.1: Let p is a prime, p > 2, $a \in p\mathbb{Z}_p$ and

$$sinh_p(ax) = \sum_{n=0}^{\infty} a_n {x \choose n} \quad (x \in \mathbb{Z}_p).$$

Then, the coefficients a_n satisfy the formulas for all n

$$a_{2n} = \frac{1}{2} \frac{(\exp_p a - 1)^{2n} \cdot [(\exp_p a)^{2n} - 1]}{(\exp_p a)^{2n}}$$

$$a_{2n+1} = \frac{1}{2} \frac{(\exp_p a - 1)^{2n+1} \cdot [(\exp_p a)^{2n+1} + 1]}{(\exp_p a)^{2n+1}}$$

Proof: It is well known that

$$\sinh_p(ax) = \frac{\exp_p(ax) - \exp_p(-ax)}{2}.$$

Hence, we write

$$\sinh_p(ax) = \frac{1}{2} \exp_p(ax) - \frac{1}{2} \exp_p(-ax)$$

$$= \frac{1}{2} \sum_{n=0}^{\infty} (\exp_p(a) - 1)^n {x \choose n}$$
$$-\frac{1}{2} \sum_{n=0}^{\infty} (\exp_p(-a) - 1)^n {x \choose n}$$

$$= \sum_{n=0}^{\infty} \frac{1}{2} \left[(\exp_p(a) - 1)^n - (\exp_p(-a) - 1)^n \right] {x \choose n}$$

Now, we can determine the coefficients a_n as

$$a_n = \frac{1}{2} (\exp_p(a) - 1)^n - \frac{1}{2} (\exp_p(-a) - 1)^n$$

$$= \frac{1}{2} [(\exp_p(a) - 1)^n - \left(\frac{1}{\exp_p(a)} - 1\right)^n]$$

$$= \frac{1}{2} [(\exp_p(a) - 1)^n - \frac{(1 - \exp_p(a))^n}{(\exp_p(a))^n}$$

$$a_n = \frac{1}{2} \frac{(\exp_p(a) - 1)^n \cdot (\exp_p(a))^n - (1 - \exp_p(a))^n}{(\exp_p(a))^n}$$

So, we obtain the relations

$$a_{2n} = \frac{1}{2} \frac{(\exp_p(a) - 1)^{2n} \cdot [(\exp_p(a))^{2n} - 1]}{(\exp_p(a))^{2n}},$$

and

$$a_{2n+1} = \frac{1}{2} \frac{(\exp_p(a) - 1)^{2n+1}.[(\exp_p(a))^{2n+1} + 1]}{(\exp_p(a))^{2n+1}}.$$

Teorem 2.2: The Mahler coefficients a_m of $\sinh_p(ax)$ are

$$a_m = \frac{a^{2m+1}}{(2m+1)!} \sum_{n=2m+1}^{\infty} a_{n,2m+1}.$$

Here, $a_{m,n}$ are the Stirling numbers of second type.

Proof: Let $a \in E$. Assume that

$$\sinh_p(ax) = \sum_{n=0}^{\infty} a_n \binom{x}{n}.$$

Let us determine the coefficients a_m .

By the definition of $sinh_p(ax)$ we write

$$\sinh_p(ax) = \sum_{m=0}^{\infty} \frac{a^{2m+1}}{(2m+1)!} x^{2m+1}$$

By the definition of Striling number of second type we know that

$$x^m = \sum_{n=0}^{\infty} a_{n,m} \binom{x}{n}$$

If j < j then $a_{j,i} = 0$, and so, we obtain that

$$\sinh_p(ax) = \sum_{m=0}^{\infty} \left(\sum_{n=2m+1}^{\infty} \frac{a^{2m+1}}{(2m+1)!} a_{n,2m+1} \right) {x \choose n}$$

Hence, we obtain the Mahler coefficients a_n of $\sinh_p(ax)$ as

$$a_m = \frac{a^{2m+1}}{(2m+1)!} \sum_{n=2m+1}^{\infty} a_{n,2m+1}.$$

Now we give the Mahler expansions and Mahler coefficients of *p*-adic cosine hyperbolic function.

Theorem 2.3: Let p be a prime number, p>2, $a\in p\mathbb{Z}_p$, $x\in \mathbb{Z}_p$ and

$$\cosh_p(ax) = \sum_{n=0}^{\infty} c_n \binom{x}{n}.$$

Then, the relations hold:

$$c_{2n} = \frac{1}{2} \frac{(e \exp_p a - 1)^{2n} [\left(\exp_p a\right)^{2n} + 1]}{(\exp_p a)^{2n}},$$

$$c_{2n+1} = \frac{1}{2} \frac{(\exp_p a - 1)^{2n+1} [(\exp_p a)^{2n+1} - 1]}{(\exp_p a)^{2n+1}}.$$

Proof: We know that

$$\cosh_p(ax) = \frac{\exp_p(ax) + \exp_p(-ax)}{2}.$$

By the Mahler expansion of exponential function we write

$$\cosh_p(ax) = \frac{1}{2} \exp_p(ax)(ax) + \frac{1}{2} \exp_p(ax)(-ax).$$

Using the equalities

$$\exp_p(ax) = \sum_{n=0}^{\infty} (\exp_p(a) - 1)^n {x \choose n},$$

$$\exp_p(-ax) = \sum_{n=0}^{\infty} (\exp_p(-a) - 1)^n {n \choose n}$$

we obtain that

$$\cosh_p(ax) = \sum_{n=0}^{\infty} \frac{1}{2} \left[(\exp_p(a) - 1)^n + (\exp_p(-a) - 1)^n \right] {x \choose n}$$

Let us write the Mahler coefficients c_n in more obvious form. It is clear that

$$c_n = \frac{1}{2} [(\exp_p(a) - 1)^n + (\exp_p(-a) - 1)^n]$$

$$= \frac{1}{2} \left[(\exp_p(a) - 1)^n + \left(\frac{1}{\exp_p a} - 1 \right)^n \right]$$

$$= \frac{1}{2} \left[(\exp_p(a) - 1)^n + \left(\frac{1 - \exp_p a}{\exp_p a} \right)^n \right]$$

or

$$c_n = \frac{1}{2} \frac{\left(\exp_p a - 1\right)^n \left(\left(\exp_p a\right)^n + \left(1 - \exp_p a\right)^n\right)}{\left(\exp_p a\right)^n}.$$

So, we have the desire equalities:

$$c_{2n} = \frac{1}{2} \frac{(\exp_p a - 1)^{2n} [(\exp_p a)^{2n} + 1]}{(\exp_p a)^{2n}},$$

$$c_{2n+1} = \frac{1}{2} \frac{(\exp_p a - 1)^{2n+1} [(\exp_p a)^{2n+1} - 1]}{(\exp_p a)^{2n+1}}.$$

The Mahler coefficients of $\cosh_p(ax)$ in term of Stirling number of second type are given below:

Theorem 2.4: The coefficients c_m of $cosh_p(ax)$ have the form of

$$c_m = \frac{a^{2m}}{(2m)!} \sum_{n=2m}^{\infty} a_{n,2m}$$

where $a_{m,n}$ are the Stirling numbers of second type.

Proof: Let $a \in E$ and

$$\cosh_p(ax) = \sum_{n=0}^{\infty} c_n \binom{x}{n}.$$

By the definition of $\cosh(ax)$ we write

$$\cosh_p(ax) = \sum_{n=0}^{\infty} \frac{a^{2n}}{(2n)!} x^{2n}.$$

If i < j then $a_{i,i} = 0$. So we have

$$\cosh_p(ax) = \sum_{m=0}^{\infty} \left(\sum_{n=2m}^{\infty} \frac{a^{2n}}{(2n)!} a_{m,n} \right) {x \choose m}$$

Hence, we obtain the Mahler coefficients c_n in the form of

$$c_n = \frac{a^{2n}}{(2n)!} \sum_{n=2m}^{\infty} a_{m,n}$$

Theorem 2.5. If $a \in p\mathbb{Z}_p$, $x \in \mathbb{Z}_p$, then

$$\int_{\mathbb{Z}_p} \sin_p(ax) \, dx = \sum_{n=1}^{\infty} a_n \frac{(-1)^n}{n+1}.$$

Here, a_n are the Mahler coefficients of $\sin_p(ax)$ which are given in Lemma 1.2.

Proof. Let $a \in p\mathbb{Z}_p$, $x \in \mathbb{Z}_p$. By Lemma 1.2 we write

$$\sin_p(ax) = \sum_{n=0}^{\infty} a_n \binom{x}{n}$$

Taking Volkenborn integral we write

$$\int_{\mathbb{Z}_p} \sin_p(ax) dx = \int_{\mathbb{Z}_p} \left(\sum_{n=0}^{\infty} a_n {x \choose n} \right) dx$$

or

$$\int_{\mathbb{Z}_p} \sin_p(ax) \, dx = \sum_{n=0}^{\infty} a_n \int_{\mathbb{Z}_p} {x \choose n} \, dx$$

By the property

$$\int_{\mathbb{Z}_p} \binom{x}{n} dx = \frac{(-1)^n}{n+1}$$

We obtain

$$\int_{\mathbb{Z}_n} \sin_p(ax) \, dx = \sum_{n=1}^{\infty} a_n \frac{(-1)^n}{n+1}$$

Corollary 2.1: If $a \in p\mathbb{Z}_p$, $x \in \mathbb{Z}_p$ and the numbers a_n are the Mahler coefficients of $\sin_p(ax)$, then the equality holds

$$\sum_{n=1}^{\infty} a_n \frac{(-1)^n}{n+1} = -\frac{a}{2}$$

Proof: It is trivial from Lemma 1.2 and Theorem 2.5.

Theorem 2.6. If $a \in p\mathbb{Z}_p$, $x \in \mathbb{Z}_p$, then

$$\int_{\mathbb{Z}_p} \cos_p(ax) \, dx = \sum_{n=1}^{\infty} b_n \frac{(-1)^n}{n+1}$$

Here, b_n are the Mahler coefficients of $\cos_p(ax)$ which are given in Lemma 1.2.

Proof: Let

$$\cos_p(ax) = \sum_{n=1}^{\infty} b_n \binom{x}{n}$$

Taking Volkonborn integral we have

$$\int_{\mathbb{Z}_p} \cos_p(ax) \, dx = \int_{\mathbb{Z}_p} \left(\sum_{n=0}^{\infty} b_n \binom{x}{n} \right) dx$$

or

$$\int_{\mathbb{Z}_p} \cos_p(ax) \, dx = \sum_{n=0}^{\infty} b_n \int_{\mathbb{Z}_p} {x \choose n} \, dx$$

By the property

$$\int_{\mathbb{Z}_p} {x \choose n} dx = \frac{(-1)^n}{n+1}$$

We obtain that

$$\int_{\mathbb{Z}_p} \cos_p(ax) \, dx = \sum_{n=1}^{\infty} b_n \frac{(-1)^n}{n+1}$$

Corollary 2.2: If $a \in p\mathbb{Z}_p$, $x \in \mathbb{Z}_p$ and the numbers b_n are the Mahler coefficients of $\cos_p(ax)$, then the equality holds

$$\sum_{n=1}^{\infty} b_n \frac{(-1)^n}{n+1} = -\frac{a}{2}$$

Theorem 2.7: If $a \in E$, $x \in \mathbb{Z}_p$.then

$$\int_{\mathbb{Z}_p} \sinh_p(ax) \, dx = \sum_{n=1}^{\infty} a_n \frac{(-1)^n}{n+1}$$

where (a_n) are the Mahler coefficients of $\sinh_p(ax)$ which are given in Theorem 2.1.

Proof: By Theorem 2.1, we write

$$\sinh_p(ax) = \sum_{n=1}^{\infty} a_n \binom{x}{n}$$

Hence, we have

$$\int_{\mathbb{Z}_p} \sinh_p(ax) \, dx = \int_{\mathbb{Z}_p} \left(\sum_{n=1}^{\infty} a_n \frac{(-1)^n}{n+1} \right) dx$$

or

$$\int_{\mathbb{Z}_p} \sinh_p(ax) \, dx = \sum_{n=1}^{\infty} a_n \int_{\mathbb{Z}_p} {x \choose n} \, dx$$

By the relation

$$\int_{\mathbb{Z}_p} \binom{x}{n} dx = \frac{(-1)^n}{n+1},$$

we obtain that

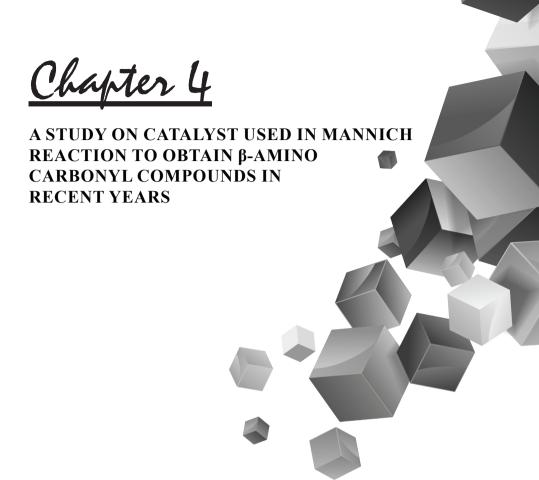
$$\int_{\mathbb{Z}_p} \sinh_p(ax) \, dx = \sum_{n=1}^{\infty} a_n \frac{(-1)^n}{n+1}.$$

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The Mannich reaction, an effective method used in reactions such as the synthesis of various biologically important β -amino carbonyl compounds, as well as the preparation of multi-functional synthetic building blocks containing excess nitrogen atoms, is constantly evolving. Various catalysts are used in the synthesis of β -amino carbonyl compounds in the Mannich reaction. Catalysis is considerable in diverse implementation of organic chemistry.

Sabahat et al. reported that they synthesized amino carbonyl compounds with high yields by using camphorsulfonate-based PILs in a shorter reaction time at ambient temperature directly in a one- pot Mannich condensation (Sabahat et al., 2020).

Figure 1. Synthesis of ionic liquids

L. Satish et al. demonstrated a simple chemical synthesis route for the synthesis of phosphate functional graphene oxide modified with copper oxide nanoparticles (CuO/PGO). The obtained nanocatalyst was used as an potent and active applicant for the synthesis of β -amino carbonyl compounds in a green synthetic ultrasonic way (L. Satish et al., 2020).

Figure 2. A model of Mannich reaction by CuO/PGO

Bhupinder et al. report that manganese perchlorate hexahydrate, a dry, odorless, pink crystalline solid, is an efficient and promising catalyst for the synthesis of organic compounds. They reported an ultrasound-assisted one-pot approach to the Mannich reaction of aromatic aldehydes, amines and ketones in ethanol catalyzed by manganese perchlorate hexahydrate at room temperature (Bhupinder et al., 2020).

Figure 3. Three-component Mannich reactions by Mn(ClO4)2.6H2O

Hasniye et al. reported a soft, useful, efficient, one-pot Mannich reaction for obtaining β -amino carbonyl compounds from ketones, aromatic aldehydes and aromatic amines using molecular iodine as a catalyst at room temperature (Hasniye et al., 2019).

Figure 4. Iodine-catalyzed Mannich reaction

Hasniye and Kübra reported a rapid synthesis of β -aminocarbonyl compounds in the presence of Bi (NO3)3.5H2O (BN) as it is non-toxic, stable in air, and cheaper (Hasniye and Kübra 2019, Jaspreet Rajput, 2013).

Figure 5. Mannich reaction by Bi(NO3)3

Ali et al. presented a new catalytic protocol for the efficient synthesis of β -amino ketone derivatives with a firmly bound styrene-divinylbenzene copolymer/TiCl4 complex as a reusable polymeric Lewis acid catalyst in ethanol medium. They reported that the catalyst not only showed high catalytic activity, but also had many advantages such as thermal stability and recyclability (Ali et al. 2019).

Figure 6. The synthesis of polymeric carrier and its titanium tetrachloride complex

Ali Maleki et al. reported a appropriate and potent method for synthesis of β -amino carbonyl compounds via the condensation of aniline, aldehyde and ketone compound in the presence of PEG-SO3H-functionalized Fe3O4 as a superparamagnetic heterogeneous nanocatalyst at room temperature in high yields (Ali Maleki et al., 2018).

Figure 7. β-amino carbonyl compounds synthesis byFe3O4@PEG-SO3H catalyst

Farhad et al. reported that they prepared a easy, potent and environmentally friendly solid acid catalyst by fixing a propyl sulfonic acid to the surface of silica-coated magnetic nanoparticles with low-cost precursors (Farhad et al., 2017).

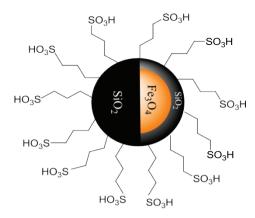


Figure 8. Sulfonated core-shell magnetic nanoparticles (SMNPs)

Manmeet & Satya report an efficient protocol for the synthesis of coumarins via the Pechmann reaction and β-amino carbonyl compounds via the aza-Michael reaction using a catalytic amount of solid Lewis acid catalyst, C@TiO₂-O₃-SbCl₂. They prepared six different catalysts by covalently immobilizing homogeneous Lewis acids on amorphous carbon and nano-titania derived ulfone carbon titania composite. Among the various catalysts tested, C@TiO₂-O₃-SbCl₂ reported superior catalytic activity (Manmeet & Satya 2017).

Figure 9. $C@TiO_2$ - O_3 - $SbCl_2$ catalyzed one-pot synthesis of β -amino carbonyl compounds via aza-Michael reaction

Susmita et al reported that they synthesized a new series of 1,3-disulfoimidazolium transition metal chloride systems: [Dsim]₂[ZnCl₄], [Dsim][FeCl4] and [Dsim]₂[NiCl₄] Brönsted - Lewis acidic solid material. They found that all three materials were reusable catalysts for the synthesis of one-chamber three-component Mannich-type amino carbonyl compounds (Susmita et al., 2016).

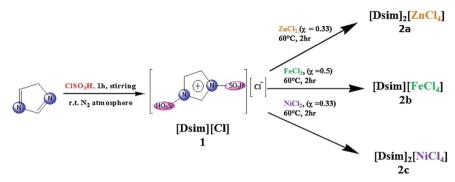


Figure 10. Synthesis of 1,3-disulfoimidazolium chlorometallate ionic liquids [Dsim]2[ZnCl4] 2a, [Dsim][FeCl₄] 2b and [Dsim],[NiCl₄] 2c

Hua et al., using the acidic IL catalyst under ultrasonic irradiation, they obtained a suitable procedure for the Mannich reaction, less reaction time and high efficiency (Hua et al., 2016).

Figure 11. Mannich reaction catalyzed by IL's under ultrasonic irradiation

Amel et al. synthesized various β -amino carbonyl derivatives of aniline, acetophenone and aromatic aldehydes at room temperature with high yields in a shorter time using PEG-600 as a safer catalyst in solvent-free conditions. This synthetic route is extremely easy and avoids the use of toxic catalysts (Amel et al., 2016).

Figure 12. *Synthesis* β -amino carbonyl derivatives

Zahra et al. developed an easy synthesis method of magnetically separable heteropoly acid (CuFe $_2$ O $_4$ -SiO $_2$ -NH $_2$ -H $_3$ PW $_{12}$ O $_{40}$) consisting of immobilized phosphotungstic acid (H $_3$ PW $_{12}$ O $_{40}$) on amine functionalized

silica coated magnetic copper-ferrite nanoparticles (CuFe₂O₄-SiO₂-NH₂). The immobilized phosphotungstic acid has been used as an efficient heterogeneous catalyst for the synthesis of β-aminocarbonyl compounds using a one-pot three-component coupling reaction of aldehydes, amines and ketones at room temperature and under solvent-free conditions (Zahra et al., 2016).

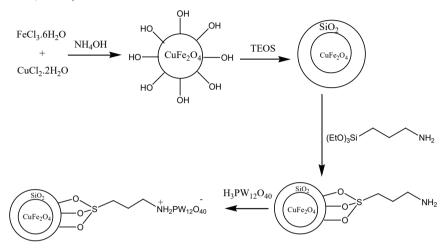


Figure 13. Synthetic pathways of (CuFe₂O₄-SiO₂-NH₂-H₃PW₁₂O₄₀)

Vishvanath et al. prepared silica supported $\text{Co(ClO}_4)_2$ catalyst and found that it is thermally stable over a long temperature range. They reported that it is an efficient, recyclable and heterogeneous catalyst for the synthesis of both 5-unsubstituted 3,4-dihydropyrimidinones and β -amino carbonyl compounds under solvent-free conditions (Vishvanath et al., 2016).

Figure 14. β -amino carbonyl compounds using silica supported $Co(ClO_{\phi})_2$ as catalyst under solvent free condition.

Parishmita et al. They prepared three efficient heterogeneous organocatalysts from three non-recyclable carboxylic acids. They found that the remarkable advantages of these catalysts were high catalytic activity in mild conditions, easy operation procedure combined with simple operation, recyclability and low cost (Parishmita et al., 2015).

Figure 15. Mannich-type reactions catalyzed by silica-supported organic acids

Di Liu et al. developed a simple and efficient method for the synthesis of β -amino carbonyl compounds using PPA/SiO₂ as catalyst, which provides short reaction time, mild reaction terms, reusable catalyst (Di Liu et al., 2015).

Figure 16. the synthesis of β -amino carbonyl compounds using PPA / SiO₂ as catalyst

Divya et al. synthesized mesoporous silica-polymer nanocomposites (SBA/PS) by wet impregnation method, functionalized with light polyphosphoric acid (PPA) for catalytic applications (Divya et al., 2015).

Figure 17. The synthesis of β -amino carbonyl compounds

Pathakota et al. Synthesized β -amino carbonyl compounds from aromatic ketones, aldehydes and amines by the Mannich reaction in the presence of amberlyst-15 as a reusable heterogeneous catalyst at room temperature under solvent-free conditions (Pathakota et al., 2015).

Figure 18. Synthesis of β -amino carbonyl compounds by Amberlyst-15

Javad Abedini et al. used PPh₃ as an efficient catalyst for the one-pot three-component Mannich reactions under solvent-free conditions at room temperature. They reported that this method provides a novel and improved method for obtaining β -amino carbonyl compounds in the lack of acidic catalyst in good yield (Javad Abedini et al.2015).

$$R_1$$
 R_2 R_1 R_2 R_3 R_4 R_4 R_5 R_6 R_7 R_8 R_8 R_9

Figure 19. Mannich reaction via PPh,

Ram Singh et al.were reported to synthesis of β -amino carbonyl compounds by FeCl₃-silica gel G catalyst and an efficient and time-saving one-pot the three-component Mannich reaction (Ram Singh et al., 2014).

Figure 20. Mannich reaction with FeCl /silica gel

Sathesh Venkatesan et al. developed a novel one-pot three-component Mannich reaction involving CDD, aniline, and benzaldehyde in the presence of NH_4OAc . They synthesized anti- β -amino-carbonyl compound as the predominant one. When the ammonium acetate was used as

catalyst were obtained a mixture of anti and syn-products. There are many advantages that mild reaction condition, easy work up procedure, and high yield of using ammonium acetate as a catalyst (Sathesh Venkatesan et al., 2014).

Figure 21. Ammonium acetate catalyzed Mannich reaction

Yi Luan et al. recorded that diazo compounds, boranes, and acyl imines undergo a three-component Mannich reaction under catalyst-free conditions to give the anti β -amino carbonyl compounds in high diastereoselectivity. They found that the obtained β -amino carbonyl compounds can be transformed to many potential bioactive molecules (Yi Luan et al., 2014).

Figure 22. Mannich reaction no catalyst

Chao Huang et al. introduced an effective and eco-friendly work for the synthesis of β -amino carbonyl compounds in this paper. They reported that the oxanorbornene b-amino esters and β -enamine esters were successfully arranged from oxabornene and amines by using solvent-free aza-Michael addition reaction in the absence of any catalyst (Chao Huang et al. 2014).

Figure 23. *Synthesis of* β *-amino carbonyl compounds*

Santosh et al. have developed an efficient method for the synthesis of β -amino carbonyl compounds using phenylboronic acid as catalyst via one-pot three-component Mannich reaction. Excellent yields, mild reaction condition, and simple experimental work-up procedure are some of the advantages of this method (Santosh et al., 2013).

Figure 24. Synthesis of Mannich Bases

Xi Cun Wang et al. developed for the synthesis of β -amino carbonyl compounds via the Mannich reaction using recyclable sulfuric acid-modified polyethylene glycol 6000 (PEG-OSO3H). they declared that the reactions were completed in short times and mild reaction conditions with good yields (Xi Cun Wang et al., 2012).

Figure 25. Mannich reaction catalyzed by PEGOSO₃H

Santosh Kamble et al. reported ultrasound promoted synthesis of β -amino carbonyl compounds in aqueous hydrotropic medium at ambient temperature. The exceptional features of the new procedure are shorter

reaction time, good yields in aqueous medium (Santosh Kamble et al., 2012).

Figure 26. *Screening of synthesis of* β *-amino carbonyl compound.*

Polyaniline/SiO₂ catalyzed one-pot mannich reaction are carried out in ethanol to obtained diverse β -amino ketones. The various wt% of polyaniline were supported on pure silica synthesized. This record has few advantages such as good yield, non-toxic, clean, simple method, easy recovery and reusability of the catalyst (Ajeet A. Yelwande et al., 2011).

Figure 27. *Synthesis of* β *-amino carbonyl compounds*

ZSM-5-SO₃H efficiently catalyzed the one-pot three-component Mannich reaction and β - aminocarbonyl compounds were obtained in fair yields and excellent stereoselectivities. Simple experimental requirements and product isolation process makes this report possible for the development of environment-friendly strategy for the synthesis of β -amino-ketones (Ahmad Reza Massah, et al., 2011).

NH₂ CHO O ZSM-5-SO₃H, rt Solvent free
$$R_1$$
 Anti R_2 R_2 R_3 R_4 R_5 R_4 R_5 R_5 R_6 R_7 R_8 R_8 R_8 R_9

Figure 28. Direct Mannich reactions by ZSM-5-SO,H

Abolghasem Davoodnia et al. reported that a simple and efficient method for the synthesis of β -amino carbonyl compounds by one-pot three-component Mannich reaction using a carbon-based solid acid (CBSA), as an effective and reusable catalyst, is described. The present procedure proposals few advantages such as simple method with an easy work-up,

shorter reaction times, and excelent yields (Abolghasem Davoodnia et al. 2011).

Figure 29. Mannich reaction catalyzed by CBSA

Hossein Eshghi et al. reported that $Fe(HSO_4)_3$ -catalyzed three-component one-pot Mannich reaction of acetophenone with different aromatic aldehydes and aromatic amines in ethanol at ambient temperature. Short reaction time, good yield, easy work-up technique, and finally reusability of the catalyst show the worth of this paper (Hossein Eshghi et al. 2011).

O CHO
$$NH_2$$

$$+ R_1 + R_2$$

$$- Fe(HSO_4)_3$$

$$- EtOH, rt$$

$$- R_1$$

Figure 30. β-aminocarbonyl compounds using ferric hydrogensulfate

Mantu Rajbangshi et al. declared that iodine-alumina was employed as a catalyst in the coupling reactions under microwave irradiation to afford β -amino carbonyl compounds in excellent yields. This environmentally friendly methodology was possessed mild reaction conditions and less reaction time (Mantu Rajbangshi et al. 2011).

Figure 31. One-pot three-component condensation

Min Wang et al. published that an efficient three-component, one-pot synthesis of β -amino carbonyl compounds were described using tin tetrachloride at room temperature in ethanol. The advantages of the process are excellent yields, simple method, and cheap catalyst (Min Wang et al. 2011).

Figure 32. *Mannich reaction by SnCl₄.5H₂O*

Firouzeh Nemati et al. reported that three-component Mannichtype reaction of acetophenon, aromatic aldehydes, and aromatic amines is catalyzed by 2,4,6- trichloro[1,3,5]triazine at ambient temperature in EtOH and solvent-free conditions. The important properties of this procedure are good yield, use of cheap and non-hazardous catalyst, and simple procedure (Firouzeh Nemati et al.. 2010).

Figure 33. *Mannich reaction via TCT*

Deepak. M. Nagrik et al. declared that effective three-component condensation in the existence of MgO/ZrO_2 as an cheap and potent catalyst for the synthesis of β -amino carbonyl compounds by Mannich reaction was explained. The present methodology proposed several advantages, like good yields, short reaction times and a recyclable catalyst (Deepak. M. Nagrik et al. 2010).

Figure 34. *Synthesis of* β *-amino carbonyl compounds*

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NEW CHARACTERIZATIONS FOR THE SLANT HELICES OF A TIMELIKE CURVE DUE TO N-BISHOP ALTERNATIVE MODIFIED FRAME



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INTRODUCTION

In differential geometry, many types of alternative frames have been produced to examine the properties of the curves. One of these alternative frames is the Bishop frame which was constructed by Bishop in 1975. The Bishop frame which is parallel to the tangent vector is a more useful frame than the Serret-Frenet frame without the need for a second derivative, [1]. Also, if the principal normal vector field of the curve makes a constant angle with a fixed straight line, then the curve is called as a slant helix which was firstly defined by Izumiya and Takeuchi in 2004, [2]. The slant helix has important applications in many branches of science such as DNA structure and has also been used very widely in differential geometry. For example, Kula et al. analyzed some features of slant helix and spherical indicators in 2005 and 2010, [3,4]. In 2009, Bükçü and colleagues introduced slant helix due to Bishop frame, [5]. In 2008, Bükçü and Karacan studied some properties of slant helix due to Bishop frame in Minkowski space, [6, 7]. Bekar et al. worked on the slant helix and their acceleration centers in Euclidean and Minkowski space [8,9]. Moreover, Koçyiğit et al. worked on the Bishop Darboux vector for the timelike curves, and calculated some of their characterizations, [10]. For the first time, S. Yilmaz and his colleagues obtained a new alternative frame called type-2 Bishop frame in 2010 [11]. The authors also defined a type-2 Bishop frame in Minkowski space [11,12]. Furthermore, numerous articles have been published in the literature related to the type-2 Bishop frame, S.Kızıltuğ et al. studied some characterizations of the slant helix due to the type-2 Bishop frame in 2013, [13]. Firstly, Scofield introduced a which new frame includes the the vectors $\{\vec{N}, \vec{C} = \vec{N}' / \|\vec{N}'\|, \vec{W} = \tau \vec{T} + \kappa \vec{B} / \|\kappa^2 + \tau^2\| \}$, in 1995, [14]. Then, Uzunoğlu et al. defined a new slant helix called C-slant helix in 2016 [15]. Keskin et al. described a new alternative frame called N-Bishop frame based by modified frame $\left\{ \vec{N}, \vec{C}, \vec{W} \right\}$, [16]. Then Samancı et al. developed the N-Bishop Darboux vector of a timelike curve, [17]. In this article, we examined some properties of the slant helices due to the N-Bishop frame of a timelike curve in the Minkowski-3 space. Some necessary and sufficient conditions to have a slant helix relative to the N-Bishop frame were given.

MATERIAL AND METHOD

A curve is said to be a helix if its unit tangent vector field makes a constant angle with a fixed direction. Similarly, a slant helix is a kind of helix in which the unit principal normal vector makes a constant angle with the fixed unit vector. In 1802, Lancret proved that a curve called a general helix when the ratio of curvature to torsion is constant. In 1975, Bishop found out a new alternative frame named Bishop frame which is relatively

parallel to the unit tangent field \vec{T} . The Bishop frame's derivative vector fields are

$$\vec{T}'(s) = k_1 \vec{N}_1(s) + k_2 \vec{N}_2(s), \quad \vec{N}_1'(s) = -k_1 \vec{T}(s), \quad \vec{N}_2'(s) = -k_2 \vec{T}(s)$$
 where k_1 and k_2 are the first and second curvatures of the Bishop frame [1]. Moreover, the derivation vector fields of the Bishop frame for a timelike curve in Minkowski 3-space is $\vec{T}' = k_1 \vec{N}_1 + k_2 \vec{N}_2, \quad \vec{N}_1' = k_1 \vec{T}, \quad \vec{N}_2' = k_2 \vec{T}$ where k_1 and k_2 are curvatures [5-7]. Yılmaz et al. introduced another Bishop frame called Type-2 Bishop frame in 2010. Type-2 Bishop frame is parallel to the binormal vector field, and the differentiation formulas of Type-2 Bishop frame are identified by

 $\vec{N}_1'(s) = -k_1\vec{B}(s), \quad \vec{N}_2'(s) = -k_2\vec{B}(s), \quad \vec{B}'(s) = k_1\vec{N}_1(s) + k_2\vec{N}_2(s)$ where k_1 and k_2 are curvatures, [11,12]. Another alternative moving frame

$$\left\{\vec{N}, \vec{C} = \vec{N}' / \left\| \vec{N}' \right\|, \vec{W} = \tau \vec{T} + \kappa \vec{B} / \left\| \kappa^2 + \tau^2 \right\| \right\}$$

which is parallel to the normal vector field is introduced by Scofield in 1995, [14]. The derivation formulas of the alternative moving frame $\left\{\vec{N},\vec{C},\vec{W}\right\}$ are computed by Uzunoğlu et al. as $\vec{N}'=f\vec{C}$, $\vec{C}'=-f\vec{N}+g\vec{W}$, $\vec{W}'=-g\vec{C}$ where the functions $f=\sqrt{\kappa^2+\tau^2}$ and $g=\kappa^2\left(\frac{\tau}{\kappa}\right)'/(\kappa^2+\tau^2)=\sigma f$ are the differentiable, [15]. As a new alternative frame N-Bishop frame were first introduced by Keskin and Yaylı in 2017, [16]. The derivative vector fields of the N-Bishop frame is given by $\vec{N}'=k_1\vec{N}_1+k_2\vec{N}_2$, $\vec{N}_1'=-k_1\vec{N}$, $\vec{N}_2'=-k_2\vec{N}$. Furthermore, the

Kusak Samancı. The derivative vector fields of the curve are $\vec{N}' = k_1 \vec{N}_1 - k_2 \vec{N}_2$, $\vec{N}'_1 = k_1 \vec{N}$, $\vec{N}'_2 = k_2 \vec{N}$, and its transition equations was given by

N-Bishop frame of a timelike curve in Minkowski 3-space is studied on by

$$\vec{N}, \quad \vec{C} = \cosh\psi \vec{N}_1 + \sinh\psi \vec{N}_2, \quad \vec{W} = \sinh\psi \vec{N}_1 + \cosh\psi \vec{N}_2$$
where
$$f = \sqrt{\kappa^2 + \tau^2} = \sqrt{\left|k_2^2 - k_1^2\right|} \quad \text{and}$$

 $\psi(s) = \int_{s_0}^{s} (-g(t)) dt = -\tanh^{-1}(k_2/k_1)$. Since a curve α is a timelike curve, then T is timelike, N and B are spacelike, whereas $\langle \vec{T}, \vec{T} \rangle_L = -1, \langle \vec{N}, \vec{N} \rangle_L = 1, \langle \vec{B}, \vec{B} \rangle_L = 1$. For the adapted N-Bishop frame

of the timelike curve \vec{N} , \vec{N}_1 , \vec{N}_2 are spacelike, timelike and spacelike, respectively. Furthermore, the vectors satisfy the following equations $\left\langle \vec{N}, \vec{N} \right\rangle_L = 1$, $\left\langle \vec{N}_1, \vec{N}_1 \right\rangle_L = -1$, $\left\langle \vec{N}_2, \vec{N}_2 \right\rangle_L = 1$ and $\vec{N} \times_L \vec{N}_1 = \vec{N}_2$, $\vec{N}_1 \times_L \vec{N}_2 = \vec{N}$, $\vec{N}_2 \times_L \vec{N} = -\vec{N}_1$, [17].

MAIN RESULTS

Definition 3.1. Let a curve $\xi: I \to E_1^3$ be a unit speed timelike curve in Minkowski 3-space with $\xi \neq 0$. If the vector field \vec{N}_1 makes a constant angle φ with a fixed certain direction u in Minkowski 3-space, then ξ is called a timelike slant helix due to N-Bishop frame $\left\{\vec{N}, \vec{N}_1, \vec{N}_2\right\}$, that is $\left\langle\vec{N}_1(s), \vec{u}\right\rangle = \cosh \varphi$.

Theorem 3.2. Suppose that $\xi: I \to E_1^3$ is a slant helix of a unit speed timelike curve. The timelike curve ξ is a timelike slant helix necessary and sufficient condition if the ratio of the curvatures k_1 and k_2 satisfy the condition k_1/k_2 is constant.

Proof. Let the timelike curve ξ be a timelike slant helix of a in E_1^3 . Then \vec{N}_1 holds the condition $\langle \vec{N}_1, \vec{u} \rangle = \cosh \varphi$ (const), where φ is constant angle for $\forall s \in I$. Calculating the derivation of $\langle \vec{N}_1, \vec{u} \rangle = \cosh \varphi$, we obtain $\langle \vec{N}, \vec{u} \rangle = 0$. Now if the derivation of $\langle \vec{N}, \vec{u} \rangle = 0$ is calculated, we found also the equation $\langle \vec{N}', \vec{u} \rangle = 0$. By substituting the Eq. (2.1) in this equation, we get the $\langle k_1 \vec{N}_1 - k_2 \vec{N}_2, \vec{u} \rangle = 0$ and $k_1/k_2 = \tanh \varphi$ (const.) . If $\langle \vec{N}, \vec{u} \rangle = 0$, then the vector $\vec{u} \in s_p \left\{ \vec{N}_1, \vec{N}_2 \right\}$ is written with a linear combination of a timelike vector \vec{N}_1 and spacelike vector \vec{N}_2 as

$$\vec{u} = (-\cosh\varphi)\vec{N}_1 + (\sinh\varphi)\vec{N}_2.$$
(3.1)

By differentiation of the Equation (3.1), we get $\vec{u}' = (-\cosh\varphi)\vec{N}_1' + (\sinh\varphi)\vec{N}_2' = [-k_1(\cosh\varphi) + k_2(\sinh\varphi)]\vec{N} = 0.$

If the derivation is obtain zero, then \vec{u} is a constant vector.

Conversely, assume that k_1/k_2 is constant. So that the ratio can be rewritten as $k_1/k_2=\lambda$ where the coefficient can be taken as $\lambda=\tanh\varphi$. Thanks to this equation, we have $k_1/k_2=\tanh\varphi$, $k_1(\cosh\varphi)-k_2(\sinh\varphi)=0$. Then from Eq.(3.1), it is shown that the vector \vec{u} is constant and the inner product $\langle \vec{N}_1, \vec{u} \rangle$ is constant. From Eq.(3.1), we get $\langle \vec{N}_1, \vec{u} \rangle = \cosh\varphi$. Then one can easily see that ξ is a timelike slant helix. Finally, the theorem is proven.

Theorem 3.3. Assume that $\xi: I \to E_1^3$ be a unit speed timelike curve due to the N-Bishop frame with the curvatures k_1 and k_2 . The curve ξ is a slant helix necessary and sufficient condition $\det(\vec{N}_1', \vec{N}_1'', \vec{N}_1''') = 0$.

Proof. Assume that the curve ξ is a slant helix of a timelike curve in Minkowski 3-space. Thus the ratio k_1/k_2 is constant. Taking the derivation of \vec{N}_1 in the Eq. (2.1), we have $\vec{N}_1'' = k_1' \vec{N} + k_1^2 \vec{N}_1 - k_1 k_2 \vec{N}_2$ and the third derivative of \vec{N}_1 is obtained by

 $\vec{N}_{1}^{""}=2k_{1}k_{1}^{\prime}\vec{N}_{1}+k_{1}^{2}(k_{1}\vec{N})+k_{1}^{"}\vec{N}+k_{1}^{\prime}(k_{1}\vec{N}_{1}-k_{2}\vec{N}_{2})$

$$-k_1'k_2\vec{N}_2-k_2'k_1\vec{N}_2-k_1k_2(k_2\vec{N})\\ =(k_1^3+k_1''-k_1k_2^2)\vec{N}+(3k_1k_1')\vec{N}_1+(-2k_1'k_2-k_2'k_1)\vec{N}_2.$$
 In this way, the equation is calculated as $\det\left(\vec{N}_1',\vec{N}_1'',\vec{N}_1'''\right)=k_1^3k_2^2\left[k_1/k_2\right]'.$ It is well known that if ξ is a slant helix, then the ratio of curvatures k_1/k_2 is constant. So we have $\left[k_1/k_2\right]'=0$. Hereby, the outcome is obtained as $\det\left(\vec{N}_1',\vec{N}_1'',\vec{N}_1'''\right)=0.$ Otherwise, suppose that $\det\left(\vec{N}_1',\vec{N}_1'',\vec{N}_1''',\vec{N}_1'''\right)=0$. Since the determination can be written as $\det\left(\vec{N}_1',\vec{N}_1'',\vec{N}_1'',\vec{N}_1'''\right)=k_1^3k_2^2\left[k_1/k_2\right]'=0$ where $k_1\neq 0$ and $k_2\neq 0$, by the simplification of this equation we obtain $\left[k_1/k_2\right]'=0$ and $k_1/k_2=\mathrm{const}.$ Therefore the timelike curve ξ is a slant helix.

Theorem 3.4. Suppose that the curve $\xi: I \to E^3$ is a unit speed timelike curve in Minkowski 3-space. The curve ξ is timelike a slant helix if and only if $\det(\vec{N}_2', \vec{N}_2'', \vec{N}_2''') = 0$.

Proof. The proof is analogous to the proof of Theorem 3.3.

Theorem 3.5. Let the unit speed curve $\xi: I \to E^3$ be a timelike curve in Minkowski 3-space. The timelike curve ξ is a slant helix necessary and sufficient condition is $\det(\vec{N}', \vec{N}'', \vec{N}''') = 0$.

Proof. Let ξ be a timelike slant helix in Minkowski 3-space. The first, second and third differentiations of N in Eq. (2.1) are

$$\vec{N}' = k_1 \vec{N}_1 - k_2 \vec{N}_2, \quad \vec{N}'' = (k_1^2 - k_2^2) \vec{N} + k_1' \vec{N}_1 - k_2' \vec{N}_2,$$

$$\vec{N}''' = (3k_1 k_1' - 3k_2 k_2') \vec{N} + (k_1'' + k_1^3 - k_1 k_2^2) \vec{N}_1 + (-k_2'' - k_1^2 k_2 + k_2^3) \vec{N}_2.$$
The determinant of the first, second and third differentiations can be

$$\det\left(\vec{N}', \vec{N}'', \vec{N}'''\right) = \left(3k_1k_2^2k_1' - 3k_2^3k_2'\right) \left(\frac{k_1}{k_2}\right)'$$
compute by
$$-\left[\left(\frac{k_1}{k_2}\right)^2 - 1\right] \left[\left(\frac{k_1}{k_2}\right)'' k_2^4 + 2k_2^3k_2'\left(\frac{k_1}{k_2}\right)'\right].$$

(3.2)

As we know that ξ is a timelike slant helix, then the ratio of the curvatures is constant. Therefore, we get $\left[k_1/k_2\right]'=0$. Also it is seen that from the Eq. (3.2) is zero. Otherwise, suppose that $\det\left(\vec{N}',\vec{N}'',\vec{N}'''\right)=0$.

Then, using the above equation, we have $\left[k_1/k_2\right]'=0$ and $k_1/k_2=$ const. Then it is said to be ξ is a timelike slant helix. Then the proof is finished. Now, assume that ξ is a slant helix parameterized by the unit speed parameter on a Minkowski space. N-Bishop frame's covariant differentiation due to the tangent vector $\alpha'(s)=T$ is computed by

$$D_T \vec{N} = k_1 \vec{N}_1 - k_2 \vec{N}_2, D_T \vec{N}_1 = k_1 \vec{N}, D_T \vec{N}_2 = k_2 \vec{N}.$$
 (3.3)
where $\vec{N}_1(s)$ and $\vec{N}_2(s)$ are two N-Bishop vector fields, k_1 and

 k_2 are the curvature functions of the curve with each $s \in I$ parameters.

Theorem 3.6. Assume that the curve $\xi: I \to E_1^3$ is a unit speed timelike curve in Minkowski 3-space and the N-Bishop frame curvatures of the curve are denoted by k_1 and k_2 . The curve ξ is a timelike slant helix if and only if the following equation is provided

$$D_T(D_T D_T \vec{N}_1) = D_T \vec{N}_1 \left(\frac{k_1''}{k_1} - k_2^2 + k_1^2\right) + 3k_1' D_T \vec{N}. \tag{3.4}$$

Proof. If the curve ξ is taken as a slant helix, using the Eq. (3.4) and (3.3) the covariant differentiation of $D_T \vec{N}_1$ is obtained by

$$D_T(D_T\vec{N}_1) = D_T(k_1\vec{N}) = k_1'\vec{N} + k_1^2\vec{N}_1 - k_1k_2\vec{N}_2, \tag{3.5}$$

Also the covariant derivativation of above equation is computed by

$$D_{T}(D_{T}D_{T}\vec{N}_{1}) = D_{T}(k_{1}'\vec{N} + k_{1}^{2}\vec{N}_{1} - k_{1}k_{2}\vec{N}_{2}),$$

$$= k_{1}''\vec{N} + k_{1}'D_{T}\vec{N} + 2k_{1}k_{1}'\vec{N}_{1} - (k_{1}'k_{2} + k_{1}k_{2}')\vec{N}_{2} \quad (3.6)$$

$$-k_{1}k_{2}(k_{2}\vec{N}) + k_{1}^{2}D_{T}\vec{N}_{1}.$$

From the Eq.(3.6), it is easy to say that the ratio k_1/k_2 is constant. From the equation $(k_1/k_2)'=0$, we have $k_1'k_2=k_1k_2'$. Moreover, from the Eq. (3.3), the equation $N=\frac{1}{k_1}D_T\vec{N}_1$ is found. Now, by substituting these

equations in the Eq.(3.6), then the third covariant differentiation of \vec{N}_1 is obtained by

$$\begin{split} D_T(D_T D_T \vec{N}_1) &= k_1'' \vec{N} - k_1 k_2^2 \vec{N} + k_1' D_T \vec{N} + 2k_1' \left(\underbrace{k_1 \vec{N}_1 - k_2 \vec{N}_2}_{D_T \vec{N}} \right) + k_1^2 D_T \vec{N}_1, \\ &= D_T \vec{N}_1 \left(\frac{k_1''}{k_1} - k_2^2 + k_1^2 \right) + 3k_1' D_T \vec{N}. \end{split}$$

On the other hand, let's suppose that the equality is satisfied in (3.4). Now, we will prove that ξ is a timelike slant helix. From the first and second covariant derivation of (3.8), we get

$$D_{T}\vec{N} = -\frac{k_{1}'}{k_{1}^{2}}D_{T}\vec{N}_{1} + \frac{1}{k_{1}}D_{T}D_{T}\vec{N}_{1}.$$

$$D_T D_T \vec{N} = -\left(\frac{k_1'}{k_1^2}\right)' D_T \vec{N}_1 - \frac{k_1'}{k_1^2} D_T D_T \vec{N}_1 - \frac{k_1'}{k_1^2} D_T D_T \vec{N}_1 + \frac{1}{k_1} D_T D_T \vec{N}_1.$$
 (3.7)

By substituting (3.4) in (3.7), the covariant derivative can also be computed by

$$\begin{split} D_T D_T \vec{N} &= - \left(\frac{k_1'}{k_1^2} \right)' D_T \vec{N}_1 - \frac{k_1'}{k_1^2} D_T D_T \vec{N}_1 - \frac{k_1'}{k_1^2} D_T D_T \vec{N}_1 \\ &+ \frac{1}{k_1} \left(D_T \vec{N}_1 \left(\frac{k_1''}{k_1} - k_2^2 + k_1^2 \right) + 3k_1' D_T \vec{N} \right) \\ &= \left(- \left(\frac{k_1'}{k_1^2} \right)' + \frac{k_1''}{k_1^2} - \frac{k_2^2}{k_1} + k_1 \right) D_T \vec{N}_1 \\ &- \frac{2k_1'}{k_1^2} D_T D_T \vec{N}_1 + \frac{3k_1'}{k_1} D_T \vec{N}. \end{split}$$

By replacing the Eq (3.5) and Eq.(3.3) in the above equation and make the necessary arrangements, then we have

$$D_{T}D_{T}\vec{N} = \left(-\left(\frac{k_{1}'}{k_{1}^{2}}\right)' + \frac{k_{1}''}{k_{1}^{2}} - \frac{k_{2}^{2}}{k_{1}} + k_{1}\right)D_{T}\vec{N}_{1}$$

$$-\frac{2k_{1}'}{k_{1}^{2}}\left(k_{1}'\vec{N} + k_{1}^{2}\vec{N}_{1} - k_{1}k_{2}\vec{N}_{2}\right) + \frac{3k_{1}'}{k_{1}}\left(k_{1}\vec{N}_{1} - k_{2}\vec{N}_{2}\right)$$

$$= \left(-\left(\frac{k_{1}'}{k_{1}^{2}}\right)' + \frac{k_{1}''}{k_{1}^{2}} - \frac{k_{2}^{2}}{k_{1}} + k_{1}\right)D_{T}\vec{N}_{1} + k_{1}'\vec{N}_{1} - \frac{k_{1}'k_{2}}{k_{1}}\vec{N}_{2} - \frac{2(k_{1}')^{2}}{k_{1}^{2}}\vec{N}. \quad (3.8)$$

Otherwise, using the Eq.(3.3), the second covariant derivative $D_T D_T \vec{N}$ can also be found by

$$D_T D_T \vec{N} = k_1' \vec{N}_1 + k_1 D_T \vec{N}_1 - k_2' \vec{N}_2 - k_2^2 \vec{N}. \tag{3.9}$$

Now, by equating (3.8) and (3.9), we get $\frac{k_1'k_2}{k_1} = k_2'$. Then after

some arrangements, we get $(k_1/k_2)' = 0$. This result shows that k_1/k_2 is constant. Therefore the curve ξ is a timelike slant helix.

Theorem 3.7. Assume that $\xi: I \to E_1^3$ is a unit speed timelike curve

in a Minkowski 3-space. The timelike curve ξ is a slant helix necessary and sufficient condition is

$$D_T(D_T D_T \vec{N}_2) = D_T \vec{N}_2 \left(\frac{k_2''}{k_2} - k_2^2 + k_1^2\right) - 3k_2' D_T \vec{N}.$$

Proof. Let ξ be a slant helix of a timelike curve in Minkowski 3-space. Using Eq.(3.3), the second covariant differentiation of \vec{N}_2 is obtained $D_T(D_T\vec{N}_2) = k_2'\vec{N} + k_1k_2\vec{N}_1 - k_2^2\vec{N}_2$. Moreover, the third covariant derivativation is obtained by $D_T(D_TD_T\vec{N}_2) = D_T(k_2'\vec{N} + k_1k_2\vec{N}_1 - k_2^2\vec{N}_2)$ $= k_2''\vec{N} + k_1^2k_2\vec{N} + k_2'D_T\vec{N} + 2k_2'(\underbrace{k_1\vec{N}_1 - k_2\vec{N}_2}_{D_T\vec{N}}) - k_2^2D_T\vec{N}_2$

Using the equation $N = \frac{1}{k_2} D_T \vec{N}_2$, the third covariant derivativation can be written as

 $=(k_2''+k_1^2k_2)\vec{N}-k_2^2D_T\vec{N}_2+3k_2'D_T\vec{N}_2$

$$D_{T}(D_{T}D_{T}\vec{N}_{2}) = (k_{2}'' + k_{1}^{2}k_{2})(\frac{1}{k_{2}}D_{T}\vec{N}_{2}) - k_{2}^{2}D_{T}\vec{N}_{2} + 3k_{2}'D_{T}\vec{N}$$

$$= D_{T}\vec{N}_{2}(\frac{k_{2}''}{k_{2}} - k_{2}^{2} + k_{1}^{2}) - 3k_{2}'D_{T}\vec{N}.$$

Therefore the theorem is proven. Moreover, the sufficient condition can be similarly proven.

Theorem 3.8. Let $\xi: I \to E_1^3$ be a unit speed timelike curve in Minkowski 3-space. The curve ξ is a slant helix if and only if the equalities

$$\mathbf{i.} D_T(D_T D_T \vec{N}) = D_T \vec{N} \left(\frac{k_1''}{k_1} + k_1^2 - k_2^2 \right) + 3k_1' D_T \vec{N}_1 - 3k_2' D_T \vec{N}_2$$

$$\mathbf{ii.} D_T(D_T D_T \vec{N}) = D_T \vec{N} \left(\frac{k_2''}{k_2} + k_1^2 - k_2^2 \right) + 3k_1' D_T \vec{N}_1 - 3k_2' D_T \vec{N}_2$$

$$\mathbf{iii.}$$

$$D_T(D_T D_T \vec{N}) = D_T \vec{N} \left(\frac{k_1''}{2k_1} + \frac{k_2''}{2k_2} + k_1^2 - k_2^2 \right) + 3k_1' D_T \vec{N}_1 - 3k_2' D_T \vec{N}_2$$
 are separately provided.

Proof. Let ξ be a timelike slant helix in Minkowski 3-space. By substituting the covariant differentiation of \vec{N} in Eq.(3.3), we found the equation as

$$\begin{split} D_T(D_T\vec{N}) &= D_T(k_1\vec{N}_1 - k_2\vec{N}_2) \\ &= k_1'\vec{N}_1 + k_1D_T\vec{N}_1 - k_2'\vec{N}_2 - k_2D_T\vec{N}_2 \\ &= k_1^2\vec{N} - k_2^2\vec{N} + k_1'\vec{N}_1 - k_2'\vec{N}_2. \end{split}$$

Differentiating the Eq.(3.3) again, we get

$$\begin{split} D_T(D_T D_T \vec{N}) &= D_T(k_1^2 \vec{N} - k_2^2 \vec{N} + k_1' \vec{N}_1 - k_2' \vec{N}_2) \\ &= 3k_1' D_T \vec{N}_1 - 3k_2' D_T \vec{N}_2 + D_T \vec{N}(k_1^2 - k_2^2) + k_1'' \vec{N}_1 - k_2'' \vec{N}_2. \end{split}$$

It is known that if ξ is slant helix, then the fraction k_1/k_2 is constant. Thus we get $(k_1/k_2)'=0$ and then $k_1'/k_2'=k_1/k_2$. Because k_1/k_2 is constant, then k_1'/k_2' is constant. Also the properties $k_1''k_2'=k_1'k_2''$ and $k_1''/k_2''=k_1'/k_2'$ can be obtained from the differentiation. As a result, $k_1''/k_2''=k_1'/k_2'=k_1/k_2$ and $k_1''/k_1=k_2''/k_2=a$ are computed. Moreover, the phrase $k_1'''\vec{N}_1-k_2'''\vec{N}_2$ can be denoted by

$$k_1''\vec{N}_1 - k_2''\vec{N}_2 = \frac{k_1''}{k_1}k_1\vec{N}_1 - \frac{k_2''}{k_2}k_2\vec{N}_2$$
$$= a(k_1\vec{N}_1 - k_2\vec{N}_2)$$
$$= aD_T\vec{N}.$$

Consequently, the results are found as $k_1''\vec{N}_1 - k_2''\vec{N}_2 = \frac{k_1''}{k_1}D_T\vec{N}$ or

 $k_1''\vec{N}_1-k_2''\vec{N}_2=\frac{k_2''}{k_2}D_T\vec{N}$. Another characterizations can be also calculated by

$$\begin{split} D_T(D_T D_T \vec{N}) &= D_T \vec{N} (k_1^2 - k_2^2) + \frac{k_1''}{k_1} D_T \vec{N} + 3k_1' D_T \vec{N}_1 - 3k_2' D_T \vec{N}_2 \\ &= D_T \vec{N} \left(\frac{k_1''}{k_1} + k_1^2 - k_2^2 \right) + 3k_1' D_T \vec{N}_1 - 3k_2' D_T \vec{N}_2, \end{split}$$

and also can be written by

$$\begin{split} D_T(D_T D_T \vec{N}) &= D_T \vec{N} (k_1^2 - k_2^2) + \frac{k_2''}{k_2} D_T \vec{N} + 3k_1' D_T \vec{N}_1 - 3k_2' D_T \vec{N}_2 \\ &= D_T \vec{N} \left(\frac{k_2''}{k_2} + k_1^2 - k_2^2 \right) + 3k_1' D_T \vec{N}_1 - 3k_2' D_T \vec{N}_2. \end{split}$$

On the other hand, using the equality of

$$\begin{split} k_1'' \vec{N}_1 - k_2'' \vec{N}_2 &= D_T \vec{N} \left(\frac{k_1''}{2k_1} + \frac{k_2''}{2k_2} \right) \text{ we can get following characterization} \\ D_T (D_T D_T \vec{N}) &= D_T \vec{N} (k_1^2 - k_2^2) + D_T \vec{N} \left(\frac{k_1''}{2k_1} + \frac{k_2''}{2k_2} \right) + 3k_1' D_T \vec{N}_1 - 3k_2' D_T \vec{N}_2 \\ &= D_T \vec{N} \left(\frac{k_1''}{2k_1} + \frac{k_2''}{2k_2} + k_1^2 - k_2^2 \right) + 3k_1' D_T \vec{N}_1 - 3k_2' D_T \vec{N}_2. \end{split}$$

The sufficent condition can be proven similarly.

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INTRODUCTION

Spinors are used in many fields of mathematics and physics . In physics, spinors are used to describe the intrinsic angular momentum or spin of electron and fermision.

The relation between spinors and the Lorentz groups has been presented by Dirac (Dirac, 1974). Recently many authors studied spinor equations for diverse frames (Castillo, 2004; Ünal et al., 2013; Kişi and Tosun, 2015; Ketenci et al., 2015; Erisir et al., 2015; Balci et al., 2015)

Some concepts relation to this chapter are presented:

Minkowski 3-space R_1^3 is given with the metric as the following:

$$\langle , \rangle = dx_1^2 + dx_2^2 - dx_3^2.$$

For any vector $x = (x_1, x_2, x_3)$ in R_1^3 ,

if $\langle x, x \rangle > 0$, x is spacelike vector, if $\langle x, x \rangle < 0$, x is timelike vector. Spacelike or timelike curves are called as nonnull curves (O'Neill, 1983).

The derivative formulas of Frenet frame $\{T, n, b\}$ for $\beta: I \subset R \to R_1^3$ nonnull curve are

$$\begin{pmatrix}
T_s \\
n_s \\
b_s
\end{pmatrix} = \begin{pmatrix}
0 & \varepsilon_2 \kappa & 0 \\
-\varepsilon_1 \kappa & 0 & \varepsilon_3 b \\
0 & -\varepsilon_2 \tau & 0
\end{pmatrix} \begin{pmatrix}
T \\
n \\
b
\end{pmatrix}$$
(1)

where s is arclength parameter. Also

$$\langle T, T \rangle = \varepsilon_1 = \pm 1, \quad \langle n, n \rangle = \varepsilon_2 = \pm 1, \quad \langle b, b \rangle = \varepsilon_3 = \pm 1.$$

respectively, the tangent, normal and binormal vector fields are \it{T} , \it{n} and \it{b} .

Let $\{T, Q, N\}$ be the Darboux frame on oriented nonnull surface M in R_1^3 . T is the tangent vector field, N is the unit normal vector of nonnonnull curve β on nonnull surface M. The binormal vector field B is

given by $B = \varepsilon_2 T \times N$. The Frenet formulas of Darboux frame $\{T, Q, N\}$ are given by

$$\begin{pmatrix}
T_s \\
Q_s \\
N_s
\end{pmatrix} = \begin{pmatrix}
0 & \varepsilon_2 k_g & \varepsilon_3 k_n \\
-\varepsilon_1 k_g & 0 & \varepsilon_3 \tau_g \\
-\varepsilon_1 k_n & -\varepsilon_2 \tau_g & 0
\end{pmatrix} \begin{pmatrix}
T \\
Q \\
N
\end{pmatrix}$$
(2)

where

$$T \times Q = \varepsilon_3 N, \ Q \times N = \varepsilon_1 T, \ N \times T = \varepsilon_2 Q,$$

$$\langle T, T \rangle = \varepsilon_1 = \pm 1, \quad \langle Q, Q \rangle = \varepsilon_2 = \pm 1, \quad \langle N, N \rangle = \varepsilon_3 = \pm 1.$$

Consider the Bishop frame $\{T, N_1, N_2\}$ of a pseudo null curve β in R_1^3 satisfying (Grbovic and Nesovic, 2018):

$$\langle T_1, T_1 \rangle = \langle N_1, N_2 \rangle = 1, \quad \langle N_1, N_1 \rangle = \langle N_2, N_2 \rangle = \langle T_1, N_1 \rangle = \langle T_1, N_2 \rangle = 0$$
 (3)

where T_1 is the unit tangent vector field, N_1 and N_2 are two relatively parallel null normal vector fields. Connection between the Bishop frame $\{T, N_1, N_2\}$ and the Frenet frame $\{T, n, b\}$ of pseudo null curve are given by

$$\begin{pmatrix} T_1 \\ N_1 \\ N_2 \end{pmatrix} = \begin{pmatrix} 1 & 0 & 0 \\ 0 & \frac{1}{k_1} & 0 \\ 0 & 0 & k_2 \end{pmatrix} \quad \begin{pmatrix} T \\ n \\ b \end{pmatrix} \tag{4}$$

The Frenet equations of the Bishop frame $\{T, N_1, N_2\}$ are given by

$$\begin{pmatrix}
T_{1s} \\
N_{1s} \\
N_{2s}
\end{pmatrix} = \begin{pmatrix}
0 & k_2 & k_2 \\
-k_1 & 0 & 0 \\
-k_2 & 0 & 0
\end{pmatrix} \quad \begin{pmatrix}
T_1 \\
N_1 \\
N_2
\end{pmatrix} \tag{5}$$

where s is pseudo arclength parameter, k_1 and k_2 are curvatures of $\{T, N_1, N_2\}$ frame (Grbovic and Nesovic, 2018).

Let $\{\mathcal{T},\mathcal{N},\mathcal{B}\}$ be the Frenet frame in pseudo-Galilean 3-space satisfying

$$\mathcal{T}$$
 = $ho T$, \mathcal{N} = $ho n$, \mathcal{B} = $ho b$

where ρ is the radius of the curve β . The equiform curvature and torsion are

$$\widetilde{\kappa} = \rho, \qquad \widetilde{\tau} = \rho \tau = \frac{\tau}{\kappa}.$$

The Frenet formulas in the equiform geometry of pseudo Galilean 3-space G_1^3 are presented as the following (Erjavec and Dvjak, 2008):

$$\begin{pmatrix}
\mathcal{T}_{\sigma} \\
\mathcal{N}_{\sigma} \\
\mathcal{B}_{\sigma}
\end{pmatrix} = \begin{pmatrix}
\tilde{\kappa} & 1 & 0 \\
0 & \tilde{\kappa} & \tilde{\tau} \\
0 & \tilde{\tau} & \tilde{\kappa}
\end{pmatrix} \begin{pmatrix}
\mathcal{T} \\
\mathcal{N} \\
\mathcal{B}
\end{pmatrix}$$
(6)

where σ is an equiform invariant parameter.

The equiform differential geometry of the simple isotropic space $I_3^{(1)}$ has been studied by Pavkovic (Pavkovic, 1986). The derivative formulas of Frenet frame $\{\mathcal{T}, \mathcal{N}, \mathcal{B}\}$ according to equiform geometry of $I_3^{(1)}$ have the following form

$$\begin{pmatrix}
\mathcal{T}_{\sigma} \\
\mathcal{N}_{\sigma} \\
\mathcal{B}_{\sigma}
\end{pmatrix} = \begin{pmatrix}
\tilde{\kappa} & 1 & 0 \\
-1 & \tilde{\kappa} & \tilde{\tau} \\
0 & 0 & \tilde{\kappa}
\end{pmatrix} \begin{pmatrix}
\mathcal{T} \\
\mathcal{N} \\
\mathcal{B}
\end{pmatrix}$$
(7)

where σ is an equiform invariant parameter defined by $\sigma = \int \frac{ds}{\rho}$.

The derivative formulas of Frenet frame $\{\mathcal{T}, \mathcal{N}, \mathcal{B}\}$ of the double isotropic space $I_3^{(2)}$ have the following form (Pavkovic, 1986):

$$\begin{pmatrix}
\mathcal{T}_{\sigma} \\
\mathcal{N}_{\sigma} \\
\mathcal{B}_{\sigma}
\end{pmatrix} = \begin{pmatrix}
\tilde{\kappa} & 1 & 0 \\
0 & \tilde{\kappa} & \tilde{\tau} \\
0 & 0 & \tilde{\kappa}
\end{pmatrix} \begin{pmatrix}
\mathcal{T} \\
\mathcal{N} \\
\mathcal{B}
\end{pmatrix}$$
(8)

The vector $\eta = (\eta_1, \eta_2, \eta_3)$ consisting of hyperbolic symmetric matrices is given by (Pauli,1927)

$$\eta_1 = \begin{pmatrix} 1 & 0 \\ 0 & -1 \end{pmatrix}, \quad \eta_2 = \begin{pmatrix} j & 0 \\ 0 & j \end{pmatrix}, \quad \eta_3 = \begin{pmatrix} 0 & -1 \\ -1 & 0 \end{pmatrix},$$

$$\hat{\psi} = -\begin{pmatrix} 0 & 1 \\ -1 & 0 \end{pmatrix} \overline{\psi},$$

$$\hat{\psi} = -\psi,$$

A hyperbolic spinor in Minkowski 3-space can be given by $\psi = \begin{pmatrix} \psi_1 \\ \psi_2 \end{pmatrix}$, where $\psi_1, \psi_2 \in H$ (Ketenci et al., 2015).

For a, b $\in R_1^3$, $a + jb = (z_1, z_2, z_3)$

$$z_1^2 + z_2^2 - z_3^2 = 0$$
, $a + jb \neq 0$

$$a + jb = \psi^t \sigma \psi = (\psi_1^2 - \psi_2^2, j(\psi_1^2 + \psi_2^2), -2\psi_1 \psi_2), \tag{9}$$

$$c = -\hat{\psi}^t \sigma \psi = \overline{\psi}^t P \psi \tag{10}$$

$$= (\psi_1 \overline{\psi}_2 + \overline{\psi}_1 \psi_2, j(\psi_1 \overline{\psi}_2 - \psi_2 \overline{\psi}_1), |\psi_1|^2 - |\psi_2|^2)$$

where the mutually orthogonal vectors. The superscript t denotes transposition (Ketenci et al., 2015). $\overline{\psi}$ is complex of conjugation of ψ and $\hat{\psi}$ is the mate of ψ .

For $U \in SU(2;R)$,

$$\overline{\psi}^{\prime\prime}\psi^{\prime} = \psi^{\prime}\psi$$

$$\psi^{\prime} = U\psi$$

1. NEW SPINOR FRENET EQUATIONS IN R_1^3

In this part new spinor Frenet equations are introduced.

Theorem 1.1. If two-component second hyperbolic spinor ϕ yields the Frenet triad $\{n, T, b\}$ with timelike curve with spacelike binormal in R_1^3 , the Frenet equations are given by second hyperbolic spinor equation:

$$\frac{d\phi}{ds} = \frac{j\kappa}{2}\phi - \frac{\tau}{2}\hat{\phi}.$$

Proof. If the second hyperbolic spinor ϕ denotes the Frenet triad $\{n, T, b\}$, using (9) and (10) it can be written by

$$n + jT = \phi^t \eta \phi \qquad b = -\hat{\phi}^t \eta \phi \tag{11}$$

with $\overline{\phi}^t \phi = 1$. Also

$$\frac{d\phi}{ds} = \gamma_1 \phi + \gamma_2 \hat{\phi} \tag{12}$$

where γ_1 and γ_2 are functions and $\phi \neq 0$. Via (1) and differentiating of (11) according to s, we have

$$\kappa T + \tau b + j \kappa n = \frac{d\phi^t}{ds} \eta \phi + \phi^t \eta \frac{d\phi}{ds}$$
 (13)

Using (11) and (12) in (13),

$$\kappa T + \tau b + j\kappa n = 2\gamma_1(n+jT) - 2\gamma_2 b \tag{14}$$

From (14), it can be obtained

$$\gamma_1 = \frac{j\kappa}{2}, \quad \gamma_2 = -\frac{\tau}{2} \tag{15}$$

Using (15) in (12), it can be obtained:

$$\frac{d\phi}{ds} = \frac{j\kappa}{2}\phi - \frac{\tau}{2}\hat{\phi}.$$

Theorem 1.2. If new second spinor ϕ represents the Frenet triad $\{n, T, b\}$ for spacelike curve with timelike principal normal in R_1^3 , the Frenet equations are presented as the following:

$$\frac{d\phi}{ds} = -\frac{j\kappa}{2}\phi - \frac{\tau}{2}\hat{\phi}.$$

Proof. It is obvious from Theorem 1.1.

Theorem 1.3. If the third hyperbolic spinor Ω represents the Frenet triad $\{T, b, n\}$ of spacelike curve with timelike normal, the Frenet equations are given by third hyperbolic spinor equation, such that

$$\frac{d\Omega}{ds} = \left(\frac{\kappa - j\tau}{2}\right)\hat{\Omega}.$$

Proof. If the third hyperbolic spinor Ω denotes the Frenet triad $\{T, b, n\}$ via (9) and (10) the followings are derived

$$T + jb = \Omega^t \eta \Omega \qquad n = -\hat{\Omega}^t \eta \Omega \tag{16}$$

with $\overline{\Omega}^t \Omega = 1$.

 $\left\{\!\Omega,\hat{\Omega}\right\}$ is a basis two-component third hyperbolic spinors, such that

$$\frac{d\Omega}{ds} = v_1 \Omega + v_2 \hat{\Omega} \tag{17}$$

where v_1 and v_2 are functions. Using (1) and differentiating of (16) the followings are obtained:

$$-\kappa n + j \tau n = \frac{d\Omega^t}{ds} \eta \Omega + \Omega^t \eta \frac{d\Omega}{ds}$$
 (18)

Using (16) and (17) in (18), it can be written by

$$n(j\tau - \kappa) = 2\nu_1(T + jb) - 2\nu_2 n \tag{19}$$

From (19), it can be obtained

$$v_1 = 0, \qquad v_2 = \frac{\kappa - j\tau}{2} \tag{20}$$

Using (20) in (17), the third hyperbolic spinor representation of Frenet frame is obtained as following

$$\frac{d\Omega}{ds} = (\frac{\kappa - j\tau}{2})\hat{\Omega}.$$

2. NEW SPINOR DARBOUX EQUATIONS IN MINKOWSKI SPACE

In this part, second and third spinor formulations for Darboux equations in Minkowski 3-space R_1^3 are obtained via various theorems.

Theorem 2.1. If second spinor ψ represents the Darboux triad $\{T, Q, N\}$ of timelike curve β on timelike surface M in Minkowski 3-space, the Darboux equations have second spinor equation

$$\frac{d\psi}{ds} = -\frac{jk_g}{2}\psi + (\frac{k_n + j\tau_g}{2})\hat{\psi}$$

Proof. If second hyperbolic spinor ψ denotes Darboux triad $\{T, Q, N\}$ with first case, then from (9) and (10)

$$T + j Q = \psi^t \eta \psi , \qquad N = -\hat{\psi}^t \eta \psi$$
 (21)

with $\overline{\psi}^t \psi = 1$.

Since $\{\psi, \hat{\psi}\}$ is basis two-component second hyperbolic spinor, then

$$\frac{d\psi}{ds} = \phi_1 \psi + \phi_2 \hat{\psi} \tag{22}$$

where ϕ_1 and ϕ_2 are functions. From (2) and (22), it can be written by

$$jk_gT + k_gQ + (k_n + j\tau_g)N = \frac{d\psi^t}{ds}\eta\psi + \psi^t\eta\frac{d\psi^t}{ds}$$
 (23)

Using (21) in (23), it can be found

$$jk_{g}T + k_{g}Q + (k_{n} + j\tau_{g})N = 2\varphi_{1}(T + jQ) - 2\varphi_{2}N$$
 (24)

From (24), the followings are derived

$$\phi_1 = \frac{k_g j}{2}, \qquad \phi_2 = -\frac{k_n + j \tau_g}{2}$$
 (25)

Using (25) in (22) it can be obtained:

$$\frac{d\psi}{ds} = \frac{k_g j}{2} \psi - \left(\frac{k_n + j\tau_g}{2}\right) \hat{\psi}.$$

Theorem 2.2. If second spinor represents the Darboux triad $\{T, Q, N\}$ of spacelike curve β on timelike surface M in Minkowski 3-space according to (21), the Darboux equations have second spinor equation

$$\frac{d\psi}{ds} = -\frac{jk_g}{2}\psi - \left(\frac{k_n + j\tau_g}{2}\right)\hat{\psi}.$$

Proof is clear from Theorem 2.1.

Theorem 2.3. If the third hyperbolic spinor γ represents third Darboux triad $\{T, Q, N\}$ of a spacelike curve on spacelike surface in R_1^3 , then Darboux equations are equivalent third hyperbolic spinor equation

$$\frac{d\gamma}{ds} = -\frac{jk_n}{2}\gamma + \left(\frac{\tau_g j - k_n}{2}\right)\hat{\gamma}.$$

Proof. If the third hyperbolic spinor γ represents the Darboux triad $\{T, Q, N\}$ with spacelike curve on spacelike surface M, then there exists

$$T + jN = \gamma^{t} \eta \gamma, \quad N = -\hat{\gamma}^{t} \eta \gamma \tag{26}$$

with $\bar{\gamma}^t \gamma = 1$.

Since $\{\gamma, \hat{\gamma}\}$ is a basis for third two-component spinor, it can be obtained:

$$\frac{d\gamma}{ds} = \lambda_1 \gamma + \lambda_2 \hat{\gamma} \tag{27}$$

where λ_1 and λ_2 are functions. From (2) and (27), it can be derived:

$$-jk_{n}T + Q(k_{g} - j\tau_{g}) - k_{n}N = \frac{d\gamma^{t}}{ds}\eta\gamma + \gamma^{t}\eta\frac{d\gamma}{ds}$$
$$= 2\lambda_{1}(T + jN) - 2\lambda_{2}Q \tag{28}$$

From (28), it can be written by

$$\lambda_1 = -\frac{jk_n}{2}, \qquad \lambda_2 = \frac{\tau_g j - k_g}{2} \tag{29}$$

Using (29) in (27), the followings are derived:

$$\frac{d\gamma}{ds} = -\frac{jk_n}{2}\gamma + \left(\frac{\tau_g j - k_g}{2}\right)\hat{\gamma}.$$

Theorem 2.4. If the third hyperbolic spinor γ represents third Darboux triad $\{T, Q, N\}$ of a timelike curve on timelike surface M, then Darboux equations are equivalent to third hyperbolic spinor equation

$$\frac{d\gamma}{ds} = \frac{jk_n}{2}\gamma + \left(\frac{\tau_g j + k_g}{2}\right)\hat{\gamma}.$$

Proof. From Theorem 2.3 proof is obvious.

3. SPINOR BISHOP EQUATIONS OF PSEUDO NULL CURVE

In this part spinor Frenet equations according to Bishop frame of pseudo-null curve β are discussed.

Theorem 3.1. If two-component hyperbolic spinor denotes the Bishop triad $\{N_1, N_2, T\}$ of pseudo null curve β , then the Frenet equations of Bishop frame of the curve β are given by the single hyperbolic spinor equation:

$$\frac{d\Gamma}{ds} = \frac{jk}{2}\Gamma + \frac{\tau}{2}\hat{\Gamma}.$$

Proof. Let $\beta: I \subset R \to R_1^3$ be a pseudo null curve with pseudo arc length s. If the hyperbolic spinor Γ represents the Bishop triad $\{N_1, N_2, T\}$ of pseudo null curve, then there exists the hyperbolic spinor Γ such that

$$N_1 + jN_2 = \Gamma^t \eta \Gamma \qquad T = -\hat{\Gamma}^t \eta \Gamma \tag{30}$$

with $\overline{\Gamma}^t \Gamma = 1$.

Since $\left\{\Gamma, \hat{\Gamma}\right\}$ is a basis two-component second hyperbolic spinors, it can be written by

$$\frac{d\Gamma}{ds} = p_1 \Gamma + p_2 \hat{\Gamma} \tag{31}$$

where p_1 and p_2 are functions and $\Gamma \neq 0$. From (5), (31) and differentiating of (30) according to s, it can be written by

$$-(k_1 + jk_2)T = 2p_1(N_1 + jN_2) - 2p_2T$$
(32)

From (19), it can be found

$$p_1 = 0, p_2 = \frac{k_1 + jk_2}{2} (33)$$

Substituting (33) in (31), the single hyperbolic spinor representation of Bishop frame of pseudo null curve is derived as

$$\frac{d\Gamma}{ds} = \left(\frac{k_1 + jk_2}{2}\right)\hat{\Gamma}.$$

4. SPINOR FRENET EQUATIONS IN THE EQUIFORM GEOMETRY OF G_1^3 AND $I_3^{(1)}$, $I_3^{(2)}$

In this part spinor Frenet equations according to the equiform geometry of the simple isotropic space space $I_3^{(1)}$, the double isotropic space $I_3^{(2)}$ and pseudo-Galilean 3-space G_1^3 are studied.

Theorem 4.1. If two-component spinor Θ denotes the Frenet triad $\{\mathcal{T}, \mathcal{B}, \mathcal{N}\}$ of the simple isotropic space $I_3^{(1)}$, the Frenet equations are given by the single hyperbolic spinor equation

$$\frac{d\Theta}{d\sigma} = \frac{j\kappa}{2}\Theta + \frac{\tau}{2}\hat{\Theta}.$$

Proof. If spinor Θ represents the Frenet triad $\{\mathcal{T},\mathcal{B},\mathcal{N}\}$, it can be obtained

$$\mathcal{T} + j\mathcal{B} = \Theta^t \eta \Theta, \qquad \mathcal{N} = -\hat{\Theta}^t \eta \Theta$$
 (34)

with $\overline{\Theta}^t \Theta = 1$. Also

$$\frac{d\Theta}{d\sigma} = \varsigma_1 \Theta + \varsigma_2 \hat{\Theta} \tag{35}$$

Here ζ_1 and ζ_2 are functions. From (7), (34) and differentiating of (35) according to σ it can be obtained

$$\tilde{\kappa} \mathcal{T} + \mathcal{N} + j \tilde{\kappa} \mathcal{B} = 2\varsigma_1 \mathcal{T} + 2\varsigma_1 \mathcal{B} - 2\varsigma_2 \mathcal{N}$$
(36)

From (36) it can be derived

$$\varsigma_1 = \frac{\widetilde{\kappa}}{2}, \ \varsigma_2 = -\frac{1}{2} \tag{37}$$

Using (37) in (31), it can be obtained

$$\frac{d\Theta}{d\sigma} = \frac{\widetilde{\kappa}}{2}\Theta + \frac{1}{2}\hat{\Theta} .$$

Theorem 4.2. If two-component spinor represents the Frenet triad $\{\mathcal{T}, \mathcal{B}, \mathcal{N}\}$ in the equiform geometry of double isotropic space $I_3^{(2)}$ then the Frenet equations are written by

$$\frac{d\varsigma}{d\sigma} = \frac{\widetilde{\kappa}}{2}\varsigma - \frac{1}{2}\hat{\varsigma}$$

Proof. If the spinor ς represents with the Frenet triad $\{\mathcal{T}, \mathcal{B}, \mathcal{N}\}$, then

$$\mathcal{T} + j\mathcal{B} = \varsigma^t \eta \varsigma \qquad \mathcal{N} = -\hat{\varsigma}^t \eta \varsigma$$
 (38)

with $\bar{\varsigma}^t \varsigma = 1$. Then

$$\frac{d\varsigma}{d\sigma} = q_1 \varsigma + q_2 \hat{\varsigma} \tag{39}$$

Here q_1 and q_2 are functions. Using (38) and differentiating of (39) according to σ , it can be obtained

$$\tilde{\kappa}\mathcal{T} + \mathcal{N} + j\tilde{\kappa}\mathcal{B} = 2q_1\mathcal{T} + 2jq_1\mathcal{B} - 2q_2\mathcal{N}$$
(40)

From (40) the followings are found

$$q_1 = \frac{\tilde{\kappa}}{2}, \quad q_2 = -\frac{1}{2} \tag{41}$$

Using (41) in (39), spinor Frenet representation in the double isotropic space is given by

$$\frac{d\varsigma}{d\sigma} = \frac{\widetilde{\kappa}}{2}\varsigma - \frac{1}{2}\hat{\varsigma}$$

Theorem 4.3. If the spinor ξ represents the Frenet triad $\{\mathcal{T}, \mathcal{B}, \mathcal{N}\}$ of the equiform geometry in the pseudo Galilean space, then the Frenet equations are expressed by

$$\frac{d\xi}{d\sigma} = \frac{\widetilde{\kappa}}{2} \xi - \frac{(j\widetilde{\tau} + 1)}{2} \hat{\xi}$$

Proof. If the spinor ξ represents the Frenet triad $\{\mathcal{T}, \mathcal{B}, \mathcal{N}\}$, there exists the spinor ζ , such that

$$\mathcal{T} + j\mathcal{B} = \xi^t \eta \xi, \qquad \mathcal{N} = -\hat{\xi}^t \eta \xi$$
 (42)

with $\bar{\xi}^t \xi = 1$. Thus

$$\frac{d\xi}{d\sigma} = \delta_1 \xi + \delta_2 \hat{\xi} \tag{43}$$

Here δ_1 and δ_2 are functions. Using (6), (42) and differentiating of (43) according to σ it is obtained

$$\tilde{\kappa}\mathcal{T} + (1+j\tilde{\tau})\mathcal{N} + j\tilde{\kappa}\mathcal{B} = 2\delta_1\mathcal{T} + 2j\delta_1\mathcal{B} - 2\delta_2\mathcal{N}$$
(44)

From (44) the followings are derived

$$\delta_1 = \frac{\widetilde{\kappa}}{2} , \quad \delta_2 = -\frac{(j\widetilde{\tau} + 1)}{2}$$
 (45)

Using (45) in (43), spinor Frenet representation in the equiform geometry of G_1^3 is presented as the following

$$\frac{d\xi}{d\sigma} = \frac{\widetilde{\kappa}}{2} \xi - \frac{(j\widetilde{\tau} + 1)}{2} \hat{\xi}$$

Theorem 4.4. If the spinor χ represent the Frenet triad $\{\mathcal{N}, \mathcal{B}, \mathcal{T}\}$ $\{\mathcal{N}, \mathcal{B}, \mathcal{T}\}$ according to the equiform geometry of pseudo Galilean space, the Frenet equations are given by

$$\frac{d\chi}{d\sigma} = \frac{\tilde{\kappa} + j\tilde{\tau}}{2}\chi$$

Proof. If the spinor χ represents the Frenet triad $\{\mathcal{N}, \mathcal{B}, \mathcal{T}\}$ in the equiform geometry of pseudo Galilean 3-space. There is exists

$$\mathcal{N} + j\mathcal{B} = \chi^t \eta \chi \,, \quad \mathcal{T} = -\hat{\chi}^t \eta \chi \tag{46}$$

with $\bar{\chi}^t \chi = 1$. $\{\chi, \hat{\chi}\}$ is a basis for two-component spinors, such that

$$\frac{d\chi}{d\sigma} = \omega_1 \chi + \omega_2 \hat{\chi} \tag{47}$$

where ω_1 and ω_2 are functions and $\chi \neq 0$. Using (46) and differentiating of (47) according to σ , it can be derived

$$(\tilde{\kappa}\mathcal{N} + \tilde{\tau}\mathcal{B}) + j(\tilde{\tau}\mathcal{N} + \tilde{\kappa}\mathcal{B}) = -2\omega_2\mathcal{T} + 2\omega_1\mathcal{N} + 2\omega_1j\mathcal{B}$$
 (48)

From (48) it is found

$$\omega_1 = \frac{\widetilde{\kappa} + j\widetilde{\tau}}{2}, \quad \omega_2 = 0$$
 (49)

Using (49) in (47), second spinor Frenet representation is given by

$$\frac{d\chi}{d\sigma} = \frac{\widetilde{\kappa} + j\widetilde{\tau}}{2} \chi.$$

CONCLUSIONS

In this chapter spinor Bishop equations of pseudo null curve, spinor Frenet equations in the equiform geometry of simple isotropic space $I_3^{(1)}$ and double isotropic space $I_3^{(2)}$ and spinor equations in the equiform geometry of the pseudo Galilean 3-space have been presented.

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EFFECTIVE ATOMIC NUMBER AND ELECTRON DENSITY DETERMINATION OF CE AND SOME SELECTED CE COMPOUNDS



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1. Introduction

One of the most fundamental events that physics must explain is the interactions between fundamental particles and fields. As an example of this interaction, the observation and understanding of the scattering of photons from free electrons is of great importance in the development of quantum mechanics. The scattering events of X or gamma rays are occurring in different types. Inmedical and biological applications Rayleigh and the Compton scattering is important than the photoelectric effects of X or gamma rays [1]. The Compton scattering is occurring in the collision of X or gamma rays with free electrons. The main objective of this study is the implementation of the Compton scattering in the determination of the effective atomic number.

The elements are represented by their unique atomic numbers. However, there is no specialized description method for compounds and mixtures. For this purpose, the atomic numbers of compounds consisting of two or more elements are referred to as the effective atomic number. The scientists have engaged in different studies to determine the effective atomic number of compounds. Some of them were examined the effective atomic number by mass absorption method. In recent years, it has become common to determine the effective atomic number by scattering method. Both determination techniques are non-destructive and used in X-ray fluorescence spectroscopy investigation experiments. The X-ray fluorescence spectroscopy experiments are useful in the non-destructive determination of atomic properties of biological, environmental, agricultural materials.

This experimental study was dedicated on the effective atomic number and electron density determination of Ce and selected Ce compounds. The target samples cerium is a lanthanide element and most abundant in nature. The Ce is found in monazite and bastnaesite mineral composition. The usage of Ce element is very common, and the cerium oxide is widely used in commercial transactions. Cerium replaces other metal oxides used in high-quality optical surfaces. Cerium compounds are one of the components sought in lighting systems. It is an important substance used in iron alloys and widely used to increase the high temperature oxidation resistance of superalloys. Cerium, an important component for phosphors used in television screens and fluorescent lamps, prevents the formation of polymers in pigments exposed to sunlight. Some areas and products using cerium, its compounds and alloys are as follows; glass polish, catalytic converters, pigments, television glass plates, jet engines alloys, carbon arc lamps, porcelain coatings, energy saving bulbs, projectors, magnets, stainless steel, incandescent lamps, magnetic optical compact discs, chrome finishes, spectacles, self-cleaning surfaces of furnaces. Also, Ce was used in superconductor applications in different compositions. Although not as widely used as MgB_2 it is possible to find Ce-doped superconductors. There are also many studies on MgB₂ superconductors. These are usually about investigating their electrical properties and effects resulting from different doping. [2-3]

Important studies have been included in this field by many researchers. Hosamani and Badiger [4]have been determined the effective atomic number of composite materials by using the backscattered photons that were emitted from ¹²⁷Cs radioactive isotope. The backscattered photons were detected via "2x2"NaI(Tl) scintillation gamma ray spectrometer that was coupled with 16 K multi-channel analyzer. The effective atomic number of target samples has been determined by knowing the saturation thicknesses of pure samples. Then, the saturation thickness of pure materials was used to measure the effective atomic number determination of composite materials. Babu and Badiger[5] have

been measured the stopping power, cross section and the effective atomic number of amino acids. The interaction between electron and sample was called narrow beam effect. They have used the narrow beam geometry setup that was coupled with Si(Li) semiconductor detector and 8K multichannel analyzer. Singh et al. [6] have been determined the effective atomic number of target samples by using the scattered gamma photons that were emitted from ²⁰³Hg radioisotope. The emitted gamma photons from the source have 279 keV energy. Rayleigh to Compton scattering intensity ratio of incident gamma photons was used in determination of brass, bronze, soldering material, Perspex and bakelite composite samples. Similar studies were performed from several researchers by using different samples. [7-8-9-10]

This recent work hasadopted to the scattering intensity ratio of 59.54 keV gamma rays that were emitted from ²⁴¹Am radioisotope. Because of the scientific and commercial importance, the Ce and some selected compounds have been selected as a target. The determined values of the effective atomic number and the electron densities of samples have been in good accordance with calculated theoretical values.

2. Theory

The effective atomic number and the electron densities of pure and composite samples have been determined by using scattering intensity ratios of incident radiation in recent years. The scattering intensity ratio measurements were carried out by using the Rayleigh (coherent) and Compton scattered peaks of target samples. The usage of this technique was containing the probabilities of related scattering. In Rayleigh (coherent) scattering event, the incident beam has collided from the target sample with no energy and wavelength change. The Rayleigh (coherent) scattering has been called elastic collision and, in this collision, no energy

change has been observed. The Rayleigh scattering probability has been formulized by the following equation,

$$\frac{d\sigma_R}{d\Omega} = \frac{r_0^2}{2} (1 + \cos^2 \theta) |F(q, Z)|^2 \tag{1}$$

In this equation, the atomic form factor, the transferred momentum to the electron, the classical electron radius and the scattering angle is representing by F(q, Z), $q, r_0 (= 2.8179 \times 10^{-15} m)$, θ respectively.

Compton scattering is a phenomenon where a high energy photon interacts with an electron and emits a low frequency scattered electron and a low energy photon. This situation shown that the Compton scattering experiment is evidence of the particle structure of electromagnetic radiation. Different from the Rayleigh scattering, the Compton scattering cross section was calculated by using Klein-Nishina equation,

$$\left(\frac{d_{\sigma}}{d_{\Omega}}\right)_{KN} = \frac{1}{2}r_0^2 \left(\frac{E}{E_0}\right)^2 \left(\frac{E_0}{E} + \frac{E}{E_0} - \sin^2\theta\right) \tag{2}$$

the scattered and incident photon energies was represented by E and E_0 respectively. In Compton scattering events, the momentum of photon has been transferred to the electron. This transportation value has been affected the incoherent scattering function (S(q,Z)). In some cases, the momentum transportation has been large values and the incoherent scattering function have been equals to the atomic electrons. If otherwise, the transferred momentum to the electron very low, approaches to the zero, the incoherent scattering function becomes zero. So, the calculation of the effective atomic number has been carried out by using the ratio of elastic and inelastic scattering cross sections. The cross-section ratio (R) equation was given below,

$$R = \left(\frac{d\sigma}{d\Omega}\right)_R / \left(\frac{d\sigma}{d\Omega}\right)_{aC} \propto \frac{|F(q,Z)|^2}{S(q,Z)}$$
 (3)

The atomic Compton scattering cross-section parameter was symbolized by using $(d\sigma/d\Omega)_{aC}$ equation. The atomic Compton

scattering has occurred from bound atomic electrons. The atomic Compton scattering cross-section given as,

$$\left(\frac{d\sigma}{d\Omega}\right)_{aC} = S(q, Z)\left(\frac{d\sigma}{d\Omega}\right)_{KN} \tag{4}$$

The calculated value of R has been proportional to the atomic number of investigated materials. The equation parameters F(q, Z) and S(q, Z) that was used in the calculation of Rayleigh to Compton scattering cross-section has been created by Hubbel *et al.*[11].

The weighted atomic percentages of target samples were used in calculation of the Rvalue. The R value has been directly related with the atomic number of target samples and calculated by using following equation,

$$R = \left(\frac{d\sigma}{d\Omega}\right)_R / \left(\frac{d\sigma}{d\Omega}\right)_{aC} \propto \frac{\sum_{j=1}^n a_j^{at} |F(q, Z_j)|^2}{\sum_{j=1}^n a_j^{at} S(q, Z_j)}$$
 (5)

In equation (5), the atomic percentage was represented by a_j^{at} term. The a_i^{at} term has been calculated via following equation,

$$a_j^{at} = \frac{(W_j/A_j)}{\sum_j (W_j/A_j)} (6)$$

where the atomic percentages have been represented by W_j and the atomic mass of the Jth element have been represented A_j respectively.

3. Experimental setup and present measurements

The present experimental study was focused on the effective atomic number determination of pure and some selected Ce compounds. For this purpose, 59,54 keV gamma photons were directed to the samples and scattered photons were detected and analyzed. The effective atomic number determination of rare earth element Ce and selected compounds were widely used in radiation shielding applications, lasers, neutron capturing, medical physics, and many other technological applications.

Target materials scattered the incident gamma photons that were emitted from the source, and also, the scattered photons were detected from the HPGe semiconductor detector. The specific feature of the detector was given in Table 1. Besides, Fig. 1 and Fig. 2 has been represented the schematic structure of the experimental setup used in this work. In this work, we have studied pure Ce and some selected Ce compounds $(CeCl_3, Ce(NO_3)_3, CeF_3,$ $Ce(SO_4)_3, Ce_2O_3).$ Canberra (AccuSpec) Pc-based multichannel analyzer card was used in the processing of the scattered spectrum of samples. Besides, Ortec 472 model amplifier was used for obtaining the best counting rates from the detector. The time constant of the amplifier was set to 6 µs. All these parameters were gained by using the Genie-2000 computer program. The Compton and coherently scattered photons were collected by MCA in 1024 channels. In addition to the collecting process, all these data were analyzed by the Origin 7.5 software program.

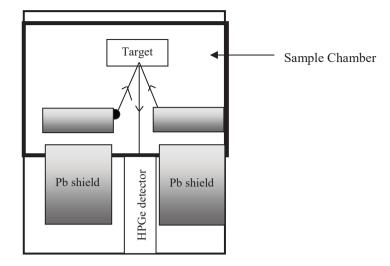


Figure 1.Experimental setup (radius of collimator is 0.53 cm). This figure is only a schematic diagram of experimental setup.

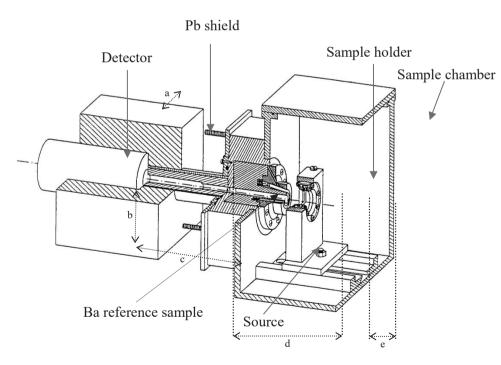


Figure 2. Sample chamber (a=6.5 cm, b=6.3 cm, c=13.5 cm, d=11 cm, e=5 cm).

Table 1. The characteristics specifications of HPGe detector.

Active Diameter	16 mm	
Active Area	200 mm^2	
Sensitive Detector Depth	10 mm	
Distance From Window	5 mm	
Beryllium window 0.12 mm		
thickness		
Supply Voltage	-1500 V	
Resolution (5,9-122 keV)	182-488 eV	

Table 2. The effective atomic number values of Ce and selected samples.

Target	R/C		
Samples	Ratio	Experimental	Theoretical
Ce			5.00E : 01
CeCl ₃	9,34E- 01	5,35E+0	5,80E+01 5,36E+
CeF ₃	1,01E+ 00	5,53E+0 1	5,56E+ 01
Ce(NO ₃) ₃	1,12E+ 00	5,89E+0	5,88E+
$Ce_2(SO_4)_3$	8,41E- 01	5,19E+0 1	5,18E+ 01
Ce ₂ O ₃	1,05E+ 00	5,66E+0	5,68E+ 01

4. Conclusion

Effective atomic number values obtained by the present study, other experimental values available in the literature, and density ratios of Rayleigh scattering to Compton scattering are given in Table 2. In addition, the theoretical effective atomic numbers of the samples used in the experiment are given in Table 2. Experimental error factors and

uncertainties arising from different factors caused errors in the evaluation of characteristic signals at 1.64-3.51%, target sample mass thicknesses at 1.37-3.18%, photopic efficiency at 1.10-2.04%, and statistical errors less than 1%. It has been determined that all parameters affecting the experimental values have values between 2.68% and 5.40%. Molecular and solid-state effects were not taken into account in the calculation of theoretical values. Also, the XCOM [12] database ignores the interaction between atoms, the coulomb interaction between photoelectron, and positively charged ions. In addition, the XCOM database does not calculate energy absorption components. This situation causes an inconsistency between theoretical and experimental values.

It is seen that there is a good harmony between experimental values and theoretical values. This has shown that the present experimental technique is a usable method for effective atomic number calculation. It is clear that more experimental data must be obtained. Especially, more precise results can be obtained with experiments with compounds having different scattering angles and chemical compositions.

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EVALUATION OF MAGNETO-RESISTIVITY PERFORMANCES AND FLUX PINNING CENTERS WITH VANADIUM ADDITION IN BI-2223 MAIN MATRIX



140 · Mustafa Burak Türköz , Muhammed Oz, Tahsin Turgay, Gürcan Yıldırım

1. INTRODUCTION

Materials science takes an important place in daily life of people. The history of science points out that there appear a number of turning points for the materials sciences and technology application fields. Especially, in the year of 1911 H. Kamerlingh Onnes discovered the superconducting phenomenon in the mercury heavy-metal element at temperature of 4.2 K [1]. Those years made a lot of positive effects in the scientific world, and the superconductivity has widely attracted the remarkable interest of research community. In the following years, the researchers have discovered new superconducting phases and materials such as metals, semi-metals, metal or semi-metal-based compounds, alloys, carbon-containing compounds, materials with heavy fermions, organic compounds, A-15 inorganic samples, rare-earth borocarbide materials, rutheno-cuprate samples, compounds with chevrel phase, chalcogens and related compounds, silicon-containing materials, pyrochlore oxides, magnesium diboride with two-gap superconductivity and distorted oxygen deficient multi-layered perovskite structures (Tl-, Bi- and Hg-containing high-Tc superconducting compounds) [2]. Thus, the superconducting compounds have a chance to take much more places in the application-oriented materials science, innovative energy infrastructure, advanced engineering, electro-optics, medicine, industry, and different energy sectors [3–5]. This is because, the superconducting compounds exhibit extremely larger magnetic field and especially current carrying capability, much smaller power consumption, energy losses and dissipations in the crystal system [6–8]. In particular, among the parents of cuprate-layered perovskite superconducting materials the bismuthbased copper oxides show relatively high critical temperature (up to about 110 K), critical current density (≈10⁷A/cm²) and upper critical magnetic field (≈100 T). Moreover, the Bi-based copper oxides are easily prepared with the simple chemical composition and production procedure. Similarly, the easy and fast phase formations, easy availability of chemicals, low-cost efficiency, and environmental benefits can be mentioned for the other superior properties for the preparation of Bicontaining superconducting compounds. At the same time, the bi-based superconducting parents phase has great attracted attention of many researchers as a consequence of their highly thermodynamic stability, compositional and oxygen stability, (in the atmospheric pressure without any gas atmospheres/high-pressure environments) and resistant towards to the water/humid atmosphere under rather higher sintering temperature values [9–12].

As for the crystal structure of the materials, all the family members are crystallized in perovskite structures within the homologous series of $Bi_2Sr_2Ca_{n-1}Cu_nO_{4+2n+x}$ and composed of the active and effective shortrange-ordered antiferromagnetic Cu-O₂ sheets of adjacent superconductive layers [13–15]. The parents consist of three different participants depending on the conducting Cu-O2 plane numbers (observed by number of n) in the unit cell: Bi-2201 (very low phase), Bi-2212 (low phase) and Bi-2223 (high phase). Accordingly, in the tetragonal structure of superconducting systems, the Ca-site forms a layer within the interior of Cu-O₂ layers for the low and high phase whereas no Ca layer takes place in the very low phase [16]. Likewise, the n value is directly related to the number of Cu-O₂ planes in the crystal system. To illustrate, the Bi-2201 superconducting phase possesses a Cu-O₂ plane in the structure while the low phase has two Cu-O₂ planes as against three Cu-O₂ planes for the Bi-2223 phases [17, 18]. Besides, the Cu atom takes the different positions in the coordination of superconducting system. Throughout the Bi-2201 superconducting phase the octahedral coordination formation of a Cu atom with the oxygen atoms is observed in the crystal structure. On the other hand, the five oxygen atoms surround a Cu atom in a pyramidal arrangement for the Bi-2212 superconducting phase while in case of Bi-2223 phase the Cu atoms obtain two different coordination types with regard to the oxygen atoms: (I) one Cu atom is bonded by four oxygen atoms to configure in the square planar molecular geometry; and (II) another Cu atom is coordinated by five oxygen atoms to be arranged in a pyramidal molecular geometry [9, 17–19]. Moreover, it is directly received that the number of n plays an important role on the fundamental characteristics such as the normal-state resistivity, residual resistivities, offset/onset critical transition temperatures, degree of broadening parameters and flux pinning ability of Bi-based superconducting parents [20].

It is to be strongly stressed here that the parents have some problematic faults that prevent seriously the materials to use in some specific application fields of new industrial and technological application fields [21, 22]. At the beginning of these faults, the structural layers, grain interaction coupling problems, grain alignment distributions, operating temperature range problems, low charge carrier (hole)

densities, brittleness nature, extremely sensitivity towards the external magnetic field and current, tremendously large penetration depths and short coherence lengths can be mentioned. In the present study, we try to improve the restrictions given above with the vanadium dopant in bulk Bi-2223 phase (for *n* number of 3) so that we can enhance the usage areas in newly, novel and feasible application fields (especially future heavyindustrial technology, application-oriented materials science, advanced engineering, electro-optics, innovative energy infrastructure and energyrelated sectors) of Bi-2223 superconducting materials. On this basis, we investigate the fundamental differentiations in the magneto-resistivity performance and related scientific quantities as regards the active and dynamic electron-phonon couplings throughout the intra-grain regions, flux pinning abilities for two-dimensional pancake vortices, energy barriers for vortices motions, recoupling line-like mechanism, lattice period for the vortices, interplane interaction energy, movement of vortices to neighboring state length throughout pinning regions, dynamic nucleation centers in the superconducting system with the addition of different vanadium concentration (between 0.0 and 0.3 molar ratio) in the Bi-2223 superconducting crystal lattice with the assistant of magnetic field measurements exerted at external magnetic fields varying from zero T to five T.

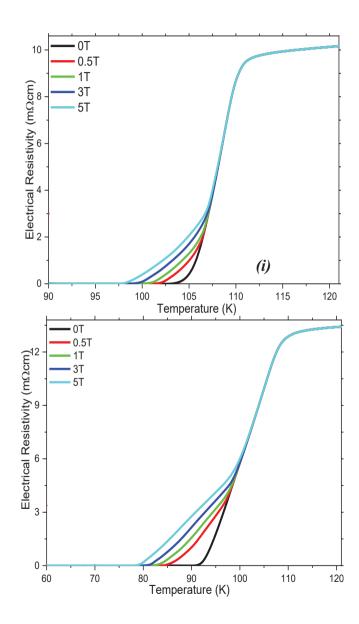
2. EXPERIMENTAL PROCEDURES FOR VANADIUM ADDED SOLID Bi-2223 CERAMIC MATERIALS

This present study is a continuation of methodical characterization searches and focuses sensitively on the critical influence of V addition into the high phase superconducting ceramic compounds on the lattice period for the vortices, energy barriers for vortices motions, formation of the active and dynamic electron-phonon couplings throughout the intragrain regions, flux pinning abilities for two-dimensional pancake vortices, interplane interaction energy for Josephson coupling lengths, pinning center sizes, recoupling line-like mechanism, dynamic nucleation centers in the superconducting system with the aid of magneto transport measurements performed the external magnetic fields intervals 0–5T. One can see some fundamental characteristic properties, viz. the temperature-dependent resistivity values at room temperature, dc electrical resistivity at the value of 115 K, residual resistivity, residual resistivity ratio, ρ_{norm} and $\Delta \rho$ and superconducting features of onset-offset

critical transition temperature and broadening transition width parameters in Ref. [23-26]. Besides, we examined the powder X-ray diffraction results and illustrated differentiation of texturing, phase volume fractions, crystal structure qualities, lattice constant parameters (a and c-axis length), grain alignment distributions, grain size parameters, coupling problems for grain boundary and connectivity quality intra and inter-grain regions depending on the vanadium concentration level. Similarly, we determined the coherence and bulk genuine temperatures related to the critical characteristic temperatures of solid Bi_{1.8}Sr_{2.0}Ca_{2.2}Cu_{3.0}V_xO_y within $0.00 \le x \le 0.30$ molar ratio in the scientific paper published in Ref. [23,26]. It was obtained that the presence of vanadium in the Bi-2223 superconducting crystal structure damages dramatically the general characteristic properties mentioned above. Furthermore, we endeavor to find the possible maximum critical transition (onset and offset) temperatures with empirical model including defects and disorders for bulk Bi-2223 superconducting crystal system added with vanadium ions. One can see the production of Bi-2223 superconductors and vanadium addition procedure, also. However, we remind roughly the general production procedure of materials in this part of paper. All the superconducting materials including pure and vanadium added bulk Bi-2223 superconducting materials with the chemical formula of Bi_{1.8}Sr_{2.0}Ca_{2.2}Cu_{3.0}V_xO_y are experimentally prepared by the standard solid-state reaction method [27]. On this basis, the powder of chemicals (oxide and carbonate) including the high purity materials (%99.99) of Bi₂O₃, CuO, SrCO₃, CaCO₃ and V₂O₅ purchased from Alfa Aesar exclusive distributor is used for the production of main superconducting materials. Firstly, we weigh the chemicals with respect to the stoichiometric ratios of $Bi_{1.8}Sr_{2.0}Ca_{2.2}Cu_{3.0}V_xO_y$ (0.00 $\le x \le 0.30$) by use of the electronic balance and mix all the chemicals using the grinding machine for nine hours in the medium of air at the constant temperature of room temperature. We, hence, obtain much more homogenous chemical mixture within the desired particle sizes of powders. For the calcination procedure, the annealing ambient combination is preferred to be 800 °C temperature and 36 hours annealing period in the furnace under the atmospheric air pressure condition. Both the heating and cooling rates are adjusted as 5 °C/min. After the calcination process, we remove the homogenous powder of chemicals (the last color of powders turns into the blackish) from the furnace. We ground the blackish powder for 30 minutes in the medium of air in an agate by a pestle. The last (milled) powder of chemicals is exposed to pelletize into the rectangular bars with the volume of 1.5x0.5x0.2 cm³ at the load strength of 300 MPa in air atmosphere conditions. The different vanadium added bulk samples are annealed at 850 °C for 24 h. All the materials exposed to different vanadium addition level such as x=0.01, 0.03, 0.05, 0.07, 0.10 and 0.30 will thereafter be shown to be pure, V-01, V-03, V-05, V-07, V-1 and V-3, respectively. In the current work the magneto transport measurements are done in the cryostat system for 5 different applied magnetic fields varying from 0 to 5T. The experimental signals are gathered in the temperatures between 5 K and 130 K at 5mA current.

3. RESULTS AND DISCUSSION

In the current work, the influence of vanadium addition within the different molar ratios of 0.0-0.3 on the fundamental magnetic performance features, viz. the active and dynamic electron-phonon couplings throughout the intra-grain regions, flux pinning abilities for two-dimensional pancake vortices, energy barriers for vortices motions, recoupling line-like mechanism, lattice period for the vortices, interplane interaction energy, movement of vortices to neighboring state length throughout pinning regions, dynamic nucleation centers in the superconducting system, recoupling line-like mechanism, pinning center sizes, interplane interaction energy for Josephson coupling lengths of Bi-2223 system is investigated by the magneto transport measurement tests performed in various external magnetic fields (abbreviated as $\mu_0 H$) differing between 0T and 5T. We give all the measurement magneto resistivity results for the pure and V-doped bulk Bi-2223 samples in the whole vanadium dopant concentration ranges in Figs. 1-3. Prior to the numerical values, we will discuss the general trend in the magneto resistivity curves with the vanadium addition. We give the curves in three different figures since the vanadium added Bi-2223 superconducting compounds exhibit three different mechanism because of their own the crystallinity problems and the induced the pair-breaking mechanism (to be discussed in the following section in detail) in the superconducting system. It is visible form Fig. 1 that the superconductivity for the pure, V-01, V-03, V-05 superconducting compounds cannot be destroyed totally up to the maximum external magnetic field strength of 5 T. However, from higher vanadium concentration value of x=0.05 onwards we could not determine the offset critical transition temperature values for the bulk V-07, V-1 and V-3 ceramic samples (See Fig. 2 and 3). The V-07 superconducting compound can carry the maximum 3 T applied magnetic field whereas the V-1 and V-3 sample can carry the magnetic field strength of only 1 T. It is clear that the excess vanadium causes the impurity in the superconducting system, thus, the Bi-2223 material with excess vanadium dopant cannot carry much more magnetic field strength.



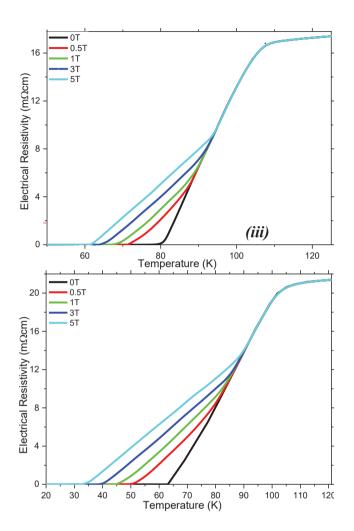


Figure 1 Variation of magneto resistivity curves against temperature at various applied field strengths for (i) virgin, (ii) V-01, (iii) V-03 and (iv) V-05 ceramic compounds.

In more detail, it is apparent from the curves that the magneto resistivity performance of Bi-2223 superconducting system is found to depend strongly on both the external magnetic field strength and vanadium addition content level. As well known that under the applied magnetic field two main mechanisms lead to damage the superconductivity phenomenon. The first of them is Pauli paramagnetic effect or spin effect due to the appearance of Zeeman splitting for the

spin-singlet pairing under the applied magnetic field strength. Diamagnetic effect or orbital limit effect is the second reason and is related to the increment of kinetic energy due to the fact that the applied magnetic field is excluded by the orbital current. Among them, the diamagnetic effect predominantly takes place at relatively higher temperature loser to the critical transition temperature. Thus, the pairbreaking mechanism (meaning: there appears an interaction between the conduction electron spins and quantum spin number (S) of vanadium impurities) is realized by the diamagnetic effect under the applied magnetic field strength. On the other hand, the Pauli paramagnetic effect is encountered at the low temperature regime and higher applied magnetic field strength. Accordingly, the superconducting characteristic features of Bi-based superconducting ceramic materials can seriously be damaged under the applications of current density and external magnetic field. Moreover, with the assistant of magneto resistivity graphics gathered in Fig. 1 we define the fundamental superconducting characteristic parameters of offset and onset (T_c^{offset}) and T_c^{onset} critical transition temperature values. It is emphasized here that the value between the offset and onset critical transition temperatures is known to be the broadening transition width (namely, it can be shortened to be $\Delta T_c = T_c^{onset} - T_c^{offset}$. We infer every parameter (T_c^{offset} , T_c^{onset} and related ΔT_c) from the magneto resistivity curves and display in Table 1 for emphasizing the influence of vanadium inclusions in the Bi-2223 crystal system. It is found that both the $T_c^{\it offset}$ and $T_c^{\it onset}$ parameters are dramatically affected (especially former one) by ascending amount of vanadium impurity concentration in the host superconducting crystal structure and external magnetic fields applied. It is to be stressed here that the T_c^{offset} parameter is connected to inter-granular component in the crystal structure, and is affected by the density of states distribution at Fermi energy level in the superconducting system. As to the T_c^{onset} parameter for the superconducting materials, the parameter is related to electrical transition of intra granular regions throughout the crystal structure into the superconducting state. That is why; the T_c^{onset} value is determined by the capability of active and dynamic electron-phonon couplings in the superconducting system [28–30]. According to measurement results provided, the reduction of critical transition

temperatures with ascending the vanadium addition in the crystal lattice of superconducting system results dominantly from the dramatic enhancement in the crystallinity problems arranged as porosity, defects and related flaws in the Bi-2223 ceramic compounds [31, 32]. Another possible mechanism being in charge of the degradation of T_c^{onset} and T_c^{offset} values is related to the reduction of localization phenomenon for the density of states distribution at Fermi energy level in the superconducting system [33, 34].

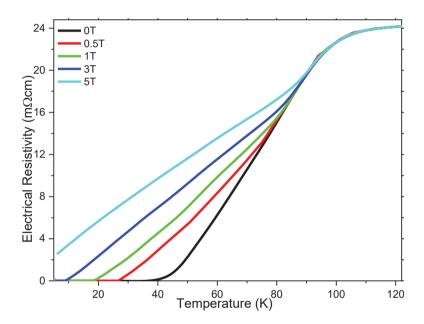


Figure 2 Magneto resistivity performances for V-07 superconducting material under applied magnetic fields.

In this regard, we can easily claim that the increment of V-ions addition in the Bi-2223 system causes to degrade of flux pinning abilities for 2D pancake vortex and dynamic nucleation centers in the superconducting system. In other words, the presence of vanadium impurities leads to truncate the active and dynamic electron-phonon couplings throughout the intra-grain regions, energy barriers for vortices motions, recoupling line-like mechanism, lattice period for the vortices, interplane interaction energy, movement of vortices to neighboring state length throughout pinning regions, interplane interaction energy for

Josephson coupling lengths and pinning center sizes in bulk Bi-2223 superconductor. Similarly, the augmentation level of V-ions concentration in the system results in much more induced recoupling line-like mechanism in the Bi-2223 crystal lattices. Moreover, based on the $T_c^{\it offset}$ and $T_c^{\it onset}$ values in Table 1 it is seen that the applied magnetic field is obtained to much more affect the $T_c^{\it offset}$ parameters as compared to the $T_c^{\it onset}$ parameters.

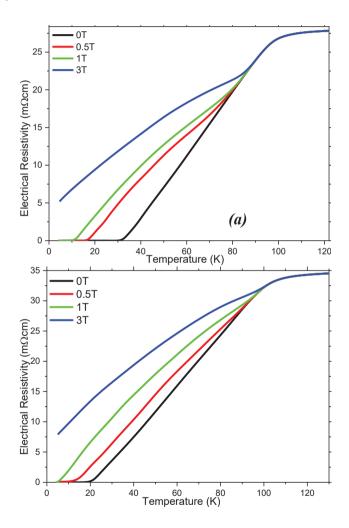


Figure 3 Differentiation in magneto resistivity curves as a function of temperature at various external magnetic fields for bulk V-1 and V-3 superconducting materials.

As for the numerical values for the $T_c^{\textit{offset}}$ and $T_c^{\textit{onset}}$ values, the Toffset parameters are recorded as about 103.17 K, 91.11 K, 79.42 K, 56.68 K, 38.07 K, 30.29 K and 18.38 K for the virgin sample, V-01, V-03, V-05, V-07, V-1 and V-3 samples, respectively, without any external field. With the increment in the applied magnetic field, it is, of course, that the T_c^{offset} values degrade rapidly. In this respect, the T_c^{offset} value for the pristine compound decreases towards to 98.06 K temperature under the maximum applied field of 5T. Similarly, the V-01 superconducting sample exhibits the T_c^{offset} value of 78.93 K at 5 T applied field. The T_{c}^{offset} values (deduced from Figure 1) are found to be about 61.43 K and 33.36 K for the V-03 and V-05 ceramic compounds, respectively. As for the second group (the bulk V-07 superconducting material), we only determine the T_c^{offset} values for the maximum external field of 3 T. On this basis, the material presents the T_c^{offset} value of 8.52 K. Finally, the last superconducting materials including the solid V-1 and V-3 superconductors show the $T_c^{\textit{offset}}$ values of 10.57 K and about 4.91 K at only 1 T external magnetic field strength. In this regard, it is to be emphasized here that the increase in V-ions impurity level degrades harshly the T_c^{offset} values. Namely the sensitivity (strength) to applied magnetic field is found to tend to enhance regularly with the increase in the vanadium addition level. Consequently, the bulk V-3 sample(highest dopant level) displays the most response to the applied field. This evidence is verified with the differentiation in the broadening transition width (ΔT_c) values, also. In more detail, the smallest value (assigned as the best sample) of 5.11 K is attributed to pristine material, on the other hand the bulk V-05 superconducting ceramic compound with the ΔT_c value of 26.32 K is received to be the worst material for the first group. It is to be mentioned here that we could not determine the ΔT_c value for 5 T (in fact for 3 T) applied magnetic field.

Table 1 Experimental findings depending on magneto resistivity measurements for pure and vanadium added ceramic materials under magnetic fields

Samples	T _c onset (K)	T _c offset (K)					•
		ОТ	0.5T	1T	3T	5T	ΔT_c
Pure	110.92	103.17	101.85	100.87	99.60	98.06	5.11
V-01	109.48	91.11	84.67	82.81	81.15	78.93	12.18
V-03	106.95	79.42	71.33	68.04	64.15	61.43	17.99
V-05	103.94	59.68	50.27	44.64	39.43	33.36	26.32
V-07	99.21	38.07	26.72	17.81	8.52		
V-1	98.98	30.29	16.39	10.57			
V-3	97.45	18.38	11.96	4.91			

As for the T_c^{onset} parameters, the compounds (pure, V-01, V-03, V-05, V-07, V-1 and V-3) present the values of about 110.92 K, 109.48 K, 106.95 K, 103.94 K, 99.21 K, 98.98 K and 97.45 K respectively (see Table 1). Based on the values observed, the T_c^{onset} parameters are noticed to reduce systematically with increasing the vanadium impurity addition level because of the augmented defects and disorders in the bulk Bi-2223 system. Conversely, the magnetic field cannot change the T_c^{onset} parameters for all the samples since the external field applied cannot harm on the capability of active and dynamic electron-phonon couplings in the Bi-2223 superconducting system

4. CONCLUSION

In the present study, we scrutinize the crucial variation of fundamental magneto-resistivity performance parameters and ability/formation of effective and active flux pinning centers in the Bi-2223 crystal system by use of the magneto-transport measurements conducted in the applied magnetic fields intervals 0–5T with the

vanadium concentration level (Bi-2223+V_x, where x is between 0.00 and 0.30). Besides, the influence of V-ions on the grain boundary coupling problems, ability/strength of vortex lattice period and elasticity of thermal fluxons in the correlated two-dimension (2D) pancake vortices. The solid compounds are prepared by the standard solid-state reaction route in the room temperature conditions at the atmospheric pressure. The magnetoresistivity curves are taken from the temperature intervals 15 K-125 K at a closed-cycle cryogenic refrigeration system. Besides, the discussions are determined from the fundamental quantitates as regards the offset (T_c^{offset}), onset (T_c^{onset}) critical transition temperature and broadening transition width $(\Delta T_c = T_c^{onset} - T_c^{offset})$ parameters. Based on the experimental measurement findings deduced from the magneto-transport curves, the presence of V-ions concentration in solid Bi-2223 crystal structure significantly diminishes the magneto-resistivity performances because of the rapid decrement (augmentation) in the formation of dynamic flux pinning centers (random distribution of artificial pinning centers) in the crystal lattice and the ability/strength of vortex lattice period and elasticity (the grain boundary coupling problems). Similarly, the vanadium concentration leads to degrade the energy barriers of flux lines between the copper-oxide consecutively stacked layers for the pinning ability so that the flux lines can easily jump to the neighboring states due to the appearance of recoupling linelike nature in the Cu-O₂ consecutively stacked layers. Finally, we examine the change in the hole localization problems for the density of states distribution at Fermi energy level in the Bi-2223 superconducting system. Numerically, the T_c^{onset} parameters are found to be about 110.92 K, 109.48 K, 106.95 K, 103.94 K, 99.21 K, 98.98 K and 97.45 K for the pure, V-01, V-03, V-05, V-07, V-1 and V-3 superconducting compounds, respectively. As for the $T_c^{\textit{offset}}$ values, the pure sample has the maximum offset critical transition temperature of 103.17 K whereas the minimum T_c^{offset} value is observed for the V-3 superconducting material. Under the maximum external magnetic field of 5 T the T_c^{offset} degrades towards to the value of 98.06 K, 78.93 K, 61.43 K and 33.36 K for the pure, V-01, V-03 and V-05 ceramic compounds (the first group), respectively. However, we could not define the T_c^{offset} value at 5 T applied field for the other bulk materials prepared in this work. In this regard, in the first group samples the pure sample

(best material) possesses the smallest ΔT_c value of 5.11 K. All in all, the pure sample is recorded as the least response to applied field. On the other hand, the V-3 compound is found to be rather more sensitive to the applied field.

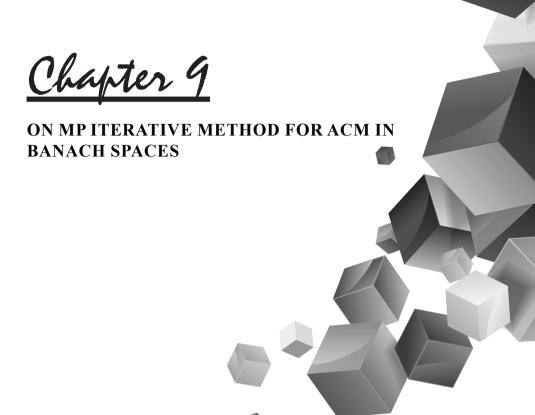
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1 Introduction and Preliminaries

Let C be a nonempty closed convex subset of a Banach space X and S be a self map of C. Suppose $s_0 \in C$ and $s_{n+1} = f(S, s_n)$ defines an iterative procedure which yields a sequence of points $\{s_n\}_{n=0}^{\infty}$ in C. For example, $s_{n+1} = f(S, s_n) = Ss_n$. Suppose $\{y_n\}_{n=0}^{\infty} \subset C$ and $\{\epsilon_n\}_{n=0}^{\infty} \subset \mathbb{R}^+$ are two sequences given by $\epsilon_n = \|y_{n+1} - f(S, y_n)\|$ and F(S) is the set of fixed points of S such that $F(S) = \{s^* \in C: Ss^* = s^*\}$ with $\{s_n\}_{n=0}^{\infty}$ converges strongly to $s^* \in F(S)$. If $\lim_{n\to\infty} \epsilon_n = 0$ implies that $\lim_{n\to\infty} y_n = s^*$, then the iterative method $\{s_n\}_{n=0}^{\infty}$ is said to be S-stable or stable with respect to S

In the last fifty years, many researchers have demonstrated an effort to identify a new iterative process and they have been successful in this area. The most famous of these iterative processes are defined as follows:

In 1890, the Picard iterative process [1] was defined by Picard as follows:

$$p_{n+1} = Sp_n$$
, $n \in \mathbb{N}$ (1) and is used to approximate fixed point of contraction mappings satisfying the following condition

$$d(Sx, Sy) \le \gamma d(x, y), \forall x, y \in B \text{and} \gamma \in [0, 1). \tag{2}$$

In 1953, the Mann iterative process [2] was defined by Mann as follows:

$$w_{n+1} = (1 - \alpha_n)w_n + \alpha_n Sw_n, for all n \in \mathbb{N} and \alpha_n \in [0,1]. \tag{3}$$

In 1974, the Ishikawa iterative process [3] was defined by Ishikawa as follows:

$$\begin{cases} I_1 \in C \\ I_{n+1} = (1 - a_n)I_n + a_n S h_n \\ h_n = (1 - b_n)I_n + b_n S I_n, \end{cases}$$
(4)

where, $\{a_n\}_{n\in\mathbb{N}}$ and $\{b_n\}_{n\in\mathbb{N}}\subset [0,1]$.

In 2013, the Picard-Mann iterative process [4] defined by Khan, as follows:

$$\begin{cases} k_1 \in C \\ k_{n+1} = St_n \\ t_n = (1 - c_n)k_n + c_nSk_n, \ for all n \in \mathbb{N}, \end{cases}$$
 where, $\{c_n\}_{n \in \mathbb{N}} \in (0,1).$ (5)

When defining an iteration process, two important criteria should be considered. These criteria are speed and simplicity. For this reason, the Picard-Mann iteration process is important. Considering the Picard-Mann and some others iterative procedures in the current literature, we introduce a new iterative procedure is called Mann-Picard iterative process. Our iterative process defined by $\{\pi_n\}_{n\in\mathbb{N}}$ as follows:

$$\begin{cases} \pi_1 \in C \\ \pi_{n+1} = (1 - \varphi_n)\rho_n + \varphi_n S \rho_n \\ \rho_n = S \pi_n, \quad \forall n \in \mathbb{N}, \end{cases}$$
 (6)

where $\{\varphi_n\}_{n\in\mathbb{N}}\in[0,1]$.

In the sequel, we will prove that iterative process (6) converges to the unique fixed point of ACM satisfying (8). We also prove that iterative process (1.6) is equivalent to Mann iterative process and faster than all Picard, Mann and Ishikawa iterative processes when all converge to the fixed point of ACM (8). Finally, using iterative process (6), we prove a data dependence result for the fixed of point of ACM satisfying (8).

To prove our main results, we need to the following Theorems, Lemmas and Definitions.

Definition 1.1 [5] If the operator $T: C \to C$ is almost-contraction mapping (ACM), then, for all $x, y \in B$, the following conditions are performed

$$(i)L \ge 0 \text{ and } \delta \in (0,1),$$

 $(ii)\|Sx - Sy\| \le \delta \|x - y\| + L\|y - Sx\|.$ (7)

Theorem 1.2 [5] Let (X, d) be a complete metric space and $T: X \to X$ be an ACM for which there exist $\delta \in (0,1)$ and some $L_1 \ge 0$ such that

$$||Tx - Ty|| \le \delta ||x - y|| + L_1 ||x - Tx||. \tag{8}$$

Then, T has a unique fixed point.

Theorem 1.3 [5] Let X be a Banach space and C be a nonempty closed convex subset of X. Assume that the map $S: C \to C$ is satisfies all of the conditions of ACM (8). Then $F(S) \neq \emptyset$.

Definition 1.4 Let $\{\omega_n\}_{n\in\mathbb{N}}$ and $\{\varpi_n\}_{n\in\mathbb{N}}$ be two sequences and $\{\omega_n\}_{n\in\mathbb{N}}$ and $\{\varpi_n\}_{n\in\mathbb{N}}$ be converging to the same fixed point p of S. Suppose that

$$\lim_{n\to\infty}\frac{\|\omega_n-s^*\|}{\|\varpi_n-s^*\|'}$$

then, it said that $\{\omega_n\}_{n\in\mathbb{N}}$ converges faster than $\{\varpi_n\}_{n\in\mathbb{N}}$ to s^* .

Definition 1.5 [5] Let $\{\omega_n\}_{n\in\mathbb{N}}$ and $\{\varpi_n\}_{n\in\mathbb{N}}$ be two sequences both converging to the same fixed point s^* , the error estimates

$$\|\omega_n - s^*\| \le a_n, \forall n \in \mathbb{N}$$

 $\|\varpi_n - s^*\| \le b_n, \forall n \in \mathbb{N}$

are valiable where the sequences $\{a_n\}_{n\in\mathbb{N}}$ and $\{b_n\}_{n\in\mathbb{N}}$ converge to zero. Suppose that

$$\lim_{n\to\infty}\frac{a_n}{b_n}=p$$

- (1) If p = 0, then $\{a_n\}_{n \in \mathbb{N}}$ converges faster than $\{b_n\}_{n \in \mathbb{N}}$.
- (2) If $0 , then <math>\{a_n\}_{n \in \mathbb{N}}$ and $\{b_n\}_{n \in \mathbb{N}}$ have equal

convergence

(3) If $p=\infty$, then $\{b_n\}_{n\in\mathbb{N}}$ converges faster than $\{a_n\}_{n\in\mathbb{N}}$. So, if $\{a_n\}_{n\in\mathbb{N}}$ converges faster than $\{b_n\}_{n\in\mathbb{N}}$ then $\{\omega_n\}_{n\in\mathbb{N}}$ converges faster than $\{\varpi_n\}_{n\in\mathbb{N}}$.

Definition 1.6 [17] Let S, $S^o: C \to C$ be two operators satisfying the ACM. $\forall p \in C$ and $\epsilon > 0$ a fixed real number, if the following condition is performed such that

$$||S - S^o|| \le \epsilon$$
,

then the operator S is said to be an approximate operator of S^o .

Lemma 1.7 [18] Let $\{s_n\}_{n\in\mathbb{N}}$ and $\{\tau_n\}_{n\in\mathbb{N}}$ be two nonnegative sequences satisfying inequality:

 $s_{n+1} \leq (1-\varsigma_n)s_n + \varsigma_n \tau_n,$ where $\forall n \in \mathbb{N}$, $\tau_n \in (0,1)$ and $\sum_{n=1}^{\infty} \varsigma_n = \infty$. Then the following inequality holds

$$0 \leq \limsup_{n \to \infty} s_n \leq \limsup_{n \to \infty} \tau_n,$$

it is implies that $\lim_{n\to\infty} s_n = 0$.

Lemma 1.8 [19] Let $\{s_n\}_{n\in\mathbb{N}}$ and $\{\tau_n\}_{n\in\mathbb{N}}$ be two nonnegative sequences satisfying inequality:

$$s_{n+1} \le (1 - \varsigma_n) s_n + \tau_n,$$

where $\forall n \in \mathbb{N}$, $\tau_n \in (0,1)$, $\sum_{n=1}^{\infty} \varsigma_n = \infty$ and $\lim_{n \to \infty} \frac{\tau_n}{\varsigma_n} = 0$. Then $\lim_{n \to \infty} s_n = 0$.

2 Main results

Theorem 2.1 Let $S: C \to C$ be an operator satisfying the condition of ACM (8) with $F(S) \neq \emptyset$ and the sequence $\{\pi_n\}_{n \in \mathbb{N}}$ defined by (6) with $\{\varphi_n\}_{n \in \mathbb{N}} \in [0,1]$ such that $\sum_{n=1}^{\infty} \varphi_n = \infty$. Then the sequence $\{\pi_n\}_{n \in \mathbb{N}}$ converges to the unique fixed point s^* of S.

Proof. From Theorems 1.2 and 1.3, we have uniqueness and existence of fixed point $s^* \in F(S)$.

By (6) and (8), we obtain

$$\begin{split} \|\pi_{n+1} - s^*\| &= \|(1 - \varphi_n)\rho_n + \varphi_n S \rho_n - s^*\| \\ &\leq (1 - \varphi_n)\|\rho_n - s^*\| + \varphi_n \|S \rho_n - s^*\| \\ &\leq (1 - \varphi_n)\|\rho_n - s^*\| + \varphi_n \delta \|\rho_n - s^*\| + \\ \varphi_n L\|s^* - S s^*\| & (9) \\ &= (1 - \varphi_n (1 - \delta))\|\rho_n - s^*\| + \varphi_n L\|s^* - S s^*\|. \end{split}$$

Again

$$\|\rho_n - s^*\| = \|S\pi_n - s^*\|$$

$$\leq \delta \|\pi_n - s^*\| + L\|s^* - Ss^*\|.$$

Substituting (10) in (9) yields that

$$\begin{split} \|\pi_{n+1} - s^*\| &\leq \delta \big(1 - \varphi_n (1 - \delta)\big) \|\rho_n - s^*\| + \\ \big(1 - \varphi_n (1 - \delta)\big) L \|s^* - Ss^*\| + \varphi_n L \|s^* - Ss^*\| \\ &\leq \big(1 - \varphi_n (1 - \delta)\big) \|\rho_n - s^*\| + (1 + \varphi_n \delta) L \|s^* - Ss^*\|. \end{split}$$

Since $||s^* - Ss^*|| = 0$, we have

$$\|\pi_{n+1} - s^*\| \le (1 - \varphi_n(1 - \delta)) \|\rho_n - s^*\|.$$

By induction assumption, we obtain

$$\begin{split} &\|\pi_{n+1} - s^*\| \leq \left(1 - \varphi_n(1 - \delta)\right) \|\rho_n - s^*\| \\ &\|\pi_n - s^*\| \leq \left(1 - \varphi_{n-1}(1 - \delta)\right) \|\rho_{n-1} - s^*\| \\ &\|\pi_{n-1} - s^*\| \leq \left(1 - \varphi_{n-2}(1 - \delta)\right) \|\rho_{n-2} - s^*\| \\ &\vdots \\ &\|\pi_1 - s^*\| \leq \left(1 - \varphi_0(1 - \delta)\right) \|\rho_0 - s^*\|, \text{forall} n \in \mathbb{N}. \end{split}$$

Hence

$$\begin{split} & \|\pi_{n+1} - s^*\| \le_{i=0}^n \left(1 - \varphi_i(1 - \delta)\right) \|\rho_0 - s^*\| \\ & \le \|\rho_0 - s^*\| e^{(-(1 - \delta)\sum_{n=1}^\infty \varphi_n)}. \end{split}$$

Since $\delta \in (0,1)$ and $\sum_{n=1}^{\infty} \varphi_n$, we have

$$\lim_{n\to\infty}||\pi_{n+1}-s^*||=0,$$

and it implies that $\lim_{n\to\infty} \pi_n = s^*$.

Theorem 2.2 Let $S: C \to C$ be an operator satisfying the condition of ACM (8) with $F(S) \neq \emptyset$ and two sequences $\{\pi_n\}_{n \in \mathbb{N}}$ and $\{w_n\}_{n \in \mathbb{N}}$ defined by (6) and (3) (respectively) with $\{\varphi_n\}_{n \in \mathbb{N}} \in [0,1]$ such that $\sum_{n=1}^{\infty} \varphi_n = \infty$. Also, if the initial values are the same, then $\{\pi_n\}_{n \in \mathbb{N}}$ and $\{w_n\}_{n \in \mathbb{N}}$ are equivalent when they convergence to the fixed point s^* of S.

Proof. Under assumption $\pi_n \to s^*$ as $n \to \infty$, we shall show that $w_n \to s^*$ as $n \to \infty$.

Using (1), (6) and (3), we obtain

$$\begin{split} \|\pi_{n+1} - w_{n+1}\| &= \|(1 - \varphi_n)\rho_n + \varphi_n S \rho_n - \\ (1 - \varphi_n)w_n + \varphi_n S w_n\| &\leq (1 - \varphi_n)\|\rho_n - w_n\| + \varphi_n \|S \rho_n - S w_n\| \\ &\leq (1 - \varphi_n)\|\rho_n - w_n\| + \varphi_n \delta \|\rho_n - w_n\| + \\ \varphi_n L \|\rho_n - S \rho_n\| & \end{split}$$

$$\leq (1 - \varphi_n(1 - \delta)) \|\rho_n - w_n\| + + \varphi_n L \|\rho_n - S\rho_n\|.$$

Again, from (6), we obtain

$$\|\rho_n - w_n\| = \|S\pi_n - w_n\|$$

 $\leq \|S\pi_n - \pi_n\| + \|\pi_n - w_n\|$

Substituting (12) in (11), we have

$$\begin{split} \|\pi_{n+1} - w_{n+1}\| &\leq \left(1 - \varphi_n(1 - \delta)\right) \|\pi_n - w_n\| + \|\pi_n - S\pi_n\| \\ &+ \varphi_n L \|\rho_n - S\rho_n\| \\ &= \left(1 - \varphi_n(1 - \delta)\right) \|\pi_n - w_n\| + \left(1 - \varphi_n(1 - \delta)\right) \|\pi_n - S\pi_n\| \\ &+ L\varphi_n \|\rho_n - S\rho_n\|. \end{split}$$

Let

$$\begin{split} s_n &= \|\pi_n - w_n\|,\\ \varsigma_n &= \varphi_n (1 - \delta),\\ \tau_n &= \left(1 - \varphi_n (1 - \delta)\right) \|\pi_n - S\pi_n\| + L\varphi_n \|\rho_n - S\rho_n\|. \end{split}$$

Since $\pi_n \to s^*$ as $n \to \infty$ it follows that

$$\|\pi_n - S\pi_n\| \le \|\pi_n - s^*\| + \|Ss^* - S\pi_n\|$$

$$\le \|\pi_n - s^*\| + \delta \|s^* - \pi_n\| + L\|s^* - Ss^*\|$$

$$= (1 + \delta)\|\pi_n - s^*\| \to 0, \text{ for all } n \in \mathbb{N},$$

and

$$\begin{split} &\|\rho_n - S\rho_n\| \leq \|\rho_n - s^*\| + \|Ss^* - S\rho_n\| \\ &\leq \|\rho_n - s^*\| + \delta \|s^* - \rho_n\| + L\|s^* - Ss^*\| \\ &= (1+\delta)\|\rho_n - s^*\| + L\|s^* - Ss^*\| \\ &= (1+\delta)\|\rho_n - s^*\| + L(1+\delta)\|s^* - Ss^*\| \\ &\leq (1+\delta)\delta \|\pi_n - s^*\| + L(1+\delta)\|s^* - Ss^*\| \\ &= (1+\delta)\delta \|\pi_n - s^*\| \to 0, \text{ forall } n \in \mathbb{N}, \end{split}$$

and therefore, $\lim_{n\to\infty}\|\rho_n-S\rho_n\|=\lim_{n\to\infty}\|\pi_n-S\pi_n\|=0$, that is, $\lim_{n\to\infty}\frac{\tau_n}{\varsigma_n}=0$. It follows from Lemma 1.8 that $\lim_{n\to\infty}\|\pi_n-w_n\|=0$. Also, we have

$$||w_n - s^*|| \le ||w_n - \pi_n|| + ||\pi_n - s^*||,$$

it implies that $\lim_{n\to\infty} ||w_n - s^*|| = 0$.

Conversely, let $w_n \to s^*$ as $n \to \infty$, we shall show that $\pi_n \to s^*$ as $n \to \infty$.

Using (1), (6) and (3), we obtain

$$||w_{n+1} - \pi_{n+1}|| = ||(1 - \varphi_n)w_n + \varphi_n Sw_n - \varphi_n Sw_n||$$

$$(1 - \varphi_n)\rho_n + \varphi_n S \rho_n \|$$

$$\leq (1 - \varphi_n) \|w_n - \rho_n\| + \varphi_n \|Sw_n - S\rho_n\|$$

$$\leq (1 - \varphi_n) \|w_n - \rho_n\| + \varphi_n \delta \|w_n - \rho_n\| +$$

$$\varphi_n L \| w_n - S w_n \|$$

$$\leq (1 - \varphi_n(1 - \delta)) \|w_n - \rho_n\| + \varphi_n L \|w_n - Sw_n\|.$$

Again, from (6), we obtain

$$||w_n - \rho_n|| = ||w_n - S\pi_n||$$

$$\leq ||w_n - Sw_n|| + ||Sw_n - S\pi_n||$$

$$\leq ||w_n - Sw_n|| + \delta ||w_n - \pi_n|| + L||w_n - Sw_n||$$

$$= \delta ||w_n - \pi_n|| + (1 + L)||w_n - Sw_n||.$$

Substituting (15) in (14), we have

$$||w_{n+1} - \pi_{n+1}|| \le (1 - \varphi_n(1 - \delta))||w_n - \pi_n|| + [(1 - \varphi_n(1 - \delta))(1 + L) + \varphi_n L]||w_n - Sw_n||.$$

Set

$$\begin{split} s_n &= \|w_n - \pi_n\|,\\ \varsigma_n &= \varphi_n(1-\delta),\\ \tau_n &= \left[\left(1-\varphi_n(1-\delta)\right)(1+L) + \varphi_n L\right] \|w_n - Sw_n\|. \end{split}$$

Since $\lim_{n\to\infty} w_n = s^*$ and $Ss^* = s^*$, it is clear that $\lim_{n\to\infty} ||w_n - Sw_n||$.

Therefore, all the conditions of Lemma 1.8 are provided, yield

$$\lim_{n\to\infty}||w_n-\pi_n||=0.$$

Since $\lim_{n\to\infty} \|\pi_n - s^*\| = 0$, we obtain $\|\pi_n - s^*\| \le \|\pi_n - w_n\| + \|w_n - s^*\| \to 0$. as $n \to \infty$. this completes the proof.

Keeping in mind the results in Gürsoy [23], Soltuz [24]-[25] and Chugh and Kumar [26], we can give the following corollary:

Corollary 2.3 Let $S: C \to C$ be an operator satisfying the condition of ACM (8) with $F(S) \neq \emptyset$. Also, if the initial values is the same, then the followings are equivalent:

- 1. The iterative process (1) converges to $s^* \in F_T$;
- 2. The iterative process (3) converges to $s^* \in F_T$;
- 3. The iterative process (4) converges to $s^* \in F_T$;
- 4. The iterative process (6) converges to $s^* \in F_T$.

Theorem 2.4 Let $S: C \to C$ be an operator satisfying the condition of ACM (8) with $F(S) \neq \emptyset$ and \tilde{S} is approximate operator of S. Let $\{\pi_n\}_{n\in\mathbb{N}}$ be a sequence generated by (6) with respect to S and $\{\tilde{\pi}_n\}_{n\in\mathbb{N}}$ be a sequence generated by

$$\begin{cases} \pi_1 \in \mathcal{C} \\ \tilde{\pi}_{n+1} = (1 - \varphi_n) \tilde{\rho}_n + \varphi_n \tilde{S} \tilde{\rho}_n \\ \tilde{\rho}_n = \tilde{S} \tilde{\pi}_n, \quad \forall n \in \mathbb{N}, \end{cases}$$
 (16)

where $\{\varphi_n\}_{n\in\mathbb{N}}\in[0,1]$ is a real sequence satisfying the following conditions:

(i)
$$1 - \varphi_n \le \varphi_n$$
,

(i)
$$1 - \varphi_n \le \varphi_n$$
,
(ii) $\sum_{n=1}^{\infty} \varphi_n = \infty$.

If $\tilde{s}^* \in F(\tilde{s})$ such that $\lim_{n \to \infty} \tilde{\pi}_n = \tilde{s}^*$, then we obtain

$$||s^* - \tilde{s}^*|| \le \frac{2\epsilon}{1-\delta}$$

Proof. From (16) and (6), we have

$$\pi_{n+1} - \tilde{\pi}_{n+1} = (1 - \varphi_n)(\rho_n - \breve{\rho}_n) + \varphi_n (S\rho_n - \breve{S}\breve{\rho}_n).$$

Thus

$$\|\pi_{n+1} - \tilde{\pi}_{n+1}\| = \|(1 - \varphi_n)(\rho_n - \breve{\rho}_n) + \|\tilde{\pi}_n\|_{2}$$

$$\varphi_n(S\rho_n - \breve{S}\breve{\rho}_n)\|$$

$$\leq (1-\varphi_n)\|\rho_n - \breve{\rho}_n\| + \varphi_n \left\|S\rho_n - \breve{S}\breve{\rho}_n\right\|$$

$$\leq (1-\varphi_n)\|\rho_n - \breve{\rho}_n\| + \varphi_n \big[\|S\rho_n - S\breve{\rho}_n\| +$$

$$||S\breve{\rho}_n - \breve{S}\breve{\rho}_n||]$$
 (17)

$$\leq (1 - \varphi_n) \|\rho_n - \breve{\rho}_n\| + \varphi_n \delta \|\rho_n - \breve{\rho}_n\| +$$

$$\varphi_n L \|\rho_n - S\rho_n\| + \varphi_n \epsilon$$

$$= (1 - \varphi_n(1 - \delta)) \|\rho_n - \breve{\rho}_n\| + \varphi_n L \|\rho_n - S\rho_n\| +$$

 $\varphi_n \epsilon$.

By using (16), (6) and (8), we obtain

$$\begin{aligned} &\|\rho_n - \breve{\rho}_n\| = \|S\pi_n - \breve{S}\widetilde{\pi}_n\| \\ &\leq \|S\pi_n - S\widetilde{\pi}_n\| + \|S\widetilde{\pi}_n - \breve{S}\widetilde{\pi}_n\| \\ &\leq \delta \|\pi_n - \widetilde{\pi}_n\| + L\|\pi_n - S\pi_n\| + \epsilon. \end{aligned}$$

Substituting (18) in (17), we get

$$\begin{split} \|\pi_{n+1} - \tilde{\pi}_{n+1}\| &\leq \left(1 - \varphi_n(1 - \delta)\right) \delta \|\pi_n - \tilde{\pi}_n\| + \\ \left(1 - \varphi_n(1 - \delta)\right) \epsilon \\ &+ \left(1 - \varphi_n(1 - \delta)\right) L \|\pi_n - S\pi_n\| + \varphi_n L \|\rho_n - S\rho_n\| + \\ \end{split}$$

 $\varphi_n \epsilon$.

Since $1 - \varphi_n \le \varphi_n$, we have

$$\|\pi_{n+1} - \tilde{\pi}_{n+1}\| \le (1 - \varphi_n(1 - \delta)) \|\pi_n - \tilde{\pi}_n\| + 1$$

$$\begin{split} \varphi_n[(1-\delta)\epsilon \\ &+ (1-\delta)L\|\pi_n - S\pi_n\| + L\|\rho_n - S\rho_n\| + \epsilon] \\ \leq \\ & \left(1 - \varphi_n(1-\delta)\right)\|\pi_n - \tilde{\pi}_n\| + \varphi_n(1-\delta)\frac{[L\|\pi_n - S\pi_n\| + L\|\rho_n - S\rho_n\| + 2\epsilon]}{(1-\delta)}. \end{split}$$
 we define

$$\begin{split} s_n &= \|\pi_n - \tilde{\pi}_n\|, \\ \varsigma_n &= \varphi_n (1 - \delta), \\ \tau_n &= \frac{[L\|\pi_n - S\pi_n\| + L\|\rho_n - S\rho_n\| + 2\epsilon]}{(1 - \delta)}. \end{split}$$

By Theorem 2.1, we note that $\lim_{n\to\infty} \|\pi_n - S\pi_n\| = \lim_{n\to\infty} \|\rho_n - S\rho_n\| = 0$. Because, both $\{\pi_n\}_{n\in\mathbb{N}}$ and $\{\rho_n\}_{n\in\mathbb{N}}$ converge to the fixed point of S.

Then, it follows from Lemma 1.7 that

$$0 \le \limsup_{n \to \infty} \|\pi_n - \tilde{\pi}_n\| \le \limsup_{n \to \infty} \frac{[L\|\pi_n - S\pi_n\| + L\|\rho_n - S\rho_n\| + 2\epsilon]}{(1-\delta)}.$$

So, the inequality (19) implies that

$$||s^* - \tilde{s}^*|| \le \frac{2\epsilon}{1-\delta}.$$

Theorem 2.5 Let $S: \mathcal{C} \to \mathcal{C}$ be an operator satisfying the condition of ACM (8) with $F(S) \neq \emptyset$ and the sequence $\{\pi_n\}_{n \in \mathbb{N}}$ defined by (6) with $\{\varphi_n\}_{n \in \mathbb{N}} \in [0,1]$ such that $\sum_{n=1}^{\infty} \varphi_n = \infty$. Then, the iterative procedure $\{\pi_n\}_{n \in \mathbb{N}}$ is S-stable.

Proof. Suppose that $\lim_{n\to\infty} \epsilon_n = 0$. Then, we shall establish that $\lim_{n\to\infty} y_n = s^*$. In view of ACM (8) and the fact that $\|Sy_n - s^*\| = \|Sy_n - Ss^*\|$,

$$\begin{split} \|y_{n+1} - s^*\| &= \left\| y_{n+1} - \left((1 - \varphi_n) \rho_n + \varphi_n S \rho_n \right) \right\| \\ &+ \left\| (1 - \varphi_n) \rho_n + \varphi_n S \rho_n - s^* \right\| \\ &\leq \epsilon_n + (1 - \varphi_n) \|\rho_n - s^*\| + \varphi_n \|S \rho_n - s^*\| \\ &\leq \epsilon_n + (1 - \varphi_n) \|\rho_n - s^*\| + \varphi_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - s^*\| + \varepsilon_n \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n - \delta \|\rho_n$$

 $L\varphi_n \| Ss^* - s^* \|$

$$\leq \epsilon_n + (1 - \varphi_n(1 - \delta)) \|\rho_n - s^*\| + L\varphi_n \|Ss^* - s^*\|$$

(20)

$$\|\rho_n - s^*\| = \|Sy_n - s^*\|$$

$$\leq \delta \|y_n - s^*\| + L\|Ss^* - s^*\|$$

(21)

Substituting (21) in (20), we have

$$||y_{n+1} - s^*|| \le \epsilon_n + (1 - \varphi_n(1 - \delta))(\delta ||y_n - s^*|| + L||Ss^* - s^*||)$$

$$+ L\varphi_n ||Ss^* - s^*||$$

$$\le \epsilon_n + (1 - \varphi_n(1 - \delta))\delta ||y_n - s^*|| + L(\varphi_n + (1 - \varphi_n(1 - \delta))L)||Ss^* - s^*||$$

Since $||Ss^* - s^*|| = 0$, we obtain

$$||y_{n+1} - s^*|| \le \epsilon_n + (1 - \varphi_n(1 - \delta))\delta ||y_n - s^*||.$$

Also, since $0 < (1 - \varphi_n(1 - \delta))\delta < 1$ and using Lemma 1.8, we obtain $\lim_{n \to \infty} y_n = s^*$.

Conversely, letting $\lim_{n\to\infty} y_n = s^*$, we shall establish that $\lim_{n\to\infty} \epsilon_n = 0$ as follows:

$$\epsilon_n = \|y_{n+1} - ((1 - \varphi_n)\rho_n + \varphi_n S \rho_n)\|$$

$$\leq \|y_{n+1} - s^*\| + \|s^* - ((1 - \varphi_n)\rho_n + \varphi_n S \rho_n)\|$$

 $\varphi_n S \rho_n) \|$

$$\leq \|y_{n+1} - s^*\| + (1 - \varphi_n)\|s^* - \rho_n\| +$$

 $\varphi_n || s^* - S \rho_n ||$

$$\leq \|y_{n+1} - s^*\| + (1 - \varphi_n)\|s^* - \rho_n\|$$

$$+ \varphi_n \delta \|s^* - \rho_n\| + \varphi_n L \|Ss^* - s^*\|$$

$$\leq \|y_{n+1} - s^*\| + (1 - \varphi_n(1 - \delta))\|s^* - s^*\|$$

 $\rho_n \|$

$$\leq ||y_{n+1} - s^*|| + (1 - \varphi_n(1 - \delta))||s^* -$$

 Sy_n

$$\leq ||y_{n+1} - s^*|| + \delta (1 - \varphi_n (1 - \delta))||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* - s^*||s^* -$$

 $y_n \|$

Since $\lim_{n\to\infty} y_n = s^*$, it follows that $\lim_{n\to\infty} \epsilon_n = 0$. Thus, the iterative procedure $\{\pi_n\}_{n\in\mathbb{N}}$ defined by (6) is S-stable.

Theorem 2.6 Let $S: C \to C$ be an operator satisfying the condition of ACM (8) with $F(S) \neq \emptyset$ and two sequences $\{\pi_n\}_{n \in \mathbb{N}}$ and $\{k_n\}_{n \in \mathbb{N}}$ defined by (6) and (5) (respectively) with $\{\varphi_n\}_{n \in \mathbb{N}} \in [0,1]$ such that

 $\sum_{n=1}^{\infty} \varphi_n = \infty$. Also, if the initial points of $\{\pi_n\}_{n \in \mathbb{N}}$ and $\{k_n\}_{n \in \mathbb{N}}$ are equal, then the sequences $\{\pi_n\}_{n \in \mathbb{N}}$ and $\{k_n\}_{n \in \mathbb{N}}$ converge to the fixed point s^* of S with the same rate of convergence.

Proof. From Theorem 2.1, we obtain

$$\|\pi_{n+1} - s^*\| \le \delta (1 - \varphi_n (1 - \delta)) \|\pi_n - s^*\|$$

$$\|\pi_{n+1} - s^*\| \le \delta^{n+1} \|\pi_0 - s^*\|_{i=0}^n \left(1 - \varphi_i(1 - \delta)\right) = a_n$$

and

$$||k_{n+1} - s^*|| \le \delta (1 - \varphi_n (1 - \delta)) ||k_n - s^*||$$

$$\|k_{n+1} - s^*\| \le \delta^{n+1} \|k_0 - s^*\|_{i=0}^n \left(1 - \varphi_i(1 - \delta)\right) = b_n$$

for all $n \in \mathbb{N}$.

By using Definition 1.4, we have

$$\lim_{n \to \infty} \frac{\|a_n - 0\|}{\|b_n - 0\|} = \lim_{n \to \infty} \frac{a_n}{b_n}$$

$$= \lim_{n \to \infty} \frac{\delta^{n+1} \|\pi_0 - s^*\|_{i=0}^n (1 - \varphi_i(1 - \delta))}{\delta^{n+1} \|k_0 - s^*\|_{i=0}^n (1 - \varphi_i(1 - \delta))}$$

$$= 1.$$

By (2) in the Definition 1.4, The MP (6) and PM (5) iterative schemes have the same rate of convergence.

Example 2.7 [16] Let $S: [0,2] \to [0,2]$ be a mapping such that $S = (x+2)^{\frac{1}{3}}$ and $\varphi_n = \frac{3}{4}$ for all $n \ge 15$ otherwise $\varphi_n = 0$. It is easy to see that S is a ACM with 1,521379706804570 $\in F(S)$ and the Example 2.7 satisfies for all conditions of Theorem 2.1. If $\{\pi_n\}_{n\in\mathbb{N}}$ (6) and $\{k_n\}_{n\in\mathbb{N}}$ (5) have the same initial condition, then we show that $\{\pi_n\}_{n\in\mathbb{N}}$ (6) and $\{k_n\}_{n\in\mathbb{N}}$ (5) iterative schemes have same the rate of convergene to fixed point 1,521379706804570 of S.

Table 1: Comparison rate of convergence between MP and PM iteration methods

MP	PM
1,990000000000000	1,990000000000000
1,544499673873150	1,544874898372010
	1,990000000000000

<i>x</i> ₃	1,522569021765430	1,522589308040770
<i>x</i> ₄	1,521441018705160	1,521442067123600
		·
<i>x</i> ₁₁	1,521379706804630	1,521379706804630
<i>x</i> ₁₂	1,521379706804570	1,521379706804570

It is shown by Example 2.7 that iterative method MP and PM iterative schemes have the same rate of convergence.

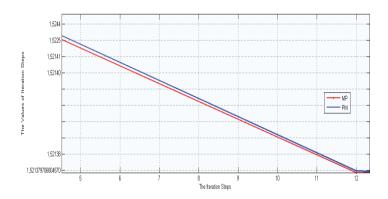


Figure 1: Convergence behavior of MP and PM iterative schemes.

Figure 1 shows the rate of convergence between the MP and PM iterative schemes.

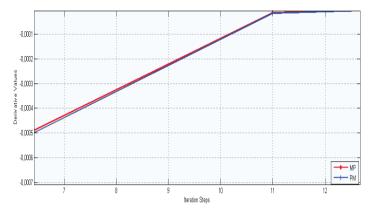


Figure 2: Derivative curves

Figure 2 is given to illustrate the changes between successive steps. A good way to show a relationship between two steps of the iteration scheme is to use Figure 2.

3. Conclusion In this study, a new iterative procedure is defined and this iteration procedure is compared with the Picard-Mann iterative procedure. It is evident that these two iterations have the same convergence rate although they are completely independent of each other. Since Picard-Mann iterative procedure is the fastest among two-step fixed point iterative schemes, the same is valid for the Mann-Picard iterative procedure.

4. References

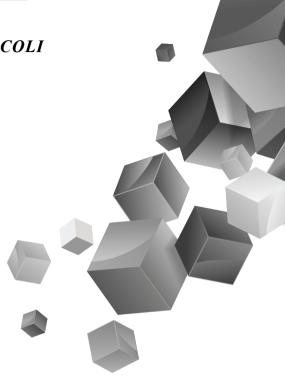
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INACTIVATION OF E. COLI

BY SONICATION



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Introduction

Conventional thermal inactivation methods such as pasteurization and sterilization are the most common techniques currently used to inactivate microorganisms in food products. However thermal process is very effective, it degrades nutritional and sensory quality of food. For this reason, new preservation techniques are needed. One of the new approaches is combined factors for inactivation or inhibition of microorganisms in foods. This techniques uses several factors (water activity/aw, pH, temperature, inhibitors etc.) at mild level instead of one factor (such as heat) at high level to control microbial activity in foods. Some emerging factors are ultrasonic waves, pulse-electric field, pulse-light, lacto peroxidase system (LPS) and high pressure (Leistner, 1996). Although sonication is used to disintegrate cells in many microbial or cellular studies since many years, it has newly discovered as inactivation factor in food preservation. Sonication is a promising alternative technique to classical thermal process especially in inactivation of foodborne pathogens.

As an early study, Hedges et al. (1980) described mechanisms of inactivation of ultrasound on *E. coli*. In the recent years, several studies published about inactivation of microorganisms with ultrasound.

E. coli is an indicator organism for most of the foodborne pathogens and hygienic condition of foods, also many strains of this species are pathogen. These strains can contaminate raw fruit and vegetable juices and other non-thermally preserved liquid food. E. coli O157:H7 can contaminate apple in the garden and survive in apple cider (Stopforth et al., 2004). Other non-pasteurized fruit and vegetable juices may be contaminated in a similar way. Sonication can be a used as a altenative process to inactivate E. coli and similar pathogens in this kind of liquid foods. Ugarte-Romeo et al. (2006) showed that, power ultrasound treatment can provide the 5-log reduction for E. coli in apple cider required by the FDA and sonication may be useful as an alternative to traditional pasteurization method. They also showed that ultrasound causes holes on E. coli surfaces and causes cell death. Although inactivation of ultrasound at sub-lethal and lethal temperatures is well studied, effect of pH and temperature combination in the inactivation of E. coli and isnot investigated.

The aim of this study was to determine the effect of pH (pH 4.0 and 7.0) and non-lethal temperatures on the inactivation of *E. coli* by ultrasound.

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Materials and methods

Test organism

The strain of *E. coli* LMG 8223 (ATCC 25922) used in this study was purchased from the Belgian Co-ordinated Collections of Microorganisms (BCCM), K. L.Ledeganckstraat, 35 B-900 Gent, BELGIUM as lyophilized culture. Lyophilized culture tube was opened aseptically and the content transferred to Nutrient Broth (NB) tube (Merck, Germany) and incubated at 37 °C for 24 h. Then NB culture transferred in to slants of Nutrient Agar (NA; Merck). NA slant cultures were stored at 4 °C until use.

Media

Tryptone-yeast extract broth (TYEB- 10 g/l tryptone, 5 g/l yeast extract, 5 g/l sodium chloride) was used for growth of culture. Violet red bile agar (Merck) was used for enumeration. Citrate—phosphate buffers at pH 4.0 and 7.0 were used as sonication medium.

Preparation of cell suspension

The culture was inoculated into 50 ml TYEB in a 500 ml flask and incubated for 24 h at 37 °C in a shaking incubator (130 rpm). The cells were harvested by centrifugation at 6000 rpm for 10 min and re-suspended and washed 3 times in 0.85% saline, than resulting suspension stored at 4 °C. The cell number of this stock suspension was determined with standard plate count technique before each replicate study.

Inactivation studies

Sonication was carried out using a VC-750 ultrasound generator Vibracell® WCX 750 model ultrasonic processor at frequency of 20 kHz. A solid probe, 13 mm diameter is used. Two AED was used as 3.0 and 7.3 W/ml in treatments. A 10-mL jacketed vessel was used to hold the samples. Temperature of the content of the vessel was controlled by a refrigerated circulating bath. After each treatment, all equipment was rinsed with sterilized distilled water and vessel and probe were sterilized before and after each replicate of study.

At the beginning each trial, 9.9 ml citrate-phosphate buffer was pipetted in to a sterile tube and left for 3 min in the water bath to bring the medium temperature to the test temperature, then inoculated with 0.1 ml of stock *E. coli* suspension, stirred on a vortex for a few seconds and poured into the vessel which has reached to the test temperature. Probe was immersed into the vessel, carefully centered and then sonication started immediately. Cell suspensions were subjected to sonication 2-6 min and sampled at predetermined intervals. The 0.5-mL portions of aliquot were taken from vessel as samples and serially diluted in sterile 0.85% saline water (1:10). Drop plate technique was used for enumeration of survival cells. Replicate

0.02 ml aliquots were taken from dilutions and plated on VRBA. Plates were incubated at 37 °C for 24 h. Viable counts were determined as same as standard plate count technique.

Experimental design

All combinations of conditions were selected from the following: temperature (20, 30 and 40 °C); pH (4.0 and 7.0); sonication (0, 3.0 and 7.3 W/ml). Each experiment was conducted in duplicate, and the experimental design was; ((two temperature \times 2 pH \times 2 AED) \times 2).

Construction of survival curves

First order reaction kinetic was used for survival studies. Inactivation coefficients were calculated from all survival curves by linear regression. Microbiologists are familiar log transformation rather than ln transformation. For this reason logarithmic order of death of microorganisms was expressed by the fallowing equation as given by Xiong et al. (1999):

$$Log \frac{N(t)}{N_0} = -\frac{t}{D}$$

Where, t is treatment time, N_t is the number of the surviving cells after t minutes of treatment, N_0 is the number of the living cells at t=0 and D is decimal reduction time.

This equation can be referred as fallows:

$$Log(N_t) = Log(N_0) - t * \frac{1}{D}$$

"1/D" is calculated from linear regression line of survival plot.

For the Arrhenious plot, inactivation rate constants are expressed as k as follows:

$$N = N_0 \times e^{-\hbar}$$
 or \hbar $(N) = \hbar$ $(N_0) - \hbar$

Where, N was the number of live E. coli cells after ultrasonic irradiation, N_0 was the initial number of the E. coli cells, k was the inactivation rate constant, and t was the treatment time.

Relationship between k and can be expressed as fallows:

$$k = 2.303 \times \frac{1}{D}$$

Process duration for 5D reductions of survivals were calculated from 1/D values

Results and discussion

Inactivation of E. coli for combination of pH and temperature without sonication application

Inactivation of E. coli for non-sonic treatments at 20-40 °C and pH 4.0 and 7.0 were negligible (Fig. 1). Reductions in the number of the survivals were changed between from no-reduction to 0.8 log in 80-160 min of treatment durations. Average of inactivation rate constants (1/D values) of the curves of non-sonic treatments was 0.0045 ± 0.0074 min⁻¹. These coefficients imply that only 1-log reduction may be observed in population of E. coli at citrate buffer, pH 4.0-7.0 in 3-5 hours of incubation at 20-40 °C. This reduction is not important in magnitude for food preservation process and probably arises from lack of maintain energy in the buffer medium, so that could not be observed in food products. Temperature and pH differences were also found statistically unimportant (p>0.05) on the reduction in non-sonic treatment. Roering, et al. (1999) reported that E.coli O157:H7 reduction level was 5 log unit for pasteurized and non-pasteurized apple cider in 21 days (stored at 4 and 10 °C). The pH and the Brix of the apple cider were between 3.3 - 3.5 and 12-14 respectively. This is a slower reduction compared to our study, probably caused by protective effect of low storage temperature or adaptation of cells to lower pH. Steenstrup and Floros (2002) predicated D₅₀ values to be 71 min for apple cider without additives. This result is more comparable with our non-sonic treatments.

Inactivation of E. coli with the sonications at the level of 3.0 and 7.3 W/ml at 4.0 and 7.0 pH, and 20, 30 and 40 °C combinations

Sonication at AED of 3.0 and 7.3 W/ml resulted 3-4 log unit reduction in the initial population of *E. coli* in 2 - 6 and 1 - 2 minutes respectively. The coefficient of inactivation kinetic was calculated from survival plot by linear regression model, as Log₁₀ (survival/ml) versus treatment time. Coefficients of inactivation of replicate treatments were not statistically different (p>0.05). Two replicate coefficients were averaged and then D value and 5D processing time were calculated. The results are presented in Table 1.

According to results, sonication was the most effective factor on the inactivation rate constant. Treatment at 20 °C and 3.0 W/ml level of sonication resulted rate constants as 0.61 and 0.89 min⁻¹ for pH 4.0 and 7.0

respectively. When sonication energy was increased from 3.0 W/ml to 7.3 W/ml (243 %), reduction rate constants mostly increased more then this ratio. Raso et al. (1997) reported that resistance of *Yersinia enterecolitica* decreased exponentially when the ultrasound amplitude was increased linearly.

Even though pH and level of AED were same, in increasing temperature, inactivation also increased (Fig. 2). However in the same temperature range of inactivation in non-sonic treatments was not show any difference. Similarly, Ross et al. (2000) reported that lower temperatures reduced the rates of death at non-thermal inactivation.

According to results, pH slightly affects the inactivation rate constants. The rate constants at pH 7.0 were slightly higher at each sonication levels and each temperature combinations then those at pH 4.0. This difference was found as statistically important (p=0.03). This situation can be seen in Fig.3. Normally, decrease in pH increases inactivation rate constant at same temperatures in both of non-thermal treatments (Scandamis and Nychas, 2000) and thermal treatments (Steenstrup and Floros, 2002). Some authors were reported similar findings, as some *Enterobacteriaceae* were more sensitive at neutral pH then low pH to high hydrostatic pressure (HHP) treatments (Raso and Barbosa-Cánovas, 2003).

Table 1. Inactivation coefficients, D and 5D values of E. coli treated with sonication at pH 4 and 7 and 20, 30 and 40 $^{\circ}$ C.

		Temperatures °C						
Acoustic energy density AED (W/ml)	рН	20		30		40		
		C.I. (log/ min)	D 5D (min) (min)	C.I. ⁽¹⁾ (log/ min)	D 5D (min) (min)	C.I. ⁽¹⁾ (log/ min)	D (min)	5D (min)
3	4.0	0,61	1,63 8,14	0,80	1,24 6,22	1,58	0,63	3,17
	7.0	0,89	1,12 5,59	1,19	0,84 4,21	1,75	0,57	2,85
7.3	4.0	1,89	0,53 2,65	2,93	0,34 1,71	5,41	0,18	0,92
	7.0	2,15	0,46 2,32	3,22	0,31 1,55	5,78	0,17	0,87
Increase in C.I. (1) (%)	4.0	309%		366%		342%		
	7.0	241%		270%		330%		

⁽¹⁾Sonication energy was increased 243 % in the second energy level.

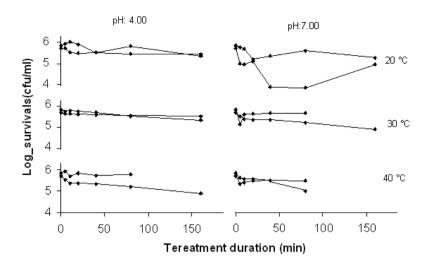


Fig.1. Inactivation of E. coli for non-sonic treatments at 20-40 °C and pH 4.0 and 7.0.

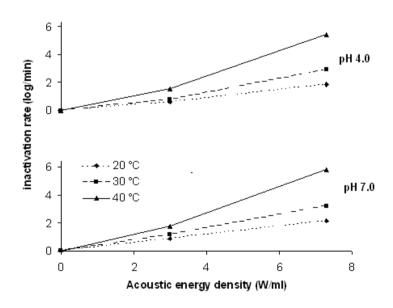


Fig. 2. Inactivation of E. coli with sonic treatments (3.0 and 7.3 W/ml AED) at different pH (4.0 and 7.0) and temperature (20, 30 and 40 °C) combinations.

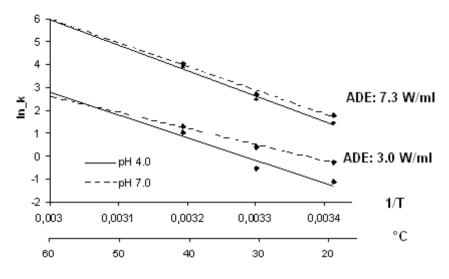


Fig. 3. Arrhenius plot of inactivation rates at pH 4.0, 7.0, AED level of 3.0, and 7.3 W/ml

Conclusion

In this study, ultrasonic inactivation of Escherichia coli LMG 8223 using non-thermal temperatures and non-lethal pH ranges was investigated. Death rates were measured for a combination of three temperatures (20°C, 30°C and 40°C), two pH levels (4.0 and 7.0) and two Acoustic Energy Densities, AED (3.0 and 7.3 W/ml). Inactivation of the E. coli cells by ultrasonic irradiation was showed first-order behavior. The inactivation rate constants calculated as log/min or 1/D. Inactivation of E. coli for nonsonic treatments was negligible. In the sonic treatments, inactivation rates are increased with increasing amplitude of the sonication. The maximum inactivation rate was determined as 5.78 log/min at 7.3 W/ml AED level at 40 °C for pH 7.0. According to result, treatment duration for 5D reduction of E. coli was 52 s for 10 ml suspension with our laboratory type equipment (750 W). Although this inactivation process is too far from industrial expectations, by increasing the thermal component of inactivation and using industrial type equipment, sonication could be an alternative food safety technique for heat sensitive food products.

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BOTANICAL DESCRIPTION, CHEMICAL COMPOSITION AND MEDICAL USE OF ZINGIBER OFFICINALE (ROSCOE) (GINGER)

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1. Botanical Definition and Features of *Zingiber officinale* (Roscoe) (Ginger)

Based on the findings of the World Health Organization based on the studies conducted with medicinal plants in 91 countries; That number for the benefit of plants for medicinal purposes and Turkey are reported to be about 20,000 around 140 (Kaplan, 2005). On the contrary (Baytop, 1999) stated that these numbers are around 500 for our country and around 100,000 for the whole world.

One of the most recent examples of these medicinal plants, which have many uses, is ginger. It was described by the English botanist William Roscoe in 1807 and was named *Zingiber officinale* by Fostor (Fostor, 2000).

Scientific classification of ginger		
Regnum	Plantae	
Phylum	Magnoliophyta	
Clasis	Liliopsida	
Ordo	Zingiberales	
Familia	Zingiberaceae	
Genus	Zingiber	
Species	Z. officinale	

Table 1. Scientific classification of ginger

(Rehman et al., 2011) also stated; Ginger is perennial, herbaceous and rhizome. It extends up to a height of 90 cm. Rhizomes aromatic, thick lobed, pale yellowish, simple bedding alternate distichous narrow oblong lanceolate leaves. It develops several lateral shoots in clusters; When the plant matures, it begins to dry. The leaves are long and 2 - 3 cm wide, sheath-based, with a blade gradually tapering to a point. Inflorescence solitary has oblong cylindrical spikes with lateral radical stems. Flowers are rare, rather small, calyx superior, gamosepalous, triangular, open cleavage on one side, spear-shaped double unequal rectangular greenish segments with three corollas (Kawai, 1994).



Figure 1. Zingiber officinale root, rhizome, leaf and flower status [14].

It is a plant that grows throughout Southeast Asia and China. It is also cultivated in some Africa countires, Austria, Japan, and Jamaica. Ginger has long been used as a spice and medicine in India and China (Sasidharan, 2010).

Throughout history, ginger has been used frequently, especially from its root and rhizome, for its high medicinal value and the flavor it adds to foods.



Figure 2. Zingiber officinale rhizome (Rehman, et al., 2011).

2. Chemical composition and nutritional content of ginger:

Zingiber officinale has many fat, wax, carbohydrate, vitamin and mineral content. Ginger rhizomes have an effective enzyme known as zingibain. (Connell, 1969).

As stated in the article [8,30] by (Pour et al., 2014), types of volatile oils in ginger; It consists of very small proportions bisabolene and b-sesquiphellandrene and farnesene (10%), curcumene (18%), mainly zingeberene (35%), sesquiterpene hydrocarbons. There are more than 40 different mono-terpenoid hydrocarbons with a small percentage, of which 1,8-cineol, neral, geraniol, linalol and borneol are the most abundant. According to (Pour et al., 2014), These active ingredients of ginger give it its unique aroma and taste.

As stated in [8,30], ginger contains biologically active ingredients such as zingerone, gingerol, paradoller and shogaol in its chemical content. And these create a feeling of warmth in the mouth. (Connell, 1969), on the other hand, gingerols have been defined as the main effective ingredients freshly dug out of the ground ginger rhizome. It has also been emphasized that when ginger is dried, the first prominent ingredient is shogaols. (Pour et al., 2014) stated that the paradol, which resembles gingerole, is formed by hydrogenation of the shogoal.

(Hasan et al, 2012) stated that ginger basically has the active ingredients gingerol, zingiberen, β -bisabolen, α -farnesen, shogaol, β -sesquiphellandren and α -curcumene.

Below are the formulas of some of the best known of these active ingredients.

Figure 3. Gingerol [7]

Figure 4. Shogaol [7]

Figure 5. Zingeron [7]

(Adel, 2010), in their study where they revealed the chemical content of ginger, they collected the component names and quantities they obtained in the table 2 and table 3 below. Accordingly, they stated that due to the components that ginger has, It can be said to be both a good source of

antioxidants and nutrients. As it is stated in Table 2, ginger besides high amounts of protein, carbohydrate and fat; It also contains many inorganic compounds that are very important for human health such as calcium, phosphorus and iron.

Component Name	Amount of	Component Name	Amount of
Protein	5,1 g/100 g	Iron	8,0 mg
Water insoluble fiber	%23,2	Phosphorus	174 mg
Water soluble fiber	%25,5	Zinc	0,9 mg
Oil	3,72 g	Calcium	88,4 mg
Moisture	%15,02	Manganese	9,13 mg
Carbohydrate	38,35 g	Chromium	70μg
Ash	3,85 g	C Vitamin	9,3 mg
		Total carotenoid	79 mg

Table 2. *Nutritional composition of 100 g of ginger [2].*

Table 3: *Ingredients and antioxidant activity of Zingiber officinale.* [2].

Component Name	Amount of
Total Phenols	840 mg
Tanin	1,51 g
Flovanoids	2,98 g
Total antioxidant activity	73529 μmol/g

As indicated in the table below, ginger root contains many vitamins that we need for healthy growth.

Table.4 The vitamin content of ginger [29].

Vitamin Name	Raw ginger (root) (in 100 g)
Riboflavin (B2)	0,034 mg
Panthenic acid (B5)	0.203 mg
Niacin (B3)	0,75 mg
Vitamin E	0.26 mg
Vitamin B6	0.16 mg
Folate (B9)	11 μg
Thiamine (B1)	0,025 mg
C Vitamin	5 mg

Due to these high nutritional properties of ginger; Although it is consumed as a sweetener all over the world, it is widely used in the food, beverage and confectionery industries. It is especially used in pickles, chutney, marmalade, wine, ginger beer, and liqueur [28].

3. Use of Ginger in Medical Field and Its Health Effects

Although drugs developed synthetically to treat diseases are widely used today; Throughout history, medicinal plants accompanied this purpose and offered an alternative to modern medicine. Moreover, according to (Kaplan, 2005); medicinal plants increasingly; They often replace artificial compounds because they only have a single effect and have multiple side effects.

According to (Güceyü et al., 2019), Zingiber officinale has been widely used in the geography to eliminate bacterial infections since ancient times due to its antimicrobial activity. (Grzanna et al., 2005) stated that ginger is used in the treatment of stomach aches and asthma, especially in China and India. Because ginger has active compounds and these compounds show different effects; It has also been used in colic, swelling and indigestion, ulcer, hypercholesterolemia and burn ailments. In addition, there are documents showing its use in the treatment of depression, liver toxicity and impotence (Jahromi, 2009).

Ginger has the potential to treat diabetes, cardiovascular disorders (hypertension and atherosclerosis), digestive health (constipation, indigestion and ulcers), arthritis and rheumatic diseases, vomiting, and cancer. It also has anti-inflammatory (anti-inflammatory) and antioxidative activities that restrain aging. In addition, it is known that its antimicrobial potential may be helpful in the treatment of infectious diseases [16, 26].

The use of ginger as an anti-inflammatory, pharmacological activities, treatment of rheumatoid and osteoarthritis has been documented. Zingiberen, one of the active ingredients of ginger, increases the effect of all these. As a result, ginger has a wide use in medicine, both as a single medicine and as a combined medicine. [23].

According to (Mashhadi et al., 2013); Medicinally important additives of ginger are now well-known. Ginger has therapeutic effects in a variety of ways, such as immuno-nutritional and anti-inflammatory. With its anti-inflammatory properties, ginger can reduce muscle pain after movement. Likewise, the cancer-preventing effect of ginger has been demonstrated by scientific studies; Active components of ginger, such as gingerol, shogaol and paradols, can inhibit various cancers with their effectiveness in angiogenesis and metastasis, apoptosis induction, and cell cycle progression. Ginger also continues to be used to treat cardiovascular diseases and gastrointestinal system disorders.

According to (Abdalla, 2018); In addition to the ginger rhizome being consumed as a spice and finding many uses in foods; With numerous industrial and traditional medicine research studies, the existence of antibacterial effect potential has been revealed. While 2 negative results were reported in 40 articles published on the antibacterial properties of ginger; In 38 of these studies, they were reported to respond effectively to the majority of the bacteria tested. For this reason, there is a need for advanced multidisciplinary studies to be conducted to investigate the antibacterial effectiveness of ginger and to eliminate contradictions.

Again according to (Abdalla, 2018); ginger is a popular medicinal herb. It has been confirmed to be antibacterial by many medical studies. In the study, It has been determined that ginger rhizome stops bacteria thanks to its essential oils. However, in order to reveal this efficiency much more clearly, it should be supported by multidisciplinary studies of microbiology, molecular physiology, pharmaceutical and medical sciences.

There is an interesting study by (Macit et al., 2019) on animal obesity. Accordingly, they revealed the important effects of ginger on obese animals and energy metabolism. However, unfortunately, the mechanism of action for these conditions in humans has not yet been revealed. When ginger is applied, it has been emphasized that the results of studies evaluating the effects of people on body weight are not very realistic. Because in affecting thermogenesis and respiratory coefficient; Various factors may play a role, including differences in active ingredients between studies, low levels of administration, differences in application times, and variations in the highest tolerated dose. For these reasons, consumption of ginger continuously and for long periods can affect both body weight and thermogenesis by increasing or decreasing values. It will be appropriate to re-evaluate these situations with subsequent studies. In order to reveal the relationship between obesity and energy metabolism in ginger applications; There is also a need for studies to determine and evaluate the active ingredient and supplementation time, optimum dose, and signal molecule pathways that affect the possible effects of long-term use.

(Savaşer et al., 2019) in their study to examine the effects of powdered ginger added to trout diets on their growth and immune systems as a nutritional supplement; They determined that ginger had positive effects especially on immune system parameters. The immune systems of the trout were negatively affected by the use of ginger at doses of 5.0 g / kg and above. However; They observed that after 0.5, 1.0 and 2.5 g / kg doses of ginger powder nutritional supplements, it made positive contributions to the body defense system.

In their study on laying hens (Gürbüz & Salih, 2017); when applied as a dietary supplement using a mixture of ginger powder and sumac (*Rhus*

coriaria L.); They observed that egg yolk tends to lower cholesterol levels in laying hens and stated that this mixture can be used as a feed additive.

(Akbulut, 2019) in their studies to determine the antiobesity effect using nicotine and ginger on obese rats; They observed that ginger was more effective than nicotine in treating obesity by inhibiting the glycogen increase. Thus, ginger has been shown to cause more weight loss in the treatment of obesity.

In our country, the usage areas of ginger among the people are quite many and it is widely used. Because it is a powerful anti-inflammatory, it can be used to prevent inflammation and strengthen the immune system. In addition, its anti-cold and anti-cold effect; prevention of vomiting (especially in early pregnancy and in chemotherapy to prevent nausea and vomiting) use as a natural pain reliever in severe pain; digestive effect; It has many uses in our daily lives with its many effects that reduce lipid and glucose levels.

4. Conclusion

Experiences gained from generation to generation, knowledge about which organs of medicinal plants, in what ways and proportions; They offer an alternative treatment to modern medicine in our age. In addition to the use of these plants in many areas such as food, spices, food additives; After realizing that they are also effective in treating some diseases, their use is increasing all over the world.

Plants used in the medical field are of very high economic importance all over the world. Zingiber officinale is one of these plants. Because it has both health and nutritional values. Ginger also appears as an important spice and sweetener all over the world. Heart protective, anti-inflammatory, antimicrobial, antioxidant, anticancer, etc. draws attention with its features. For these reasons, scientists need to do more scientific studies on ginger for drug production [11].

Due to the recent interest in herbal medicines, the values of medicinal plants have increased even more. Based on the historical and public use of medicinal plants, first their bioactivity and then their phytochemical evaluation are made. Mostly, some herbal medicines have a higher sales rate than their synthetic counterparts in Europe. The factor in the increasing use of medicinal plants among the public is the pharmacological studies conducted on this subject that direct their use. One of these plants is ginger, which has been used in treatments from ancient times to the present. The historical uses of ginger confirm its modern therapeutic application. In clinical and phytochemical studies, it confirms this use and reveals that ginger is a safe and effective drug [19].

The contributions of ginger to our health are emerging more and more every day. Both the increase in the frequency of use and the medical studies conducted on it; will make ginger even more popular and effective.

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